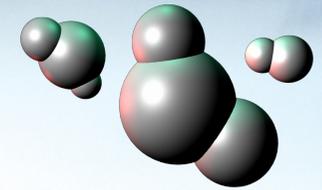


DFT+DMFT and non-periodic systems

Cedric Weber

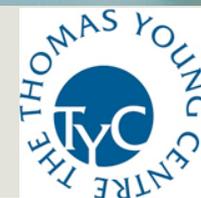


EPSRC



ICAM : DIGITAL DESIGN OF MATERIALS

School of natural science
Department of Physics



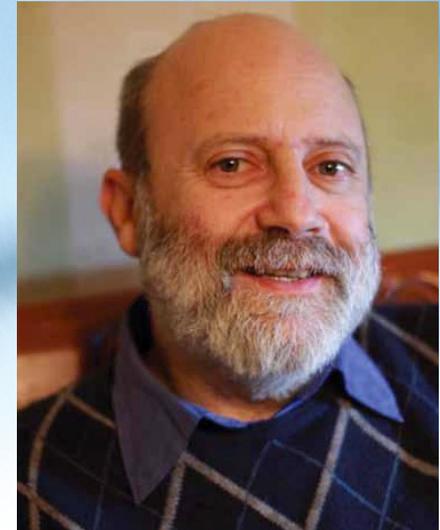
KING'S
College
LONDON



Peter B. Littlewood



Mike C. Payne



Gabriel Kotliar



David D. O'Regan



Nicholas D. M. Hine

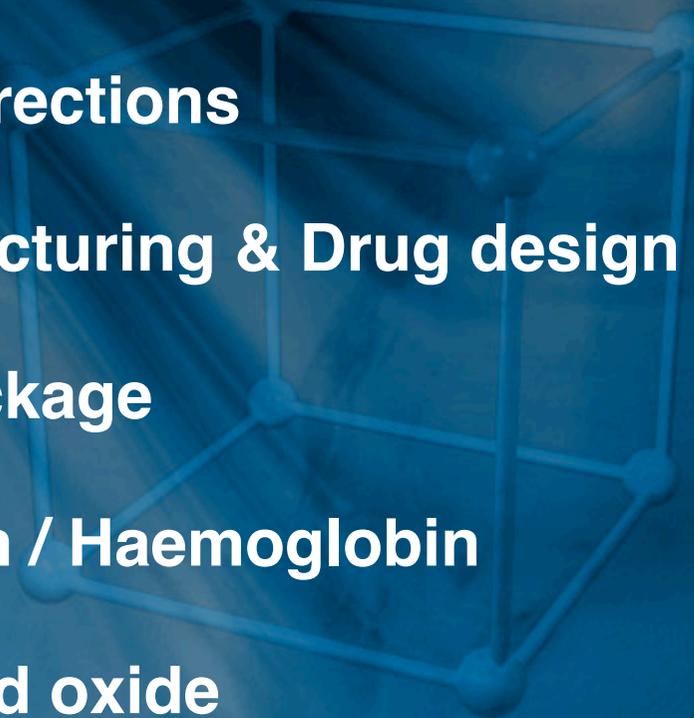
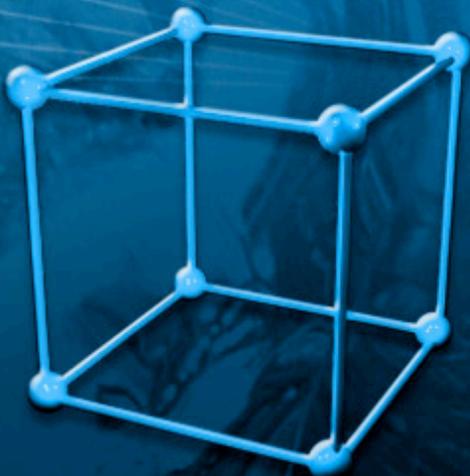


Daniel Cole



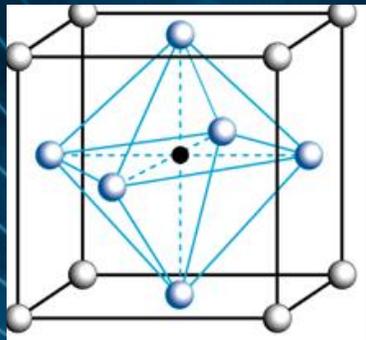
Outlines

- Strongly correlated systems
- Breakthrough in the last decades
- Promising directions
 - Nano-structuring & Drug design
- TOSCAM package
 - Myoglobin / Haemoglobin
 - Disordered oxide
 - Self-assembly
- Challenging Obstacles / Future progress
- Outlooks



Strongly correlated systems

transition metal ions



ion + oxygen cage = transition metal oxide

THE ELEMENTS

1 H Hydrogen	2 He Helium																	3 Li Lithium	4 Be Beryllium	5 B Boron	6 C Carbon	7 N Nitrogen	8 O Oxygen	9 F Fluorine	10 Ne Neon	11 Na Sodium	12 Mg Magnesium	13 Al Aluminum	14 Si Silicon	15 P Phosphorus	16 S Sulfur	17 Cl Chlorine	18 Ar Argon
19 K Potassium	20 Ca Calcium	21 Sc Scandium	22 Ti Titanium	23 V Vanadium	24 Cr Chromium	25 Mn Manganese	26 Fe Iron	27 Co Cobalt	28 Ni Nickel	29 Cu Copper	30 Zn Zinc	31 Ga Gallium	32 Ge Germanium	33 As Arsenic	34 Se Selenium	35 Br Bromine	36 Kr Krypton																
37 Rb Rubidium	38 Sr Strontium	39 Y Yttrium	40 Zr Zirconium	41 Nb Niobium	42 Mo Molybdenum	43 Tc Technetium	44 Ru Ruthenium	45 Rh Rhodium	46 Pd Palladium	47 Ag Silver	48 Cd Cadmium	49 In Indium	50 Sn Tin	51 Sb Antimony	52 Te Tellurium	53 I Iodine	54 Xe Xenon																
55 Cs Cesium	56 Ba Barium	57 La Lanthanum	58 Ce Cerium	59 Pr Praseodymium	60 Nd Neodymium	61 Pm Promethium	62 Sm Samarium	63 Eu Europium	64 Gd Gadolinium	65 Tb Terbium	66 Dy Dysprosium	67 Ho Holmium	68 Er Erbium	69 Tm Thulium	70 Yb Ytterbium	71 Lu Lutetium																	
87 Fr Francium	88 Ra Radium	89 Ac Actinium	90 Th Thorium	91 Pa Protactinium	92 U Uranium	93 Np Neptunium	94 Pu Plutonium	95 Am Americium	96 Cm Curium	97 Bk Berkelium	98 Cf Californium	99 Es Einsteinium	100 Fm Fermium	101 Md Mendelevium	102 No Nobelium	103 Lr Lawrencium																	
104 Rf Rutherfordium	105 Db Dubnium	106 Sg Seaborgium	107 Bh Bohrium	108 Hs Hassium	109 Mt Meitnerium	110 Ds Darmstadtium	111 Rg Roentgenium	112 Uub Ununbium	113 Uuq Ununquadium	114 Uup Ununpentium	115 Uuh Ununhexium	116 Uus Ununseptium	117 Uuo Ununoctium	118 Og Oganesson																			

V Cr Mn Fe Co Ni Cu

VO₂ Room temperature MIT

La_{1-x}Sr_xMnO₃ Colossal Magnetoresistance

Li_xCoO₂, Na_xCoO₂ Battery materials Thermoelectrics

La_{1-x}Sr_xCuO₄ High temperature superconductor

Failure of band picture (DFT)

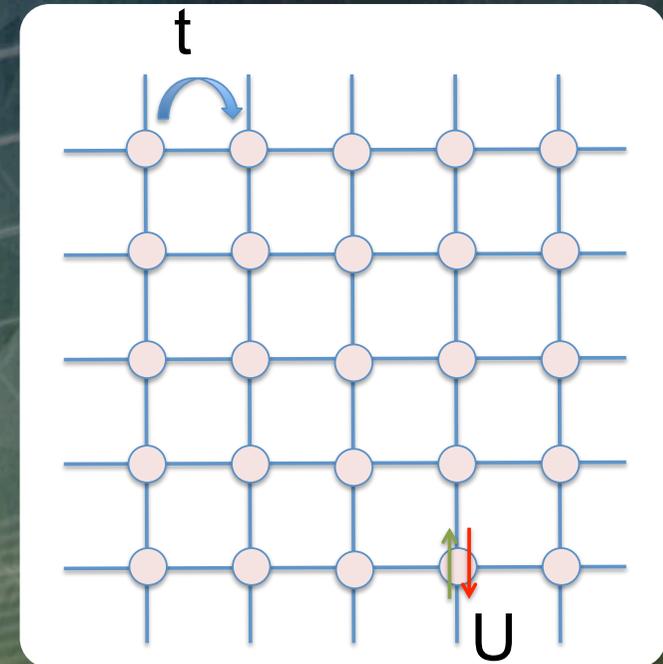
Material design for Strongly Correlated Compounds

$$H_{Hubbard} = -t \sum_{\langle i,j \rangle, \sigma} (c_{i\sigma}^\dagger c_{j\sigma} + h.c.) + U \sum_i n_{i\uparrow} n_{i\downarrow}$$

$U \ll 1$: paramagnetic Metal

$U \gg 1$: Mott insulator

- Metal to insulator transition
- Ordered phases (magnetism, superconductivity)
- Gives a good starting point to describe strongly correlated systems
- Numerical techniques available (Quantum Monte Carlo, variational Monte Carlo ...)
- Problem : not ab-initio



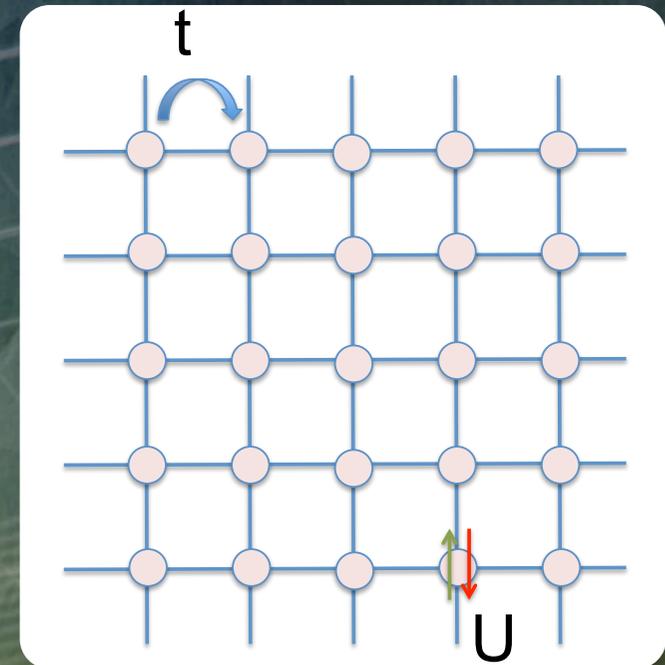
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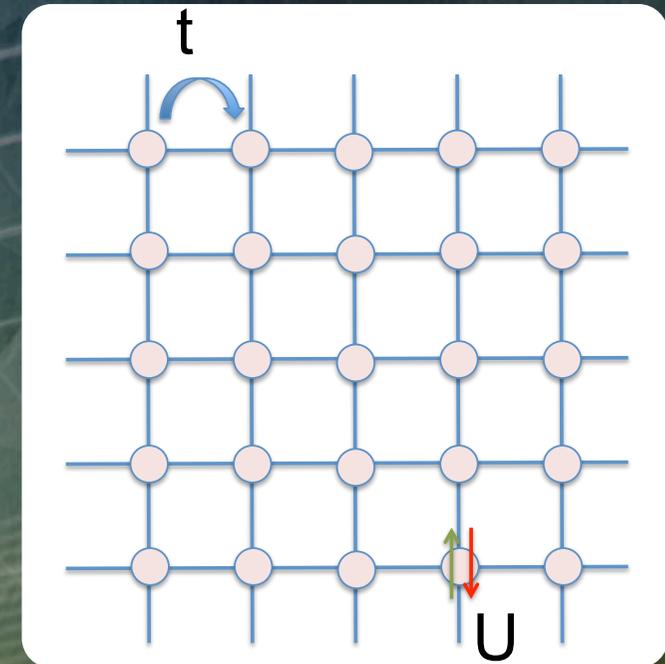


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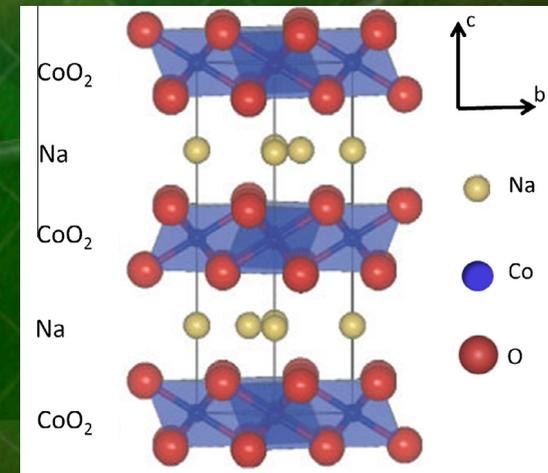
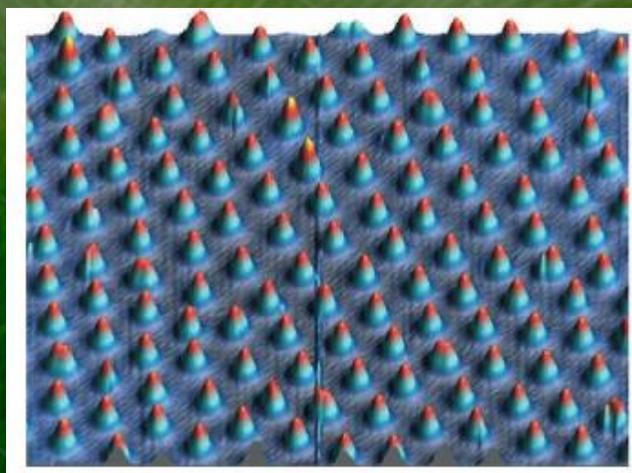
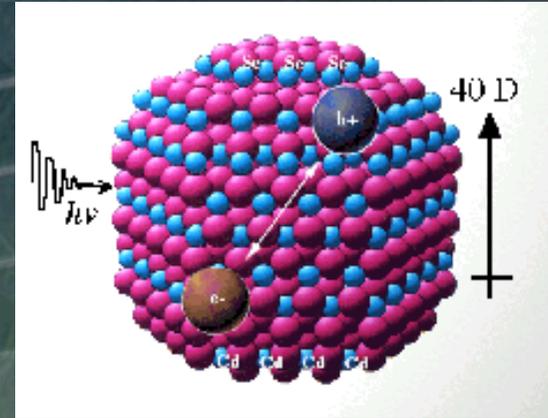
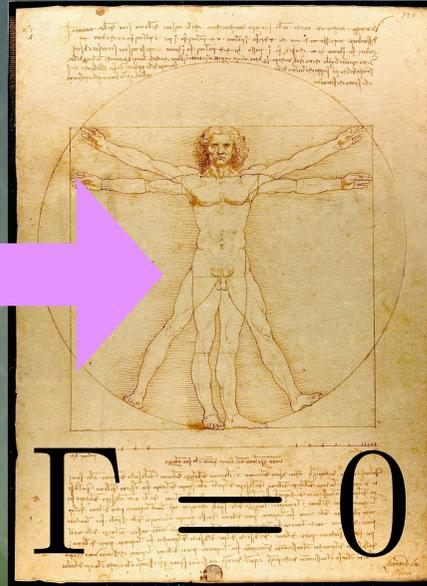
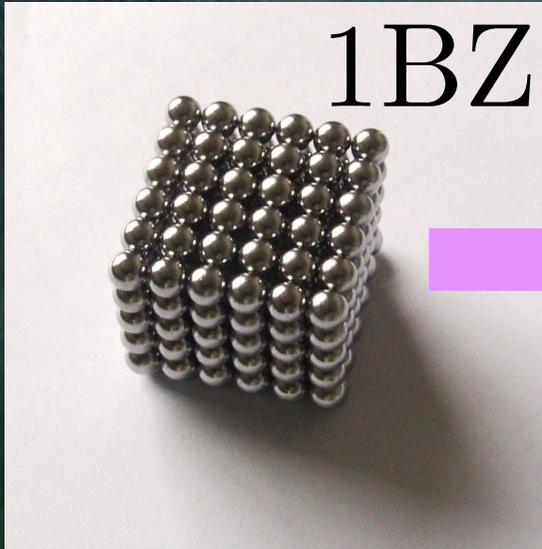
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□ Break-through:

1) bridging strong correlations approaches with first-principle calculations in a consistent way

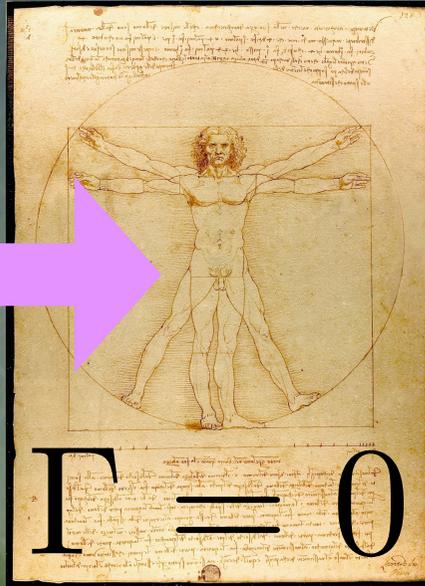
2) ***By using stable techniques with moderate computational costs*** : GW & DMFT

Promising directions : Non periodic systems

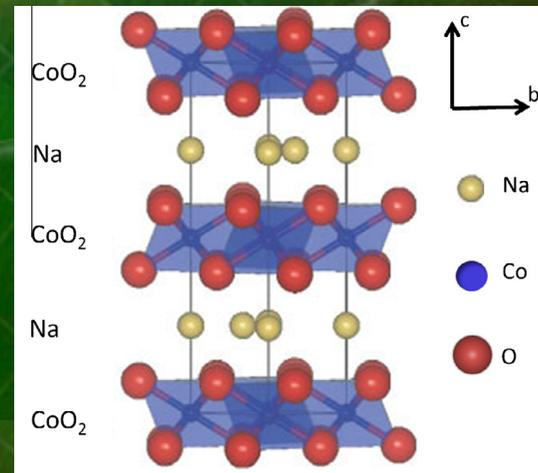
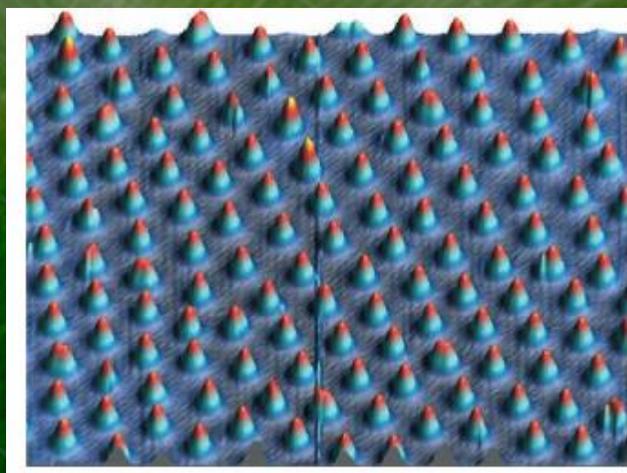


Promising directions : Non periodic systems

1BZ

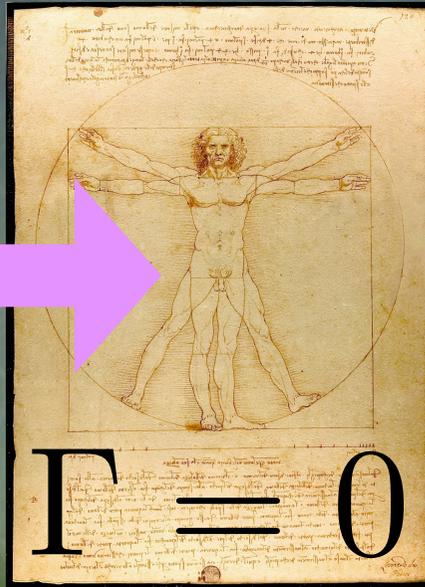


Nanocrystals
light harvesting

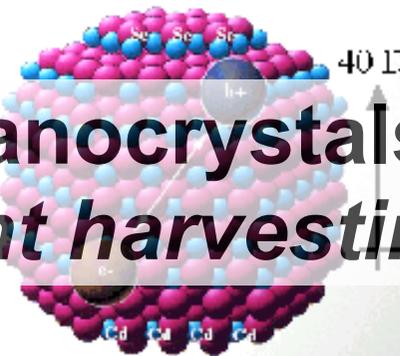


Promising directions : Non periodic systems

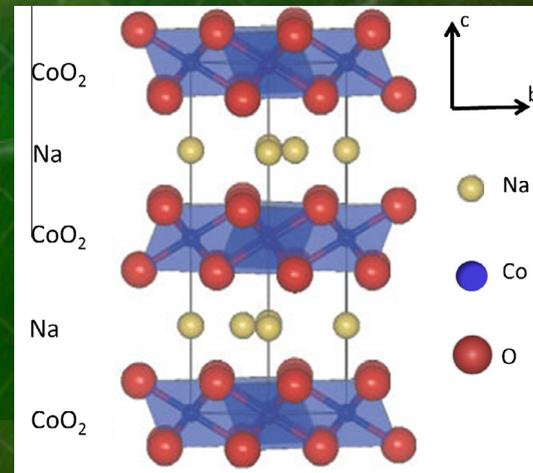
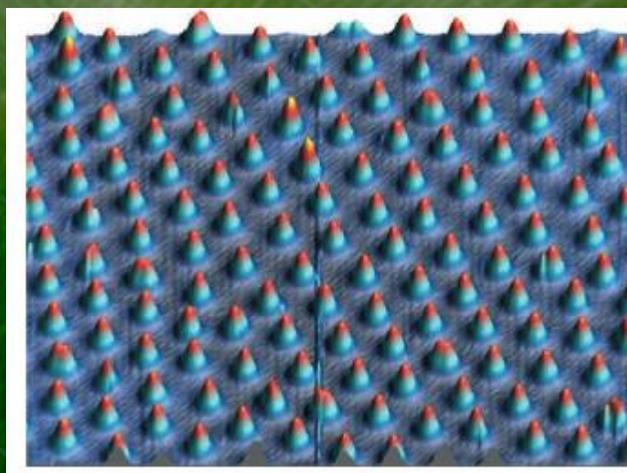
1BZ



Nanocrystals
light harvesting

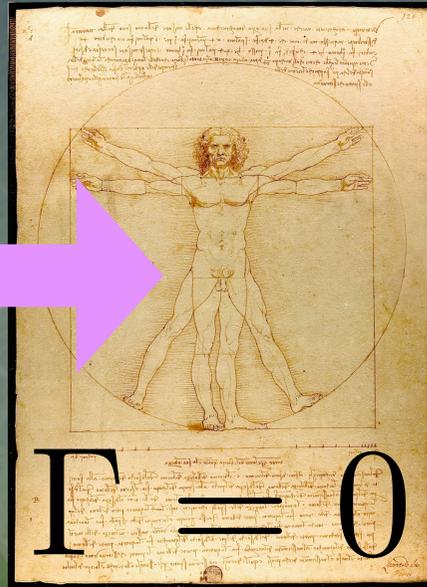


iron oxide NP
*targeted
drug release*

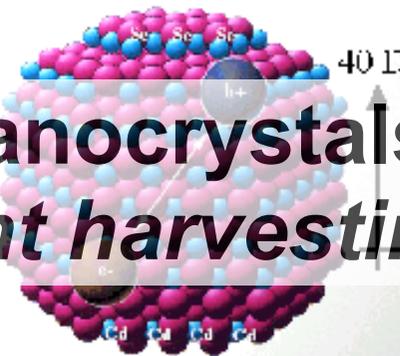


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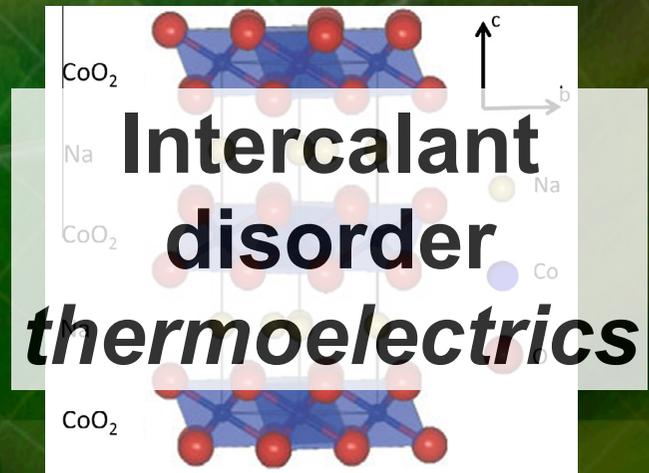
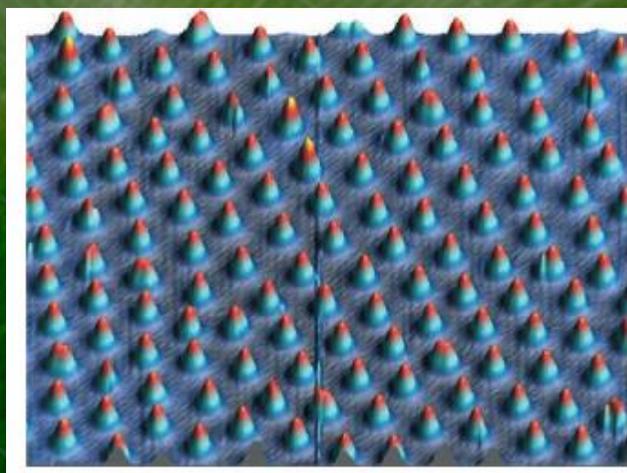
1BZ



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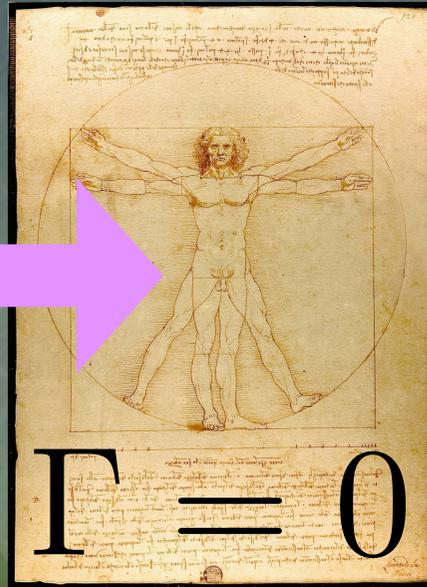


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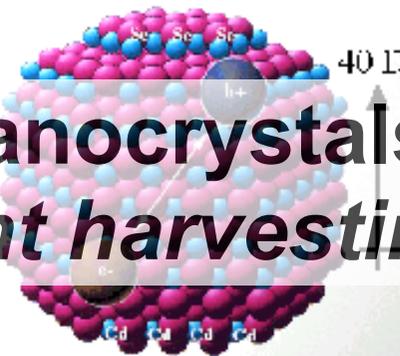


Promising directions : Non periodic systems

1BZ



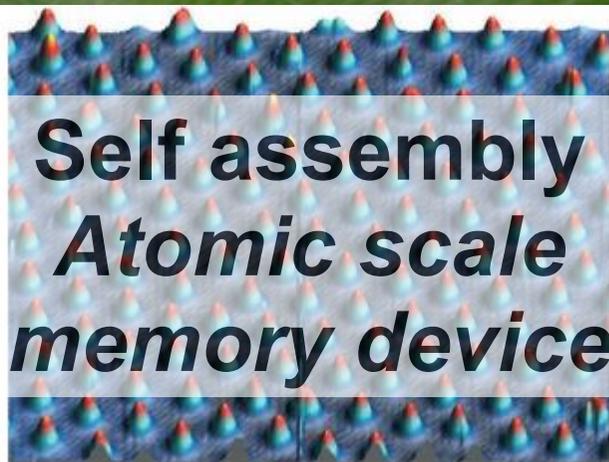
Nanocrystals
light harvesting



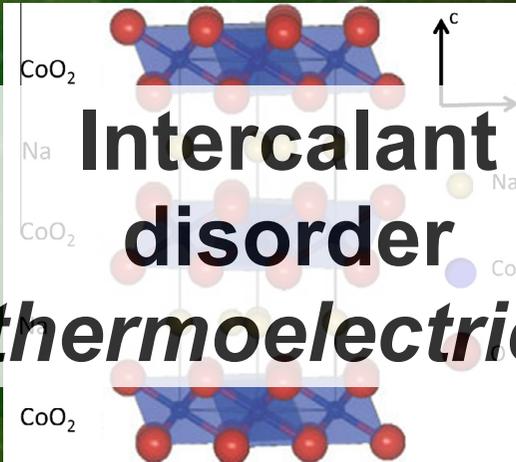
iron oxide NP
*targeted
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Self assembly
*Atomic scale
memory device*

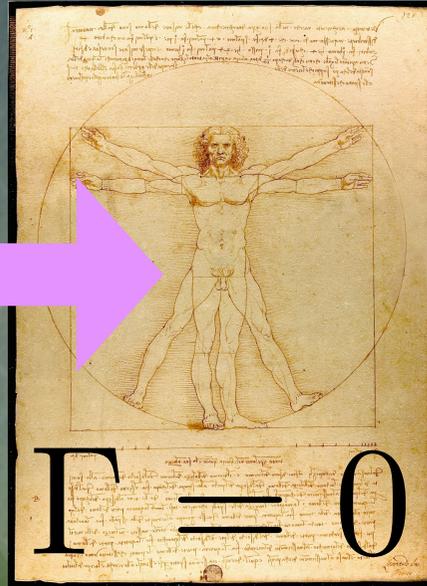


Intercalant
disorder
thermoelectrics



Promising directions : Non periodic systems

1BZ



Nanocrystals
light harvesting



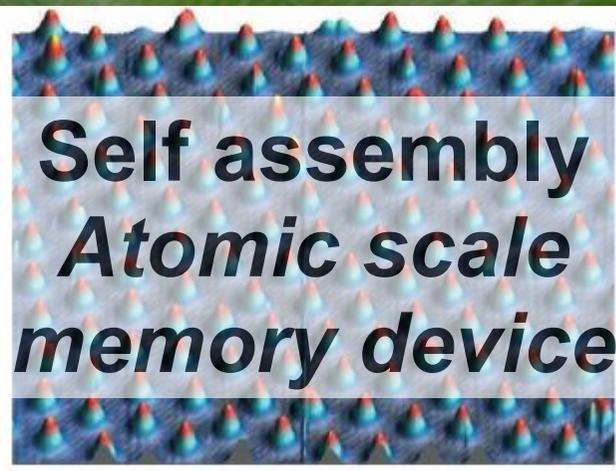
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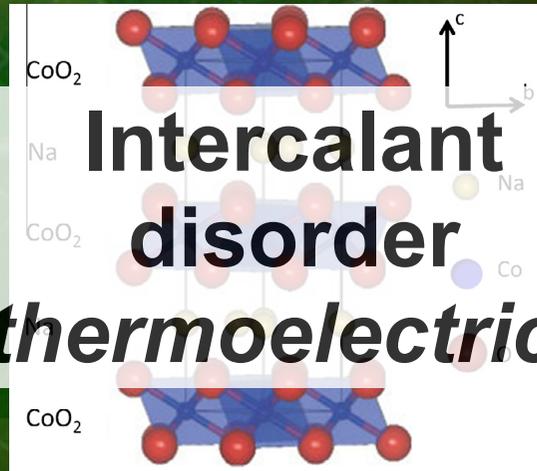
**Metallo
porphyrin
systems
drug design**



**Self assembly
Atomic scale
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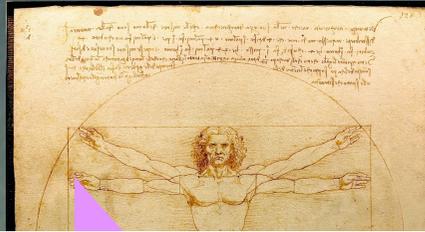


**Intercalant
disorder
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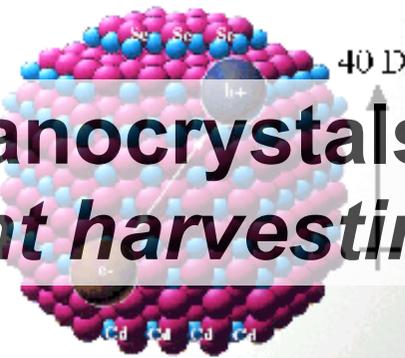


Promising directions : Non periodic systems

1BZ

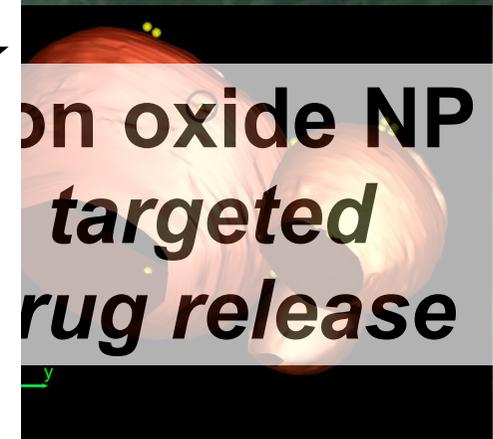


Nanocrystals
light harvesting



*~2K-20K
atoms*

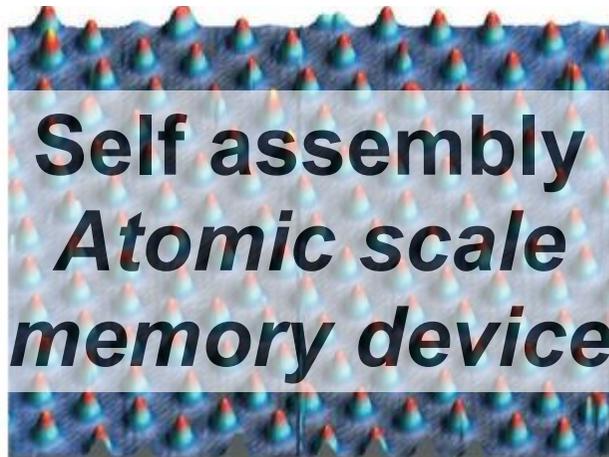
Iron oxide NP
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drug release*



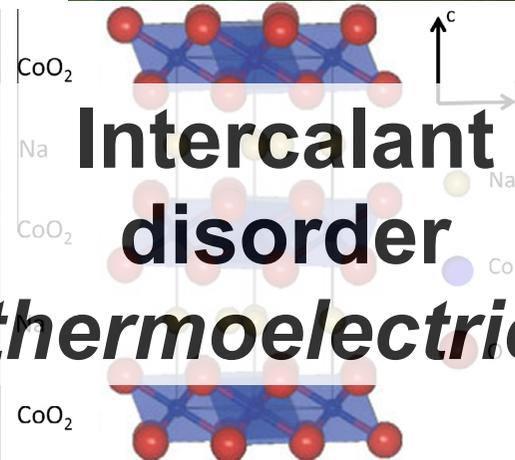
Metallo
porphyrin
systems
drug design



Self assembly
*Atomic scale
memory device*



Intercalant
disorder
thermoelectrics



Dynamical mean-field theory

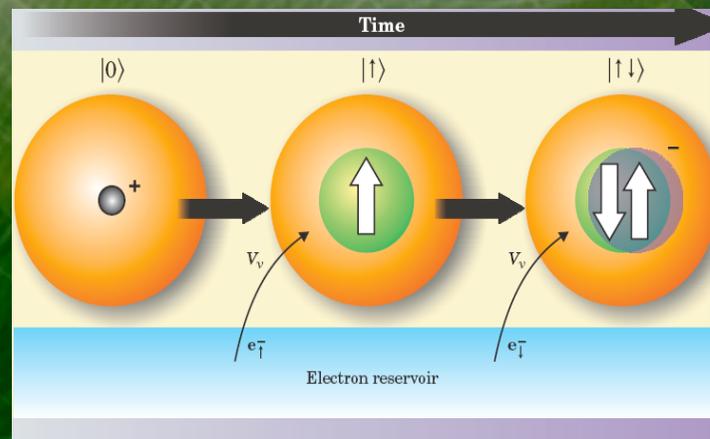
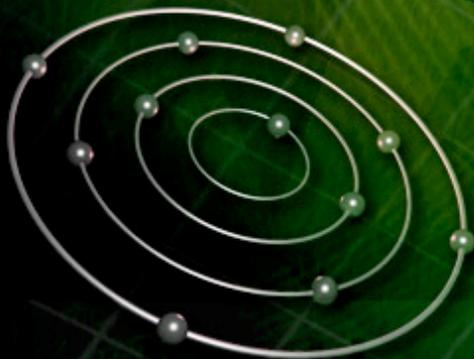
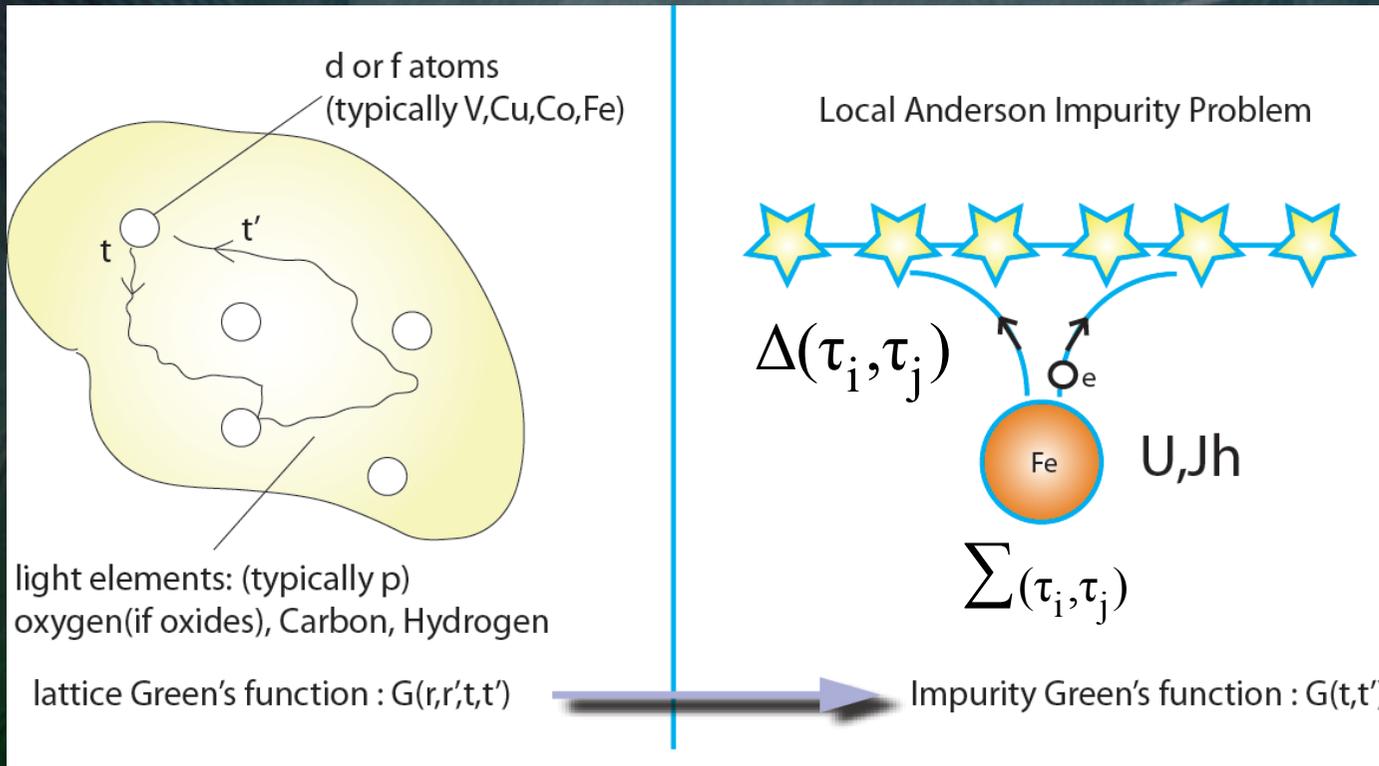
Approximation

The self energy is local in space

$$\Sigma(x,y) = \Sigma(x)$$

Projectors

connect the Kohn-Sham orbitals to the local set of atomic orbitals



Quantum Chemistry Approaches

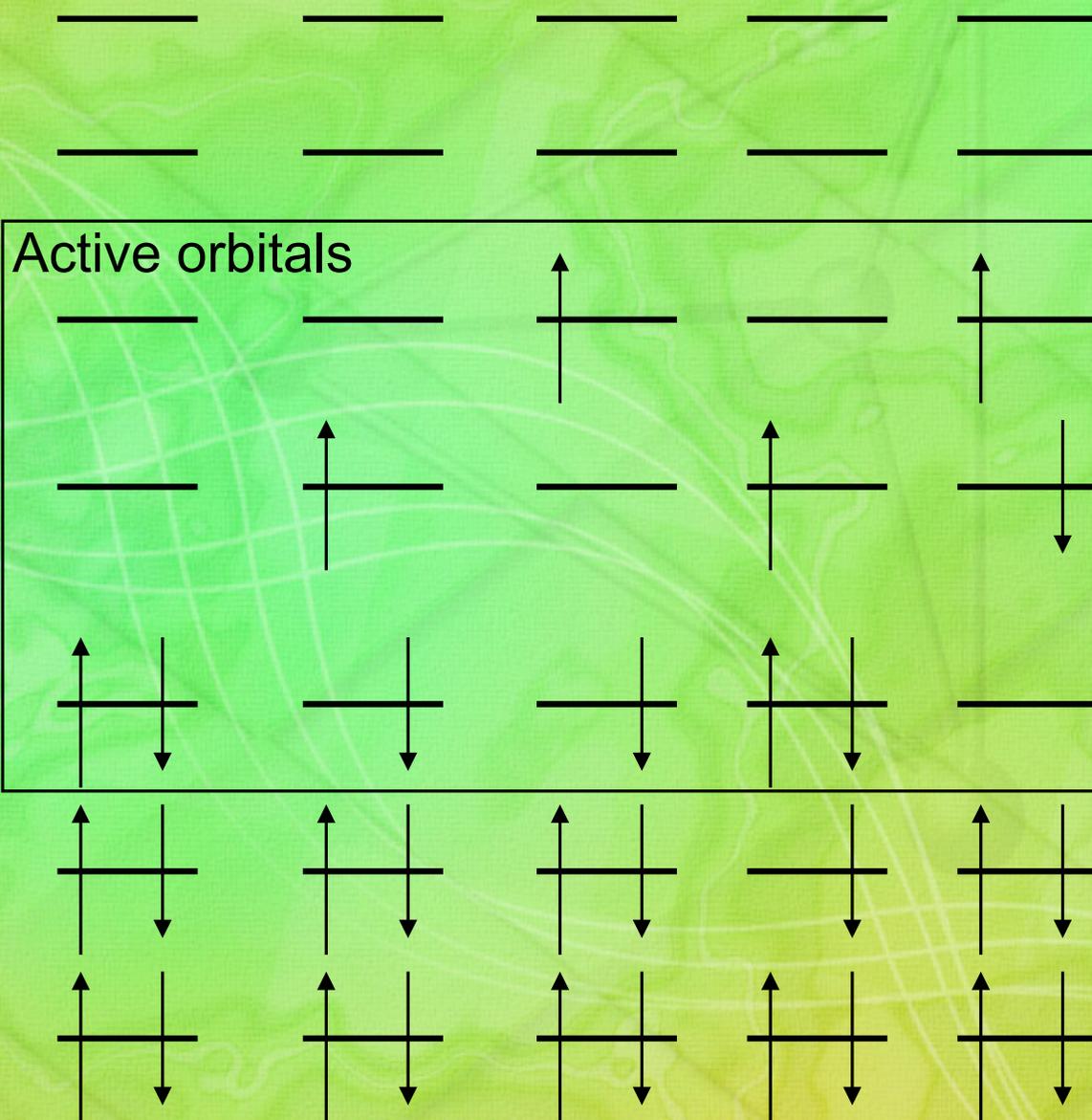
- Partition of orbitals:

- Core orbitals (filed, frozen)

- active orbitals (occupied)

- valence orbitals (unoccupied)

- virtual orbitals (unoccupied, frozen)

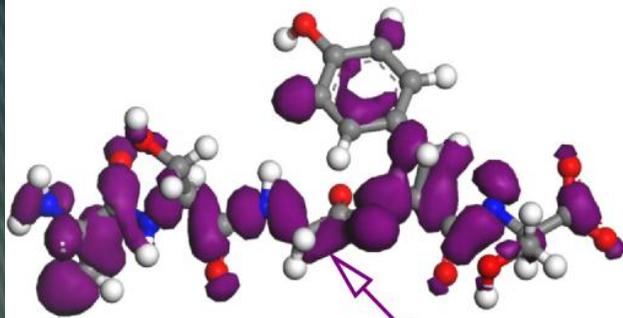


Challenges

- **Cons : Large unit-cells - DFT scales with N^3**
 - **But : Modern architectures are faster than a decade ago**
- **Cons : Real space implementation - more difficult to access bulk properties**
 - **But : natural implementation of DMFT**
- **Cons : Requires a local basis, which however can also encode the periodicity of the problem**
 - **But : Wannier functions are ideal candidates**
- **Cons : Most DFT tools are optimized for the band picture representation**
- **Cons : Chemists and Bio-Chemists already doing an outstanding job**
 - **But : We can bring our own view on this problem (Kohn-Sham instead of Hartree Fock, Green's functions, valence fluctuation ...)** 9

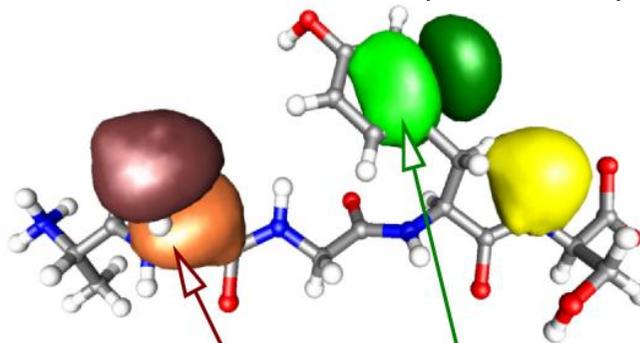
Breakthrough : Linear scaling density (ONETEP)

Molecular orbitals (MOs)



$$\rho(\mathbf{r}, \mathbf{r}') = \sum_n f_n \psi_n(\mathbf{r}) \psi_n^*(\mathbf{r}')$$

Non-orthogonal Generalised
Wannier Functions (NGWFs)



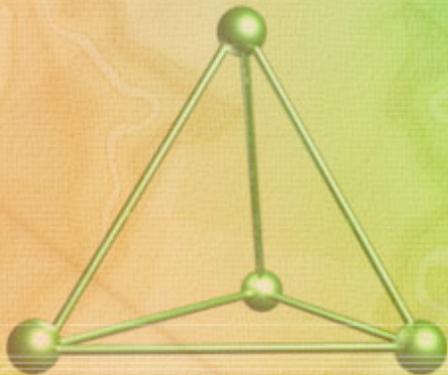
$$= \sum_{\alpha\beta} \phi_\alpha(\mathbf{r}) K^{\alpha\beta} \phi_\beta^*(\mathbf{r}')$$

- ❑ NGWFs, non-orthogonal localised basis set optimised during the DFT iterations
- ❑ Near-sighted approximation: truncation of density kernel. Efficient for solids near the localized/delocalized transition or molecules.
- ❑ Same accuracy as plane-wave methods (J.Chem. Phys 119, 8842 '03)
- ❑ Linear scaling with number of atoms, stable convergence
- ❑ Scales near linearly with the number of processors

Molecular DMFT+DFT : TOSCAM

TOIbox for Strongly Correlated Approaches to Molecules
Green's function written in the basis of a set of NGWFs :

$$G^{\alpha\beta}(i\omega_n) = ((i\omega_n + \mu)S_{\alpha\beta} - H_{\alpha\beta} - \Sigma_{\alpha\beta})^{-1}$$



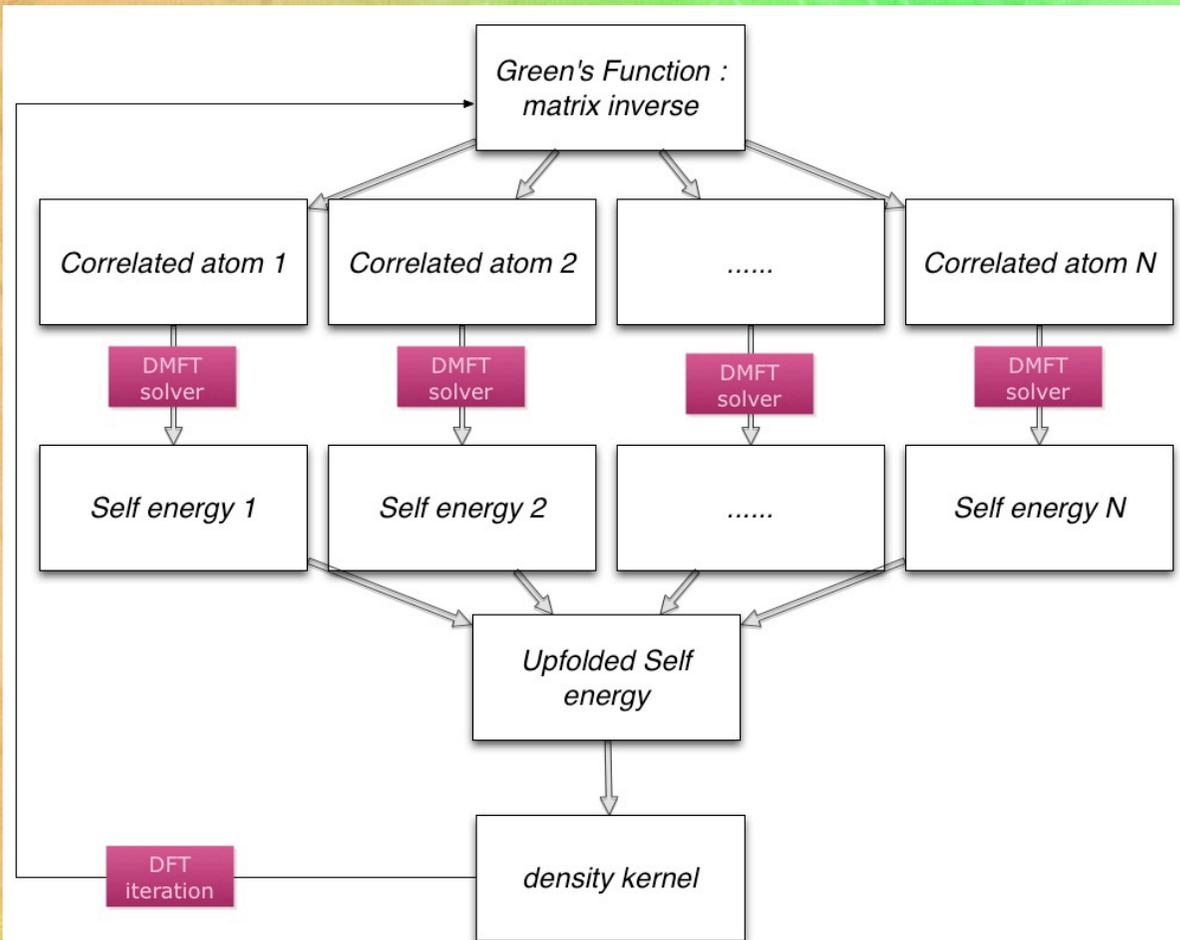
5K atoms ~1h



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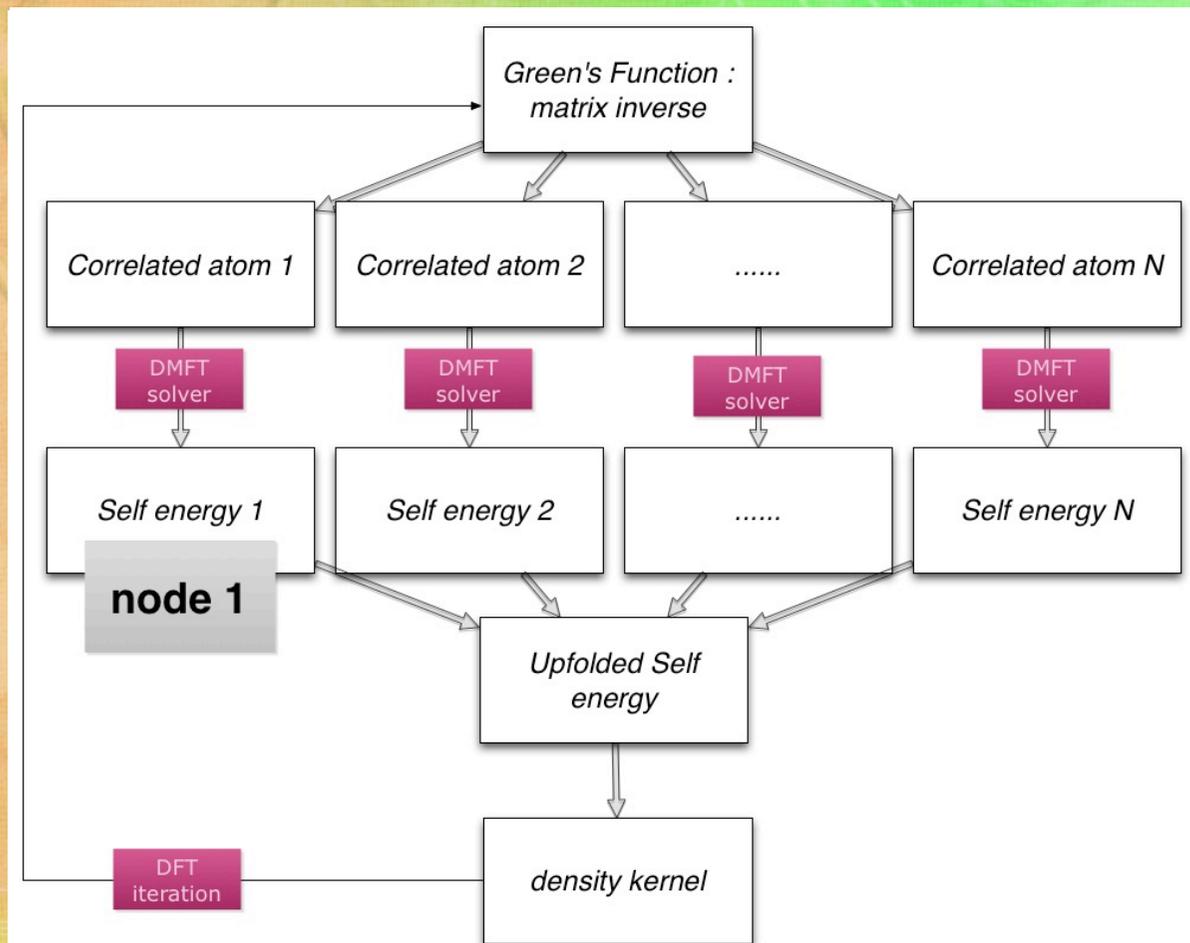
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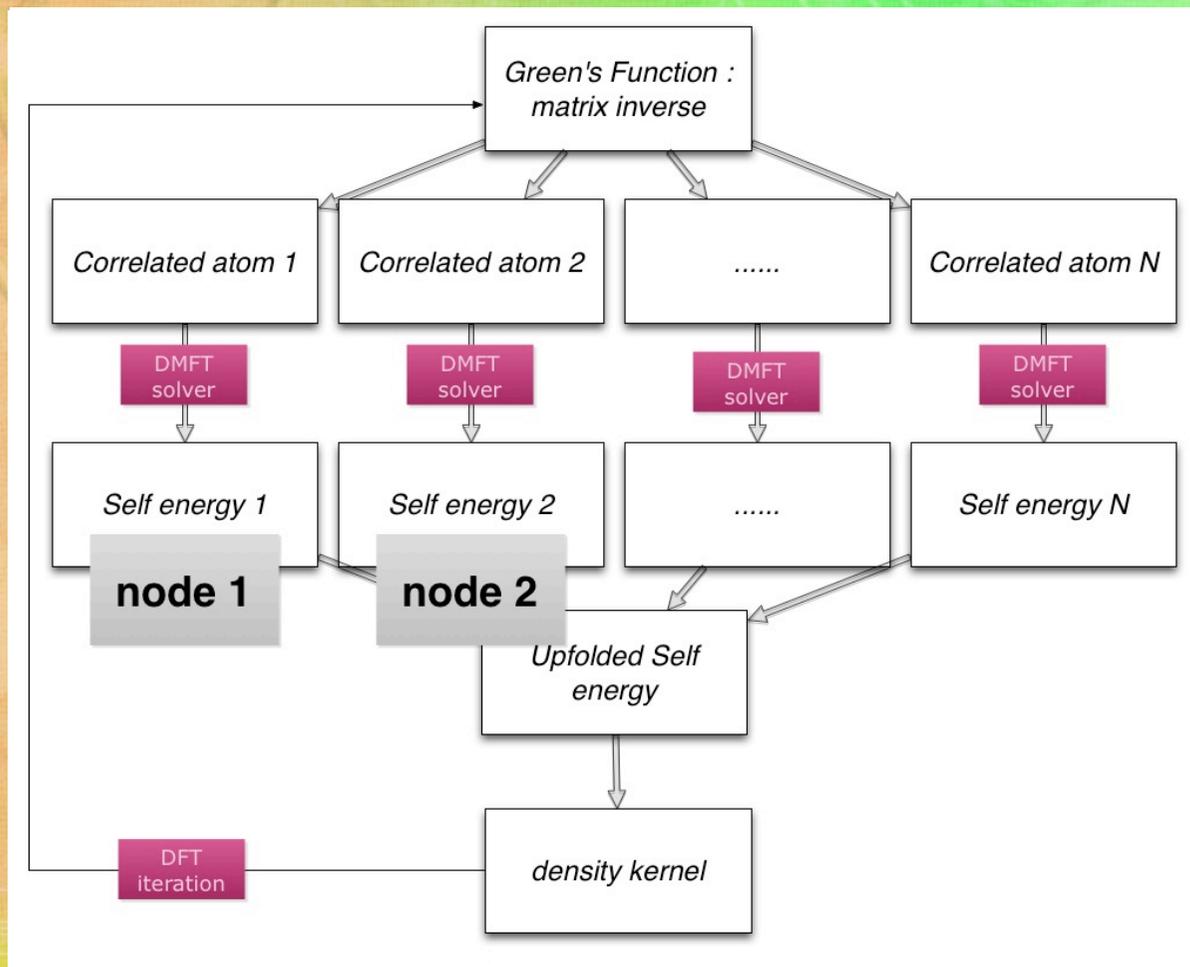
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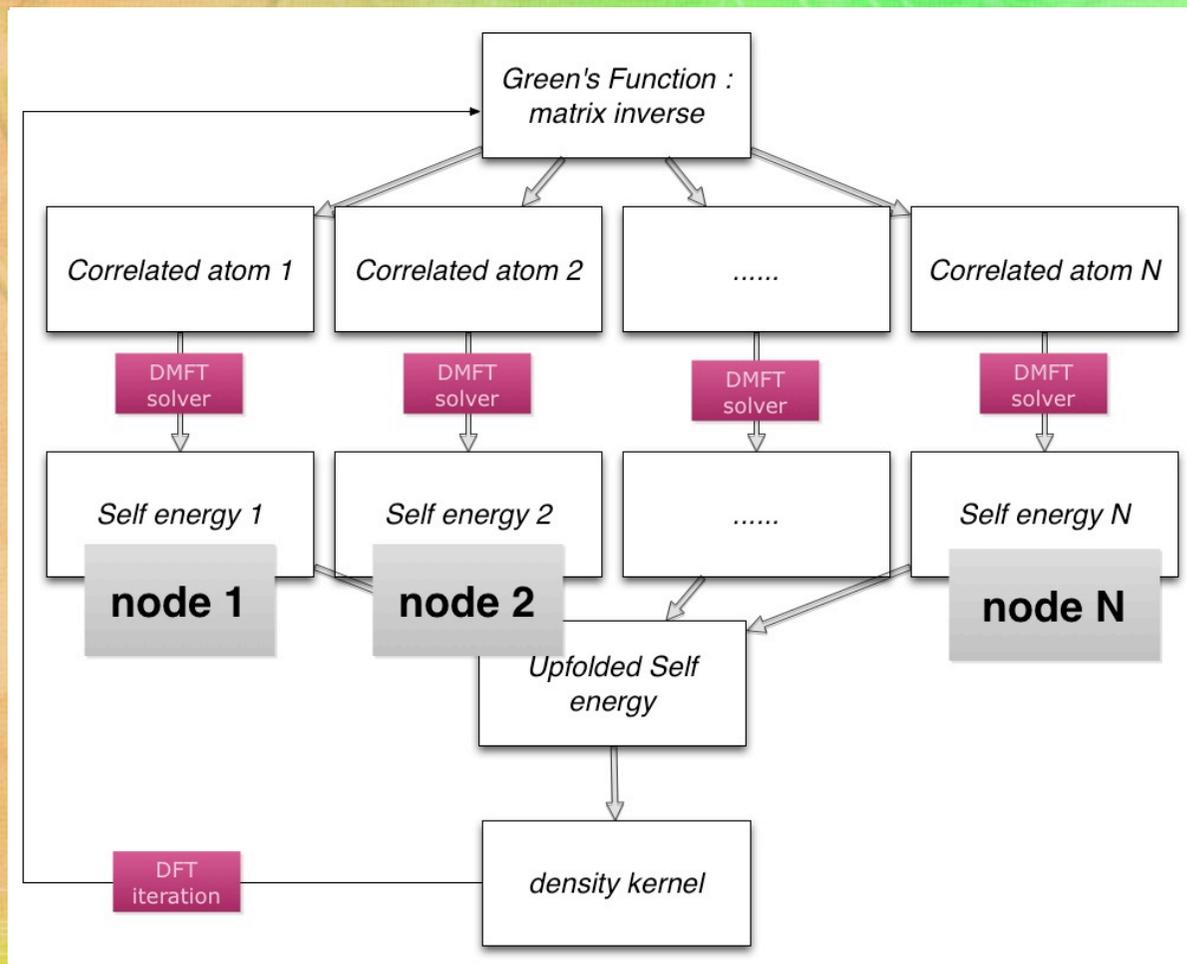
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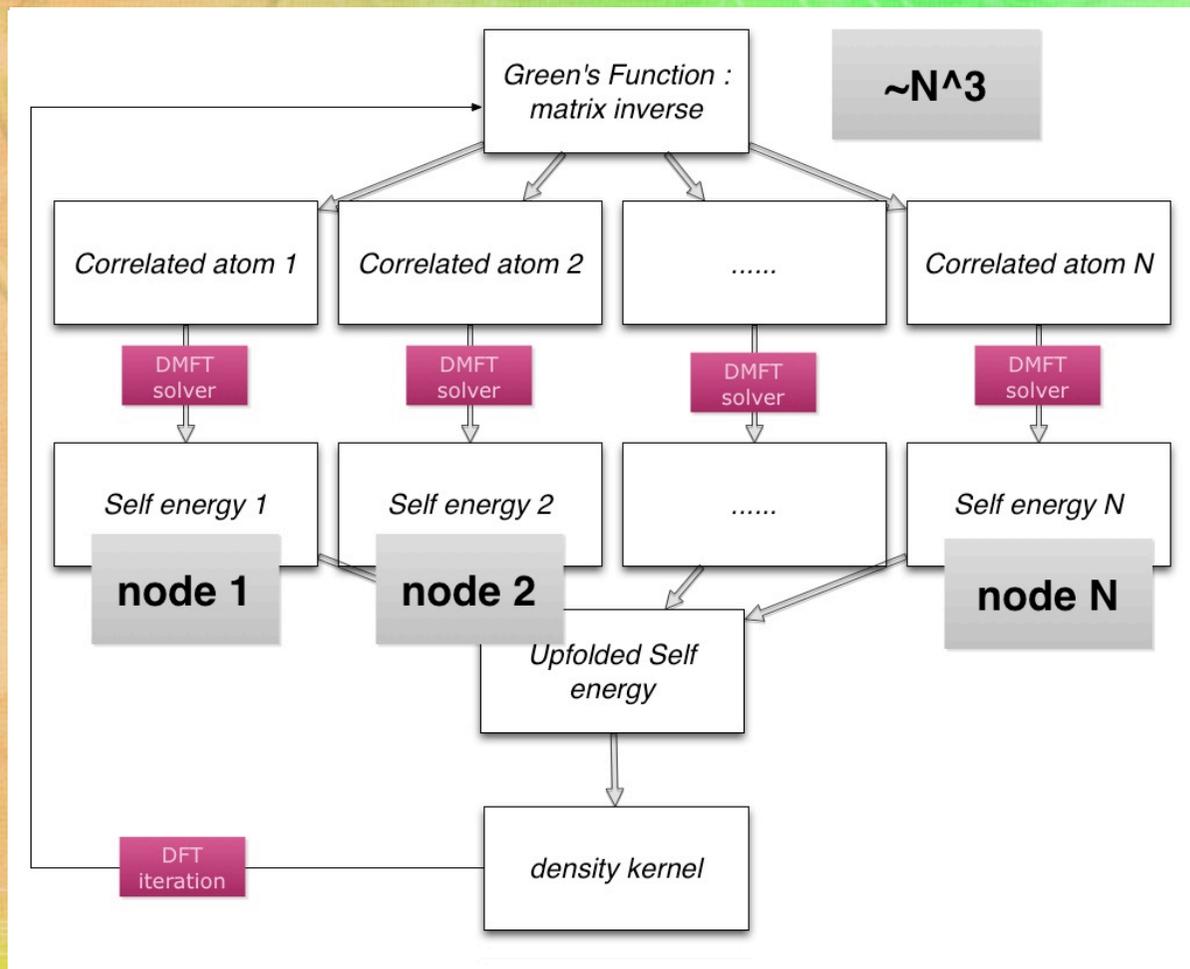
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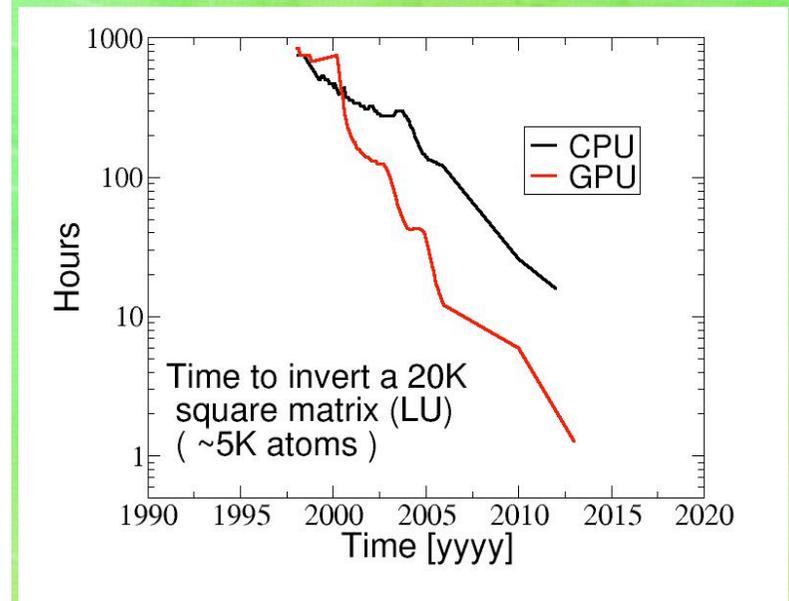
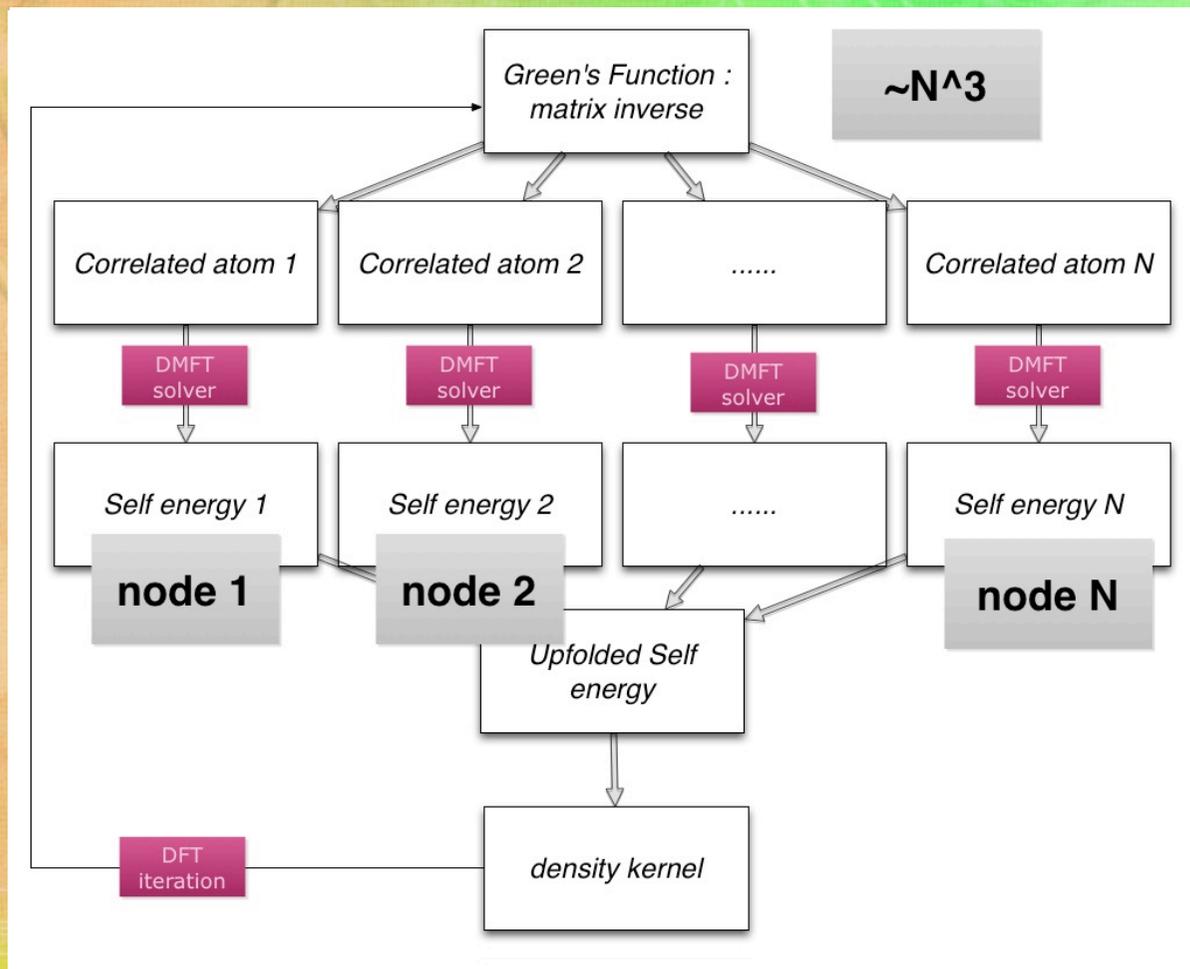
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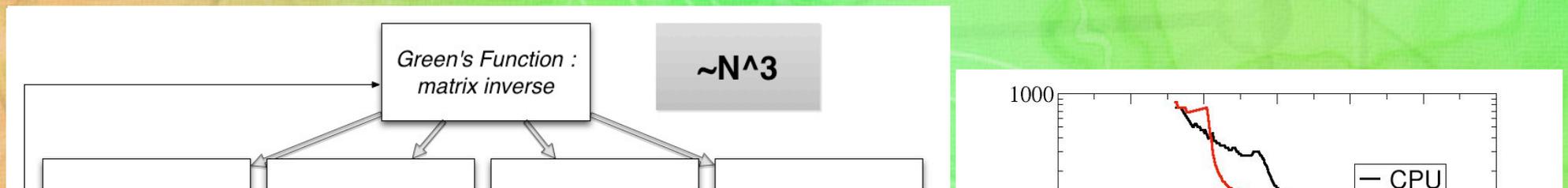
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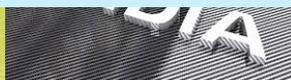
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- ❑ Difficulty :
- ❑ Some of the DFT packages have been written/finalized over a decade or more (CASTEP dev started in '99)
- ❑ What took a 1000 hours 10 years ago takes 1h today (~Moore's law)
- ❑ Most DFT codes typically are used to solve solids with ~10-50 atoms in the unit-cell

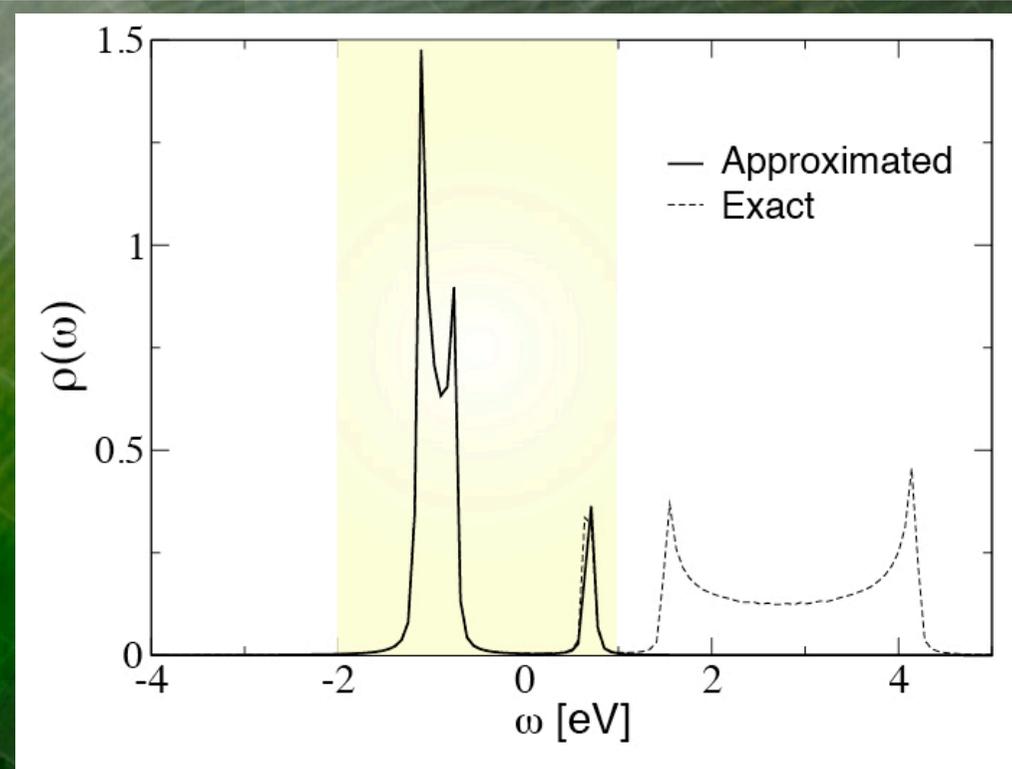
iteration

density kernel



Large scale computing

- Beyond direct inverse, impose energy window (restart Arnoldi method, Lanczos) and obtain spectral properties
- Uncorrelated Green's function
- Correlations added via updates of the Green's function ($\sim N^2$), fast if a large number of uncorrelated orbitals
- Large scale computing (Bluegene/Q)



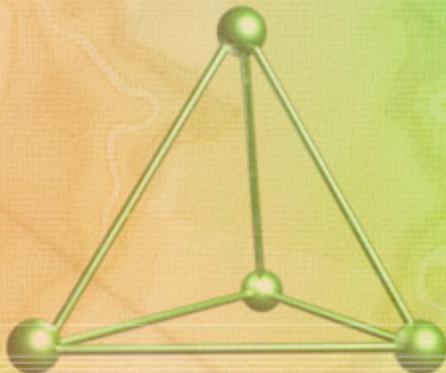
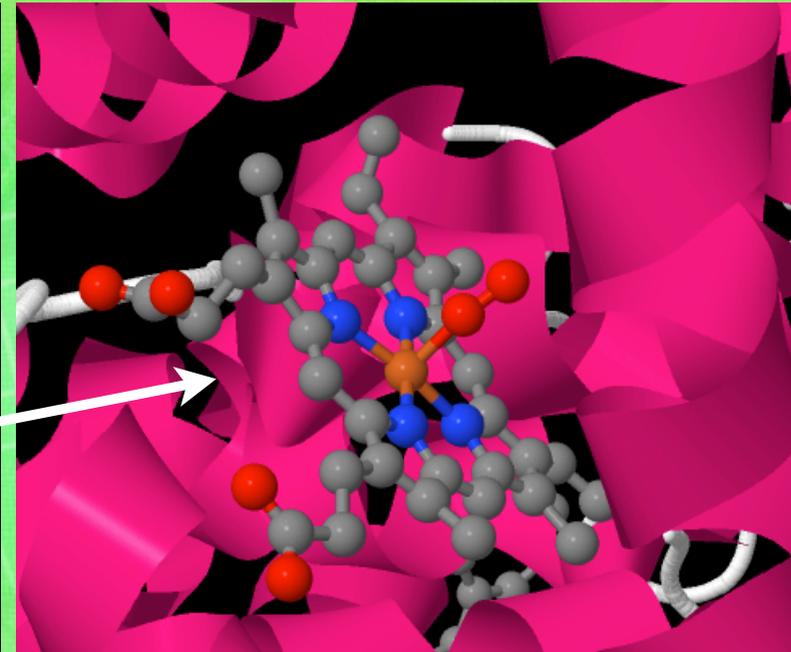
Ligand Binding : haemoglobin

Biological Molecules typically consist of large uncorrelated structures (C,H,O) surrounding a functional kernel with a correlated ion, such as iron porphyrin in haemoglobin.

Human haemoglobin



heme (kernel) binding to O₂



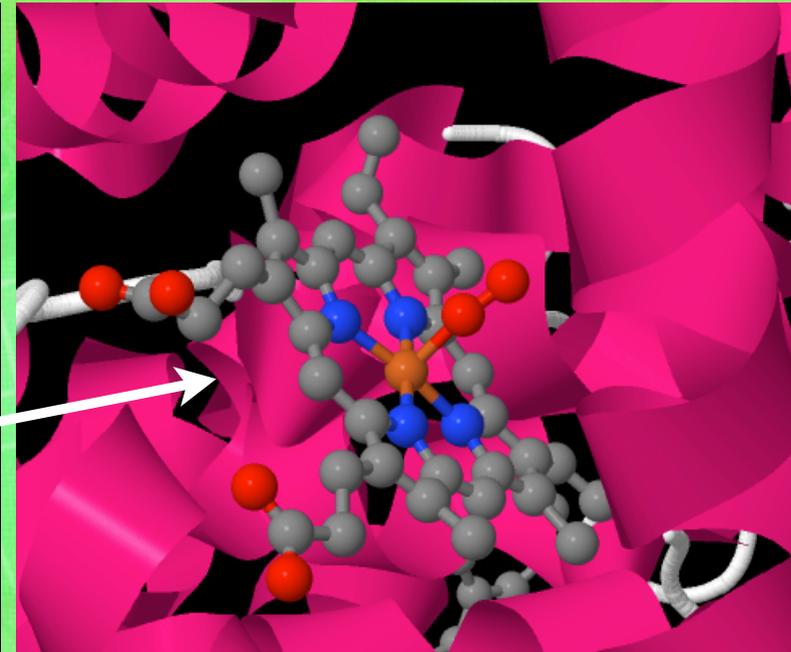
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Human haemoglobin



heme (kernel) binding to O₂



- ❑ quantum/thermal fluctuations
- ❑ multi-determinantal effects and entropy
- ❑ fluctuating magnetic moment (no spin contamination)
- ❑ ligand energetics: we need a good estimate of the orbital dependent hybridization and crystal field
- ❑ Beyond energy crossings = dynamical effects

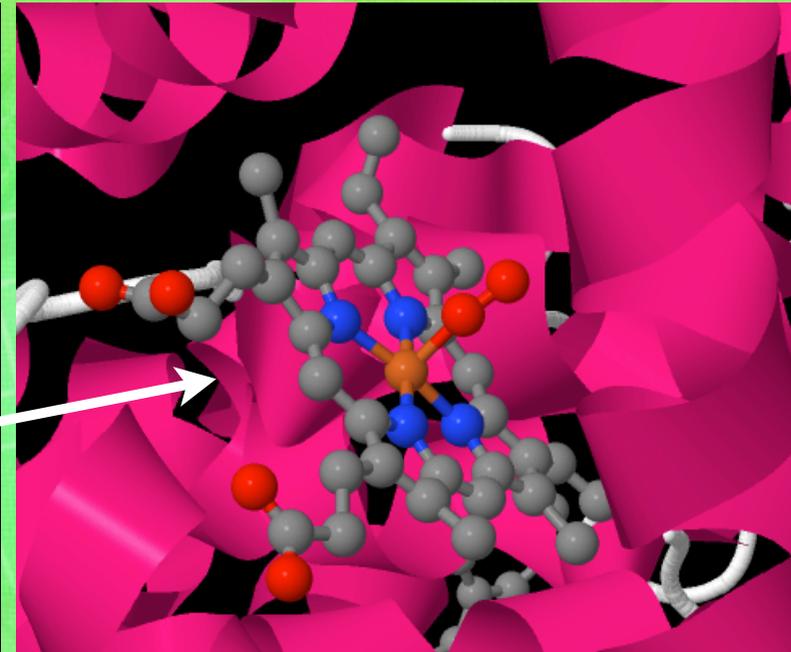
Ligand Binding : haemoglobin

Biological Molecules typically consist of large uncorrelated structures (C,H,O) surrounding a functional kernel with a correlated ion, such as iron porphyrin in haemoglobin.

Human haemoglobin

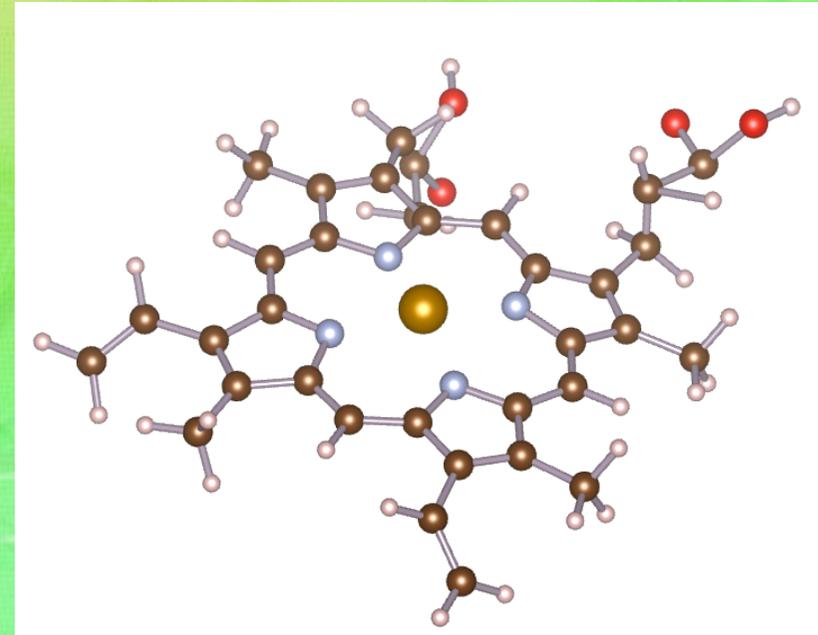
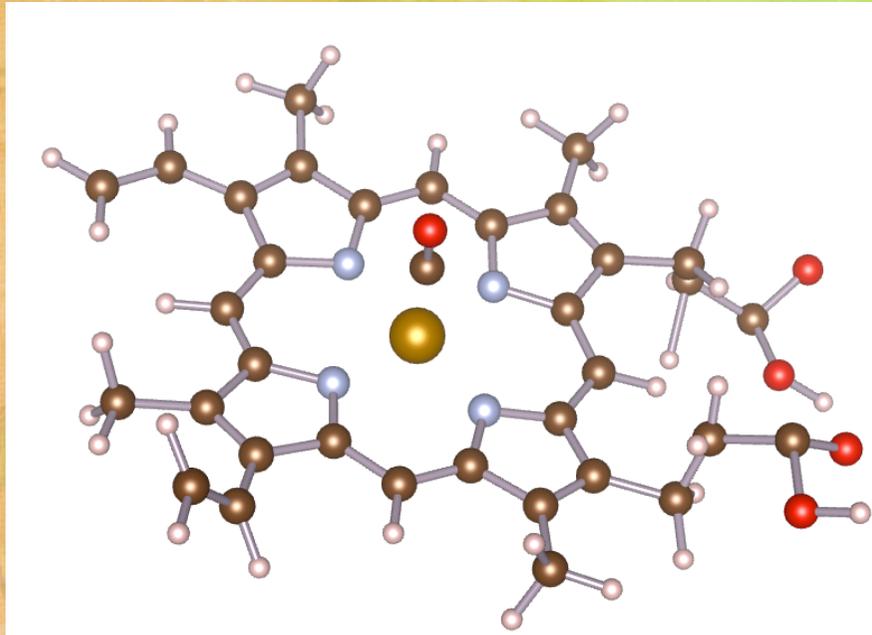


heme (kernel) binding to O₂



- quantum/thermal fluctuations
- multi-determinantal effects and entropy
- fluctuating magnetic moment (no spin contamination)
- ligand energetics: we need a good estimate of the orbital dependent hybridization and crystal field
- Beyond energy crossings = dynamical effects

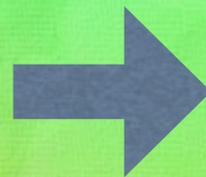
Heme : Kernel Conformation



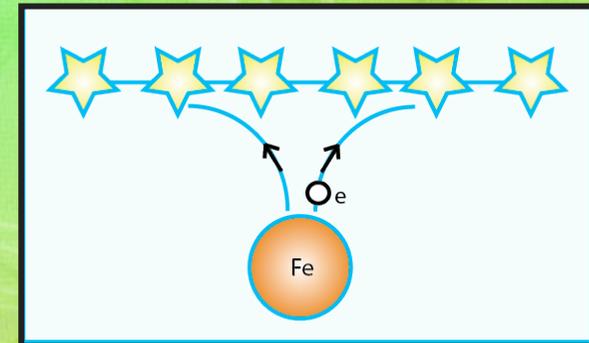
oxy-Heme (FeP(O₂)) - planar shape

desoxy-heme (FeP) - domed shape.
Fe out of the nitrogen plane by 0.35Å

Heme
(~240 orbitals)

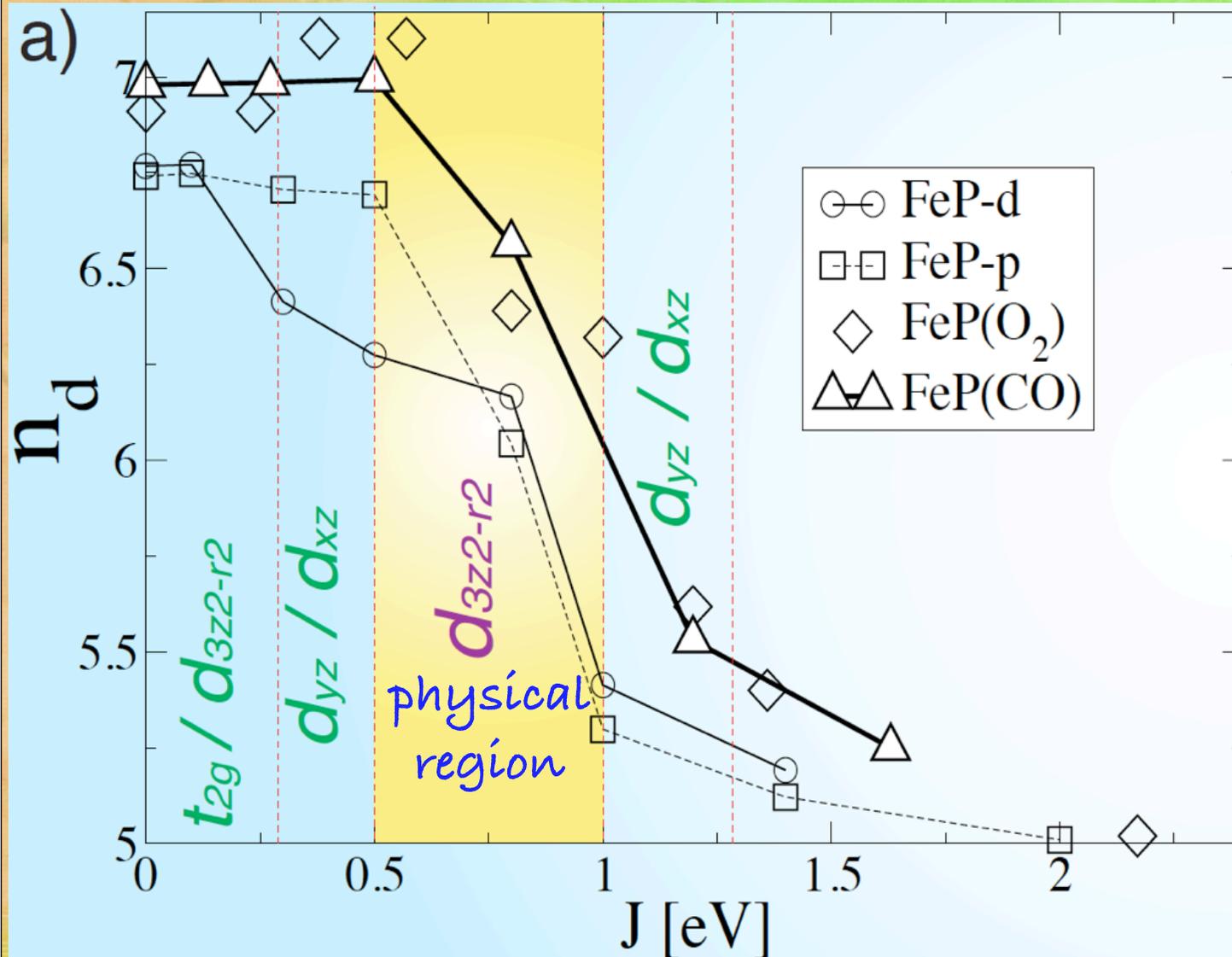


AIM
5d orbitals
+bath



What is the link between topology/Binding and
electronic states (charge/spin)?

Hund's rule J in Heme



J drives a **transition** between **low- to high-spin**

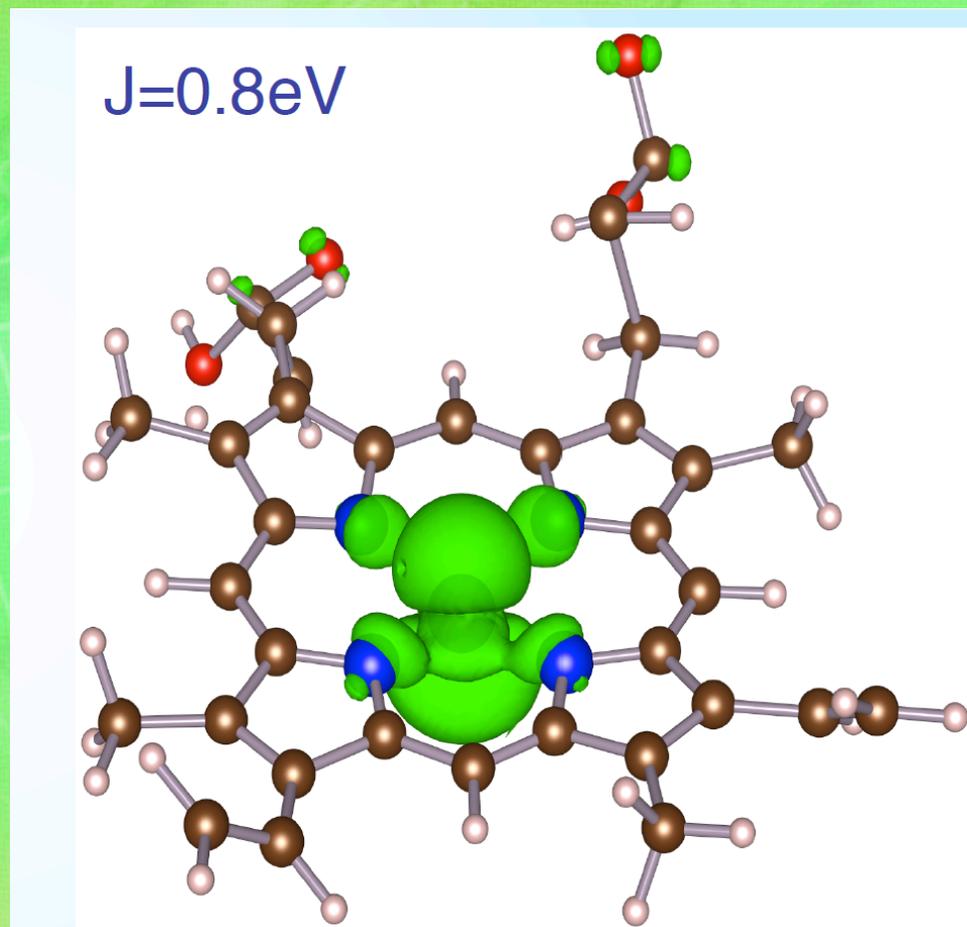
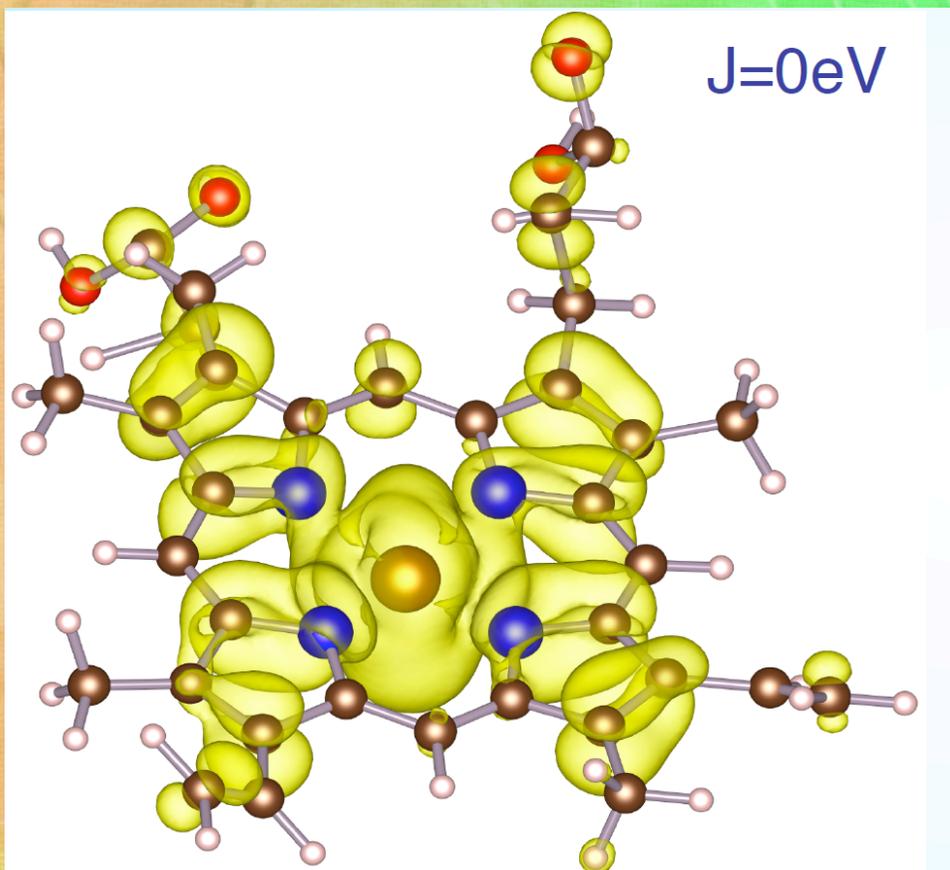
Fully polarised state has 5 electrons (d-shell)

Change of **orbital polarization** across the phase diagram

Sharp drop of the iron density at $J \sim 0.7$ eV

HOMO Orbital character in desoxy-heme

**J affects strongly
the symmetry of
the HOMO**



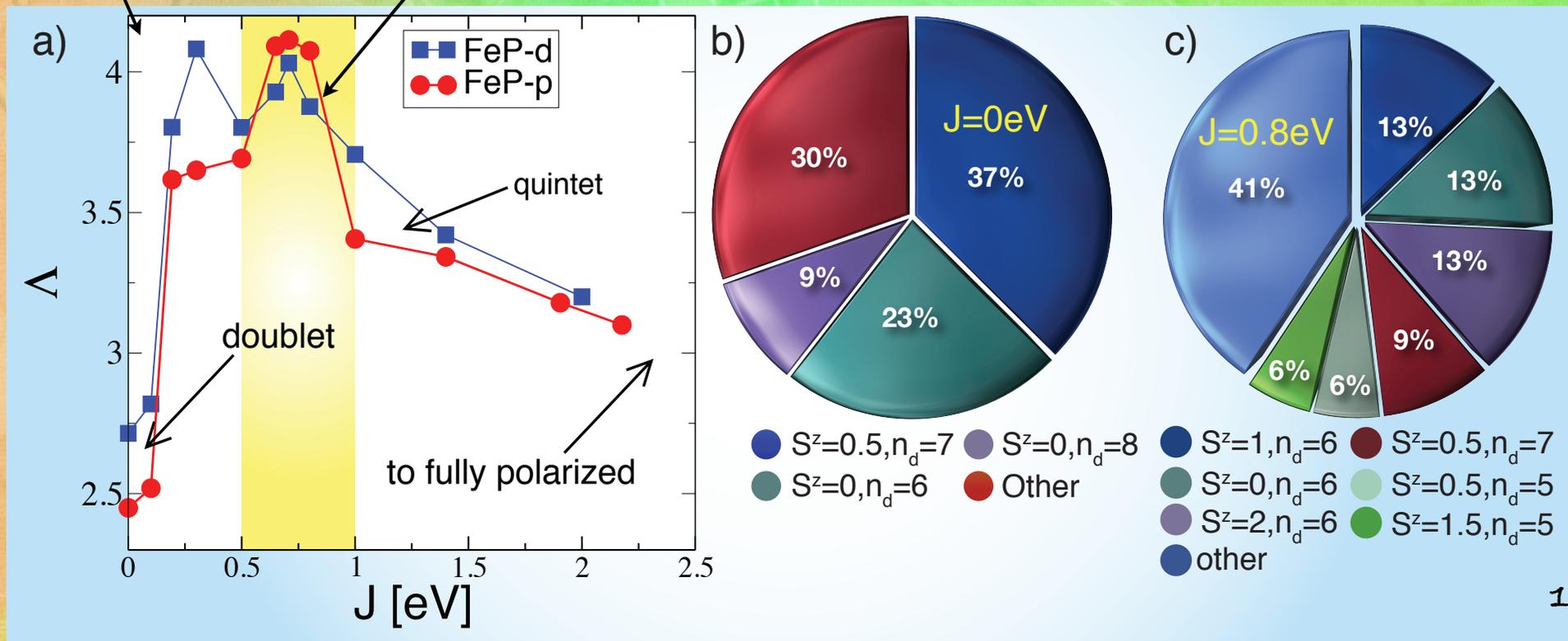
Multi-determinantal effects (unligated heme)

d-shell reduced density matrix (bath degrees of freedom are integrated out)

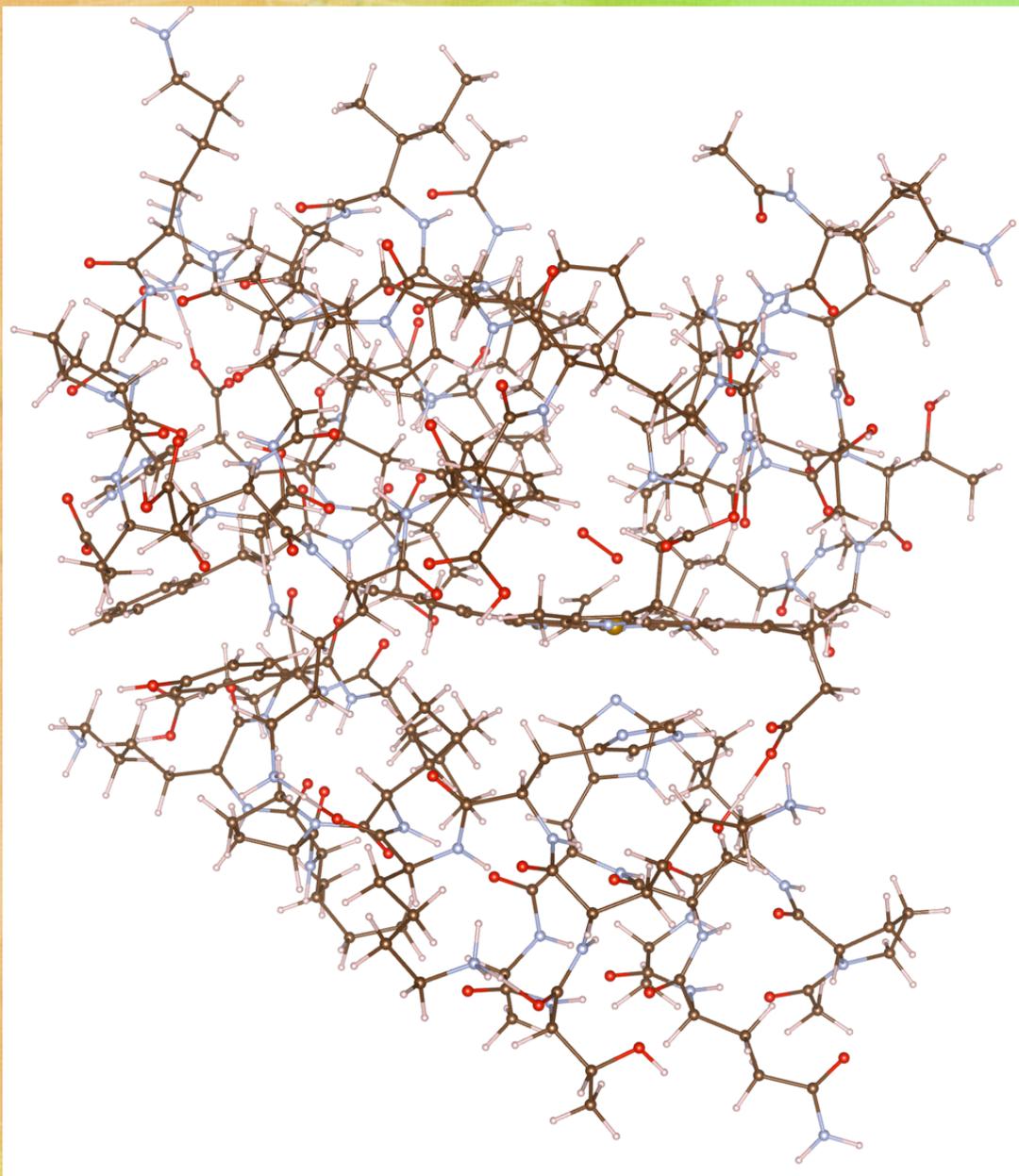
low spin Fe state,
low entropy,
classical valence

physical
region,
high entropy,
valence
fluctuation

"valence
fluctuations"



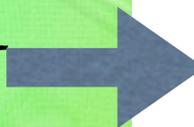
DFT+DMFT : Myoglobin



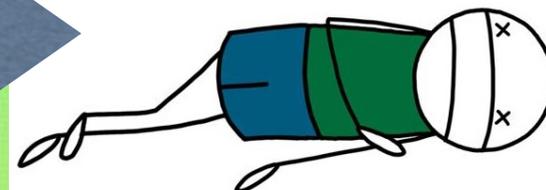
DFT calculations : binding energy to CO is 1eV greater than to O₂

Problem : CO is toxic !
(Biophys. Journ. 65, 1942 ' 93)

DFT



YOU FORGOT TO BREATHE!
I told you! Don't forget to breathe!

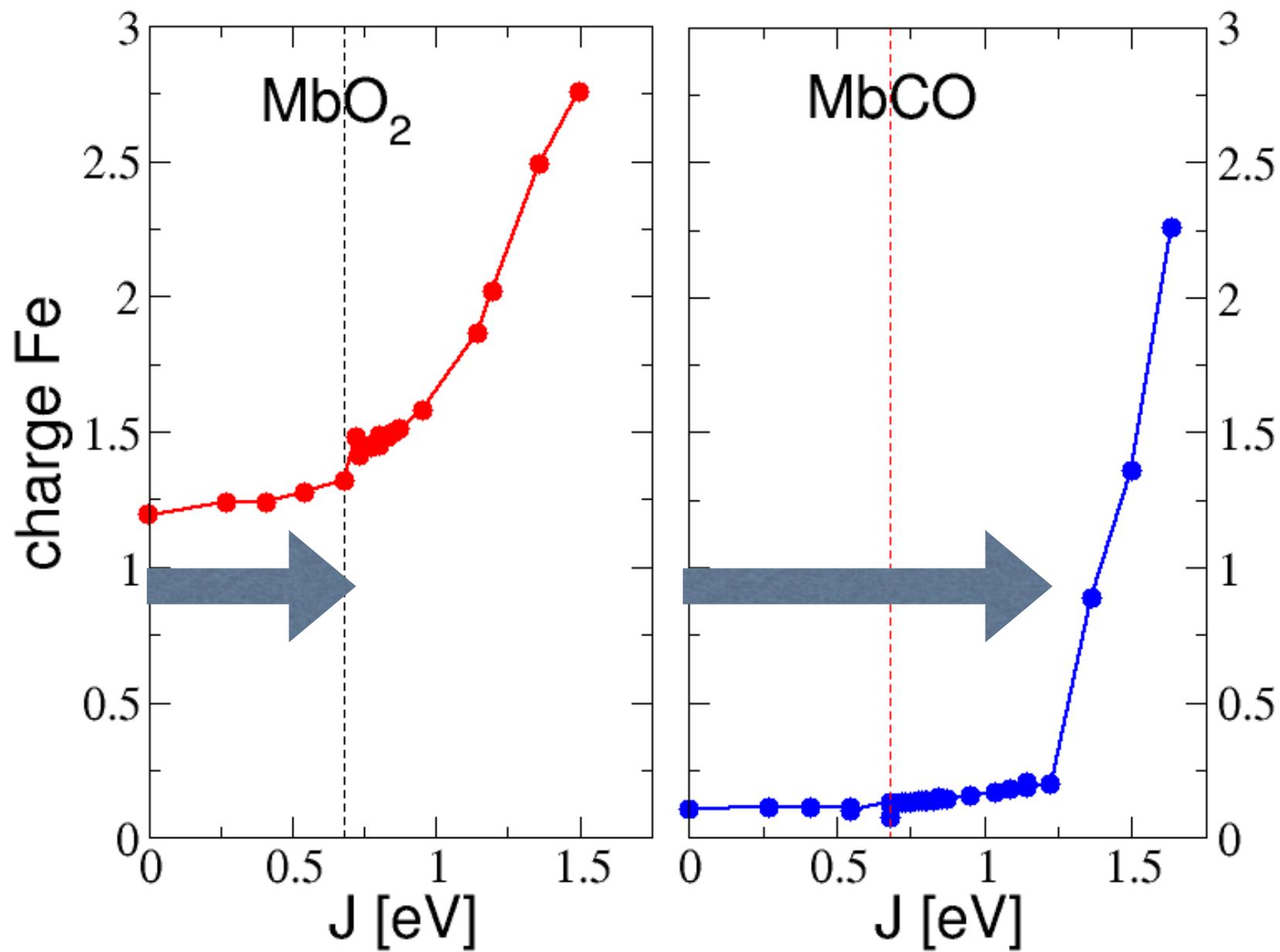


Treatment of correlations ?

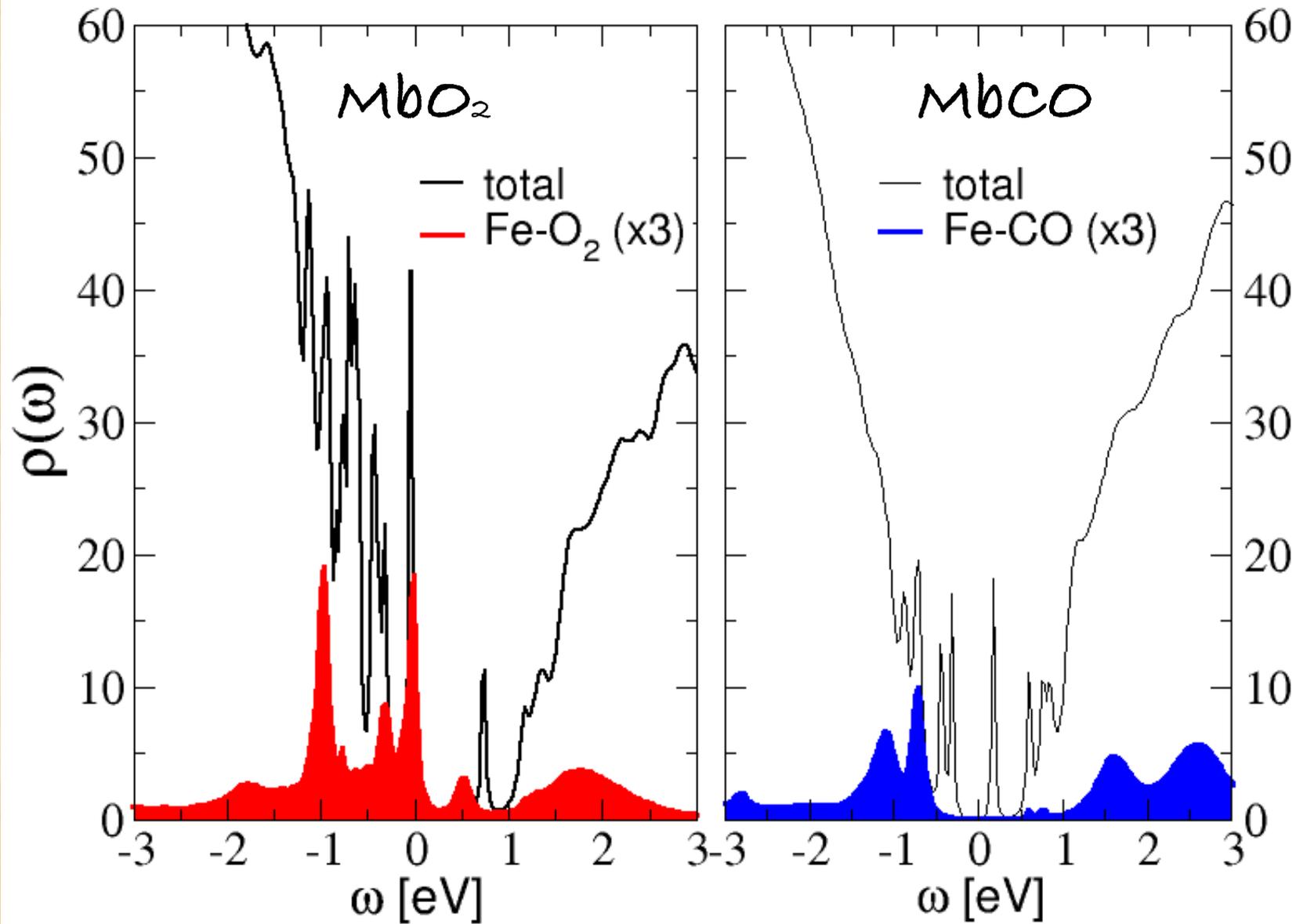
U and J

Iron atom: $J \sim 0.68\text{eV}$, $U \sim 4\text{eV}$

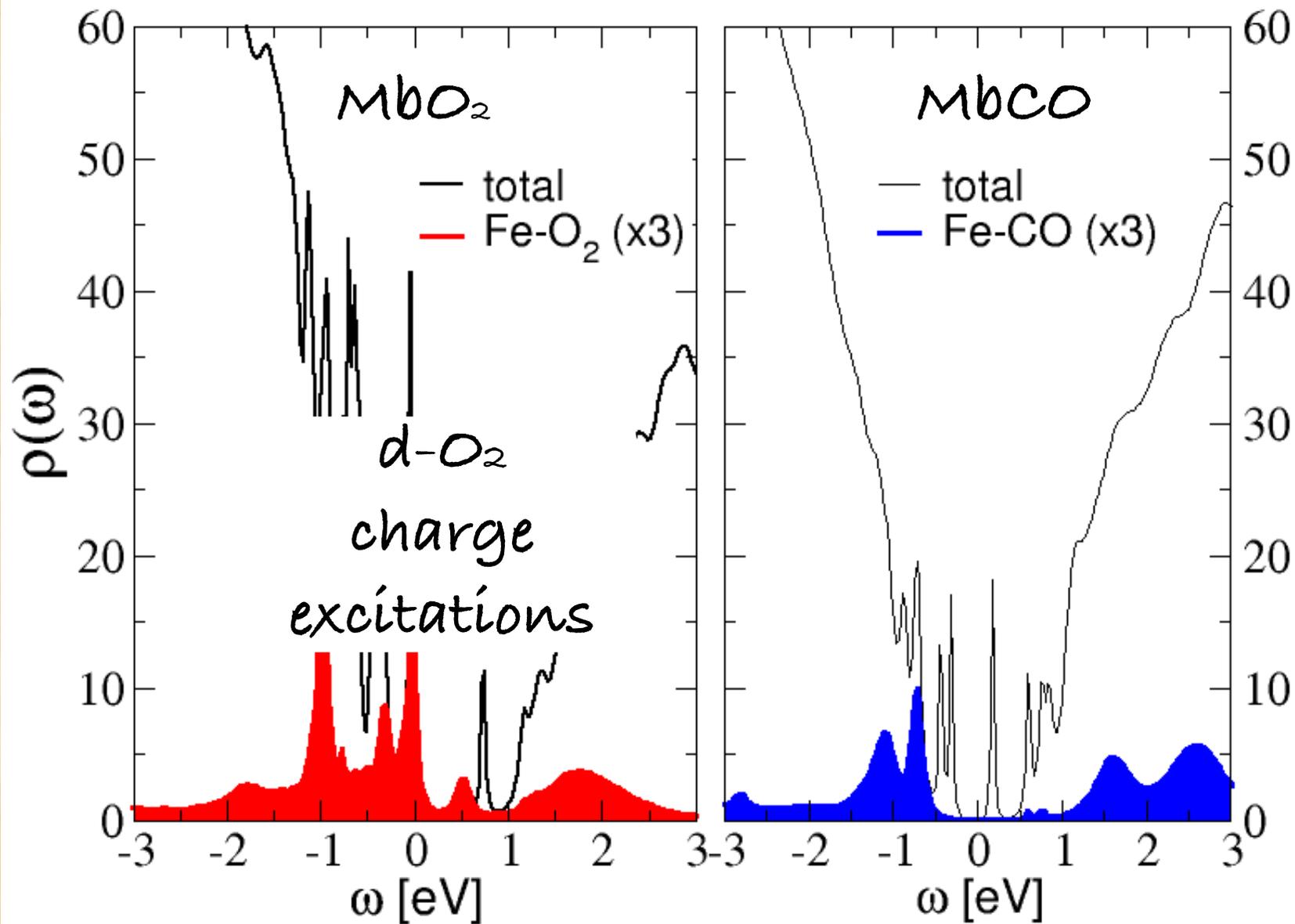
Fe charge



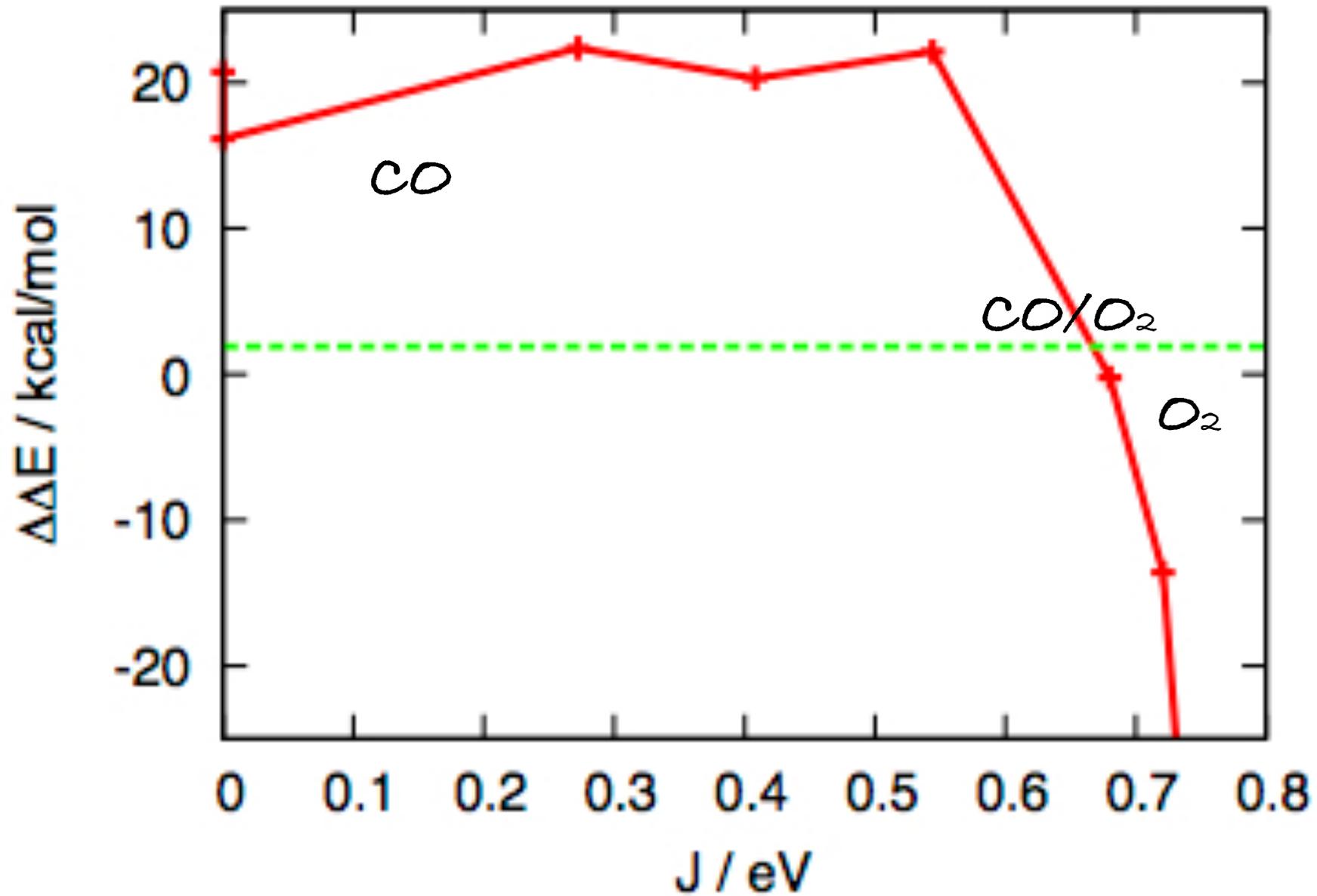
Binding discrimination



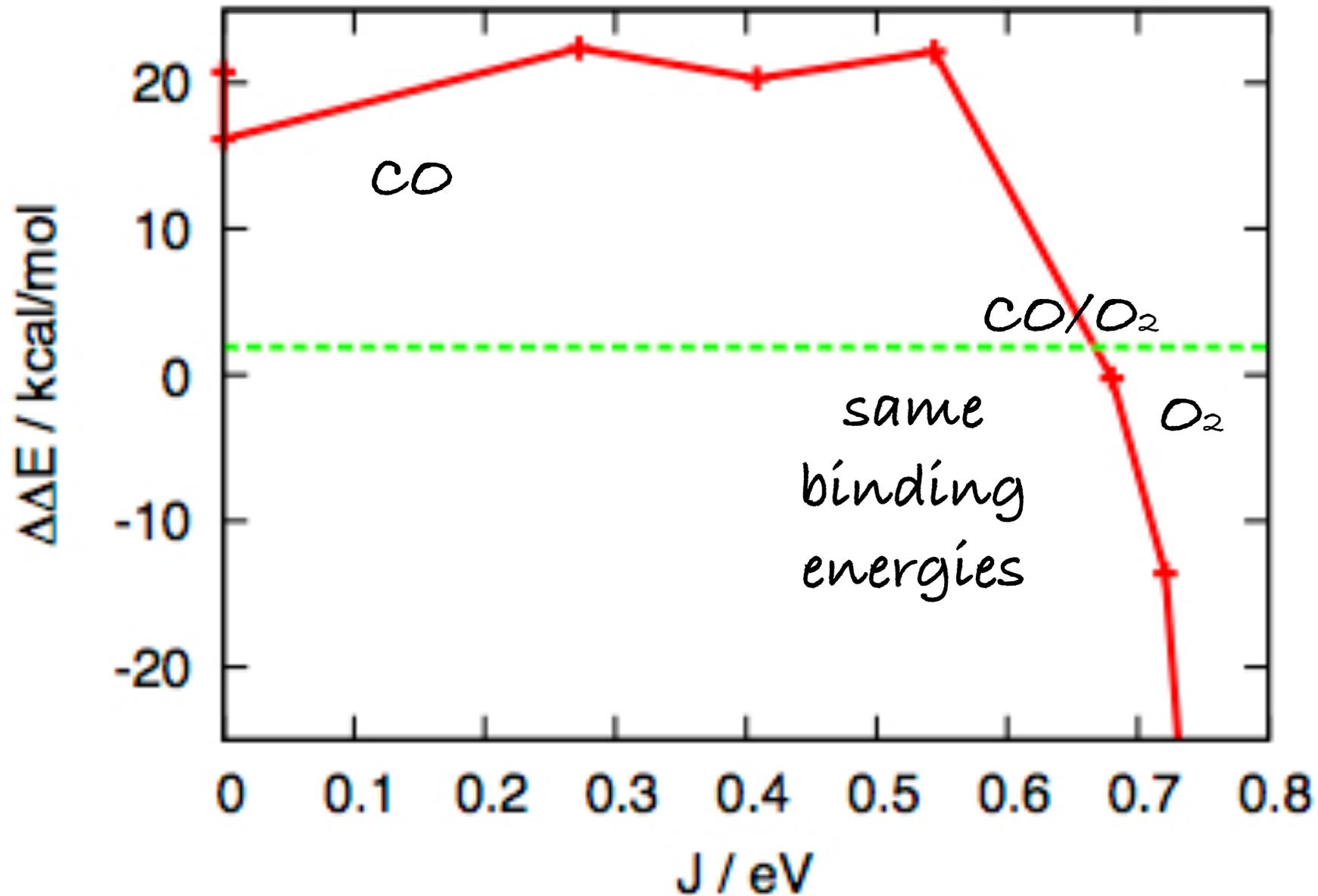
Binding discrimination



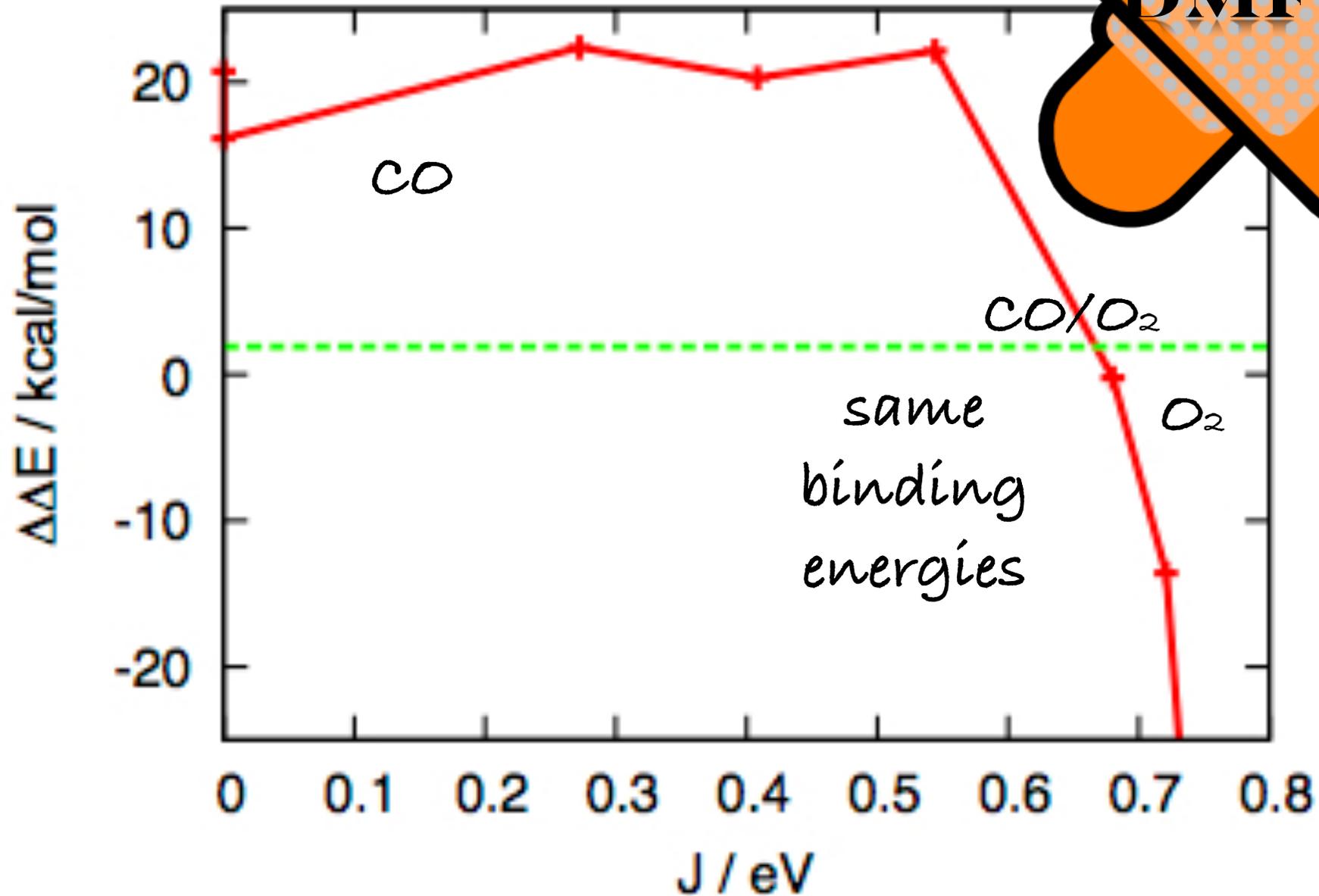
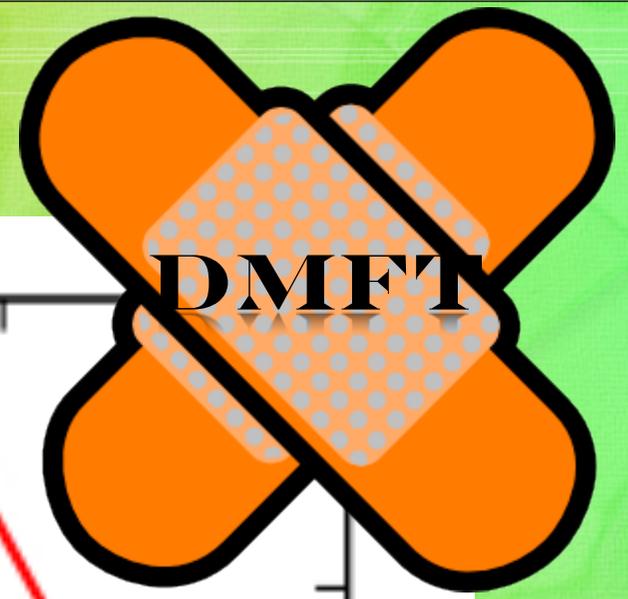
Energetics



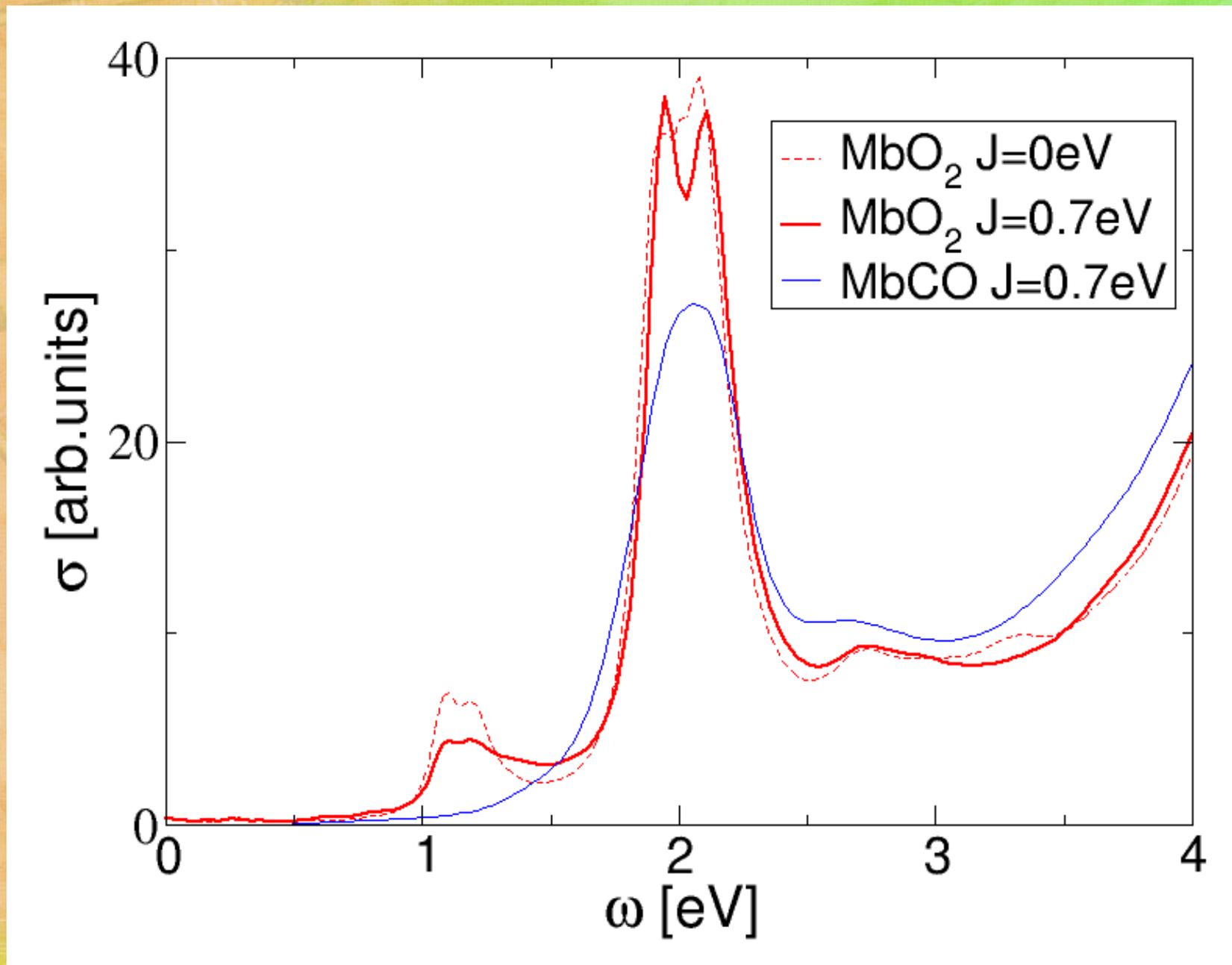
Energetics



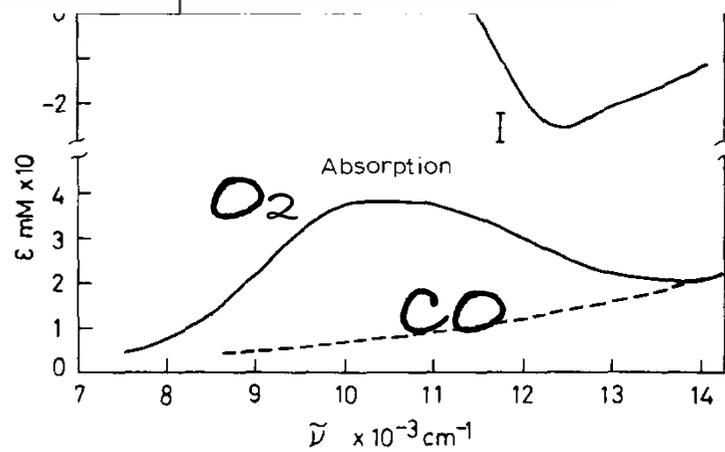
Energetics



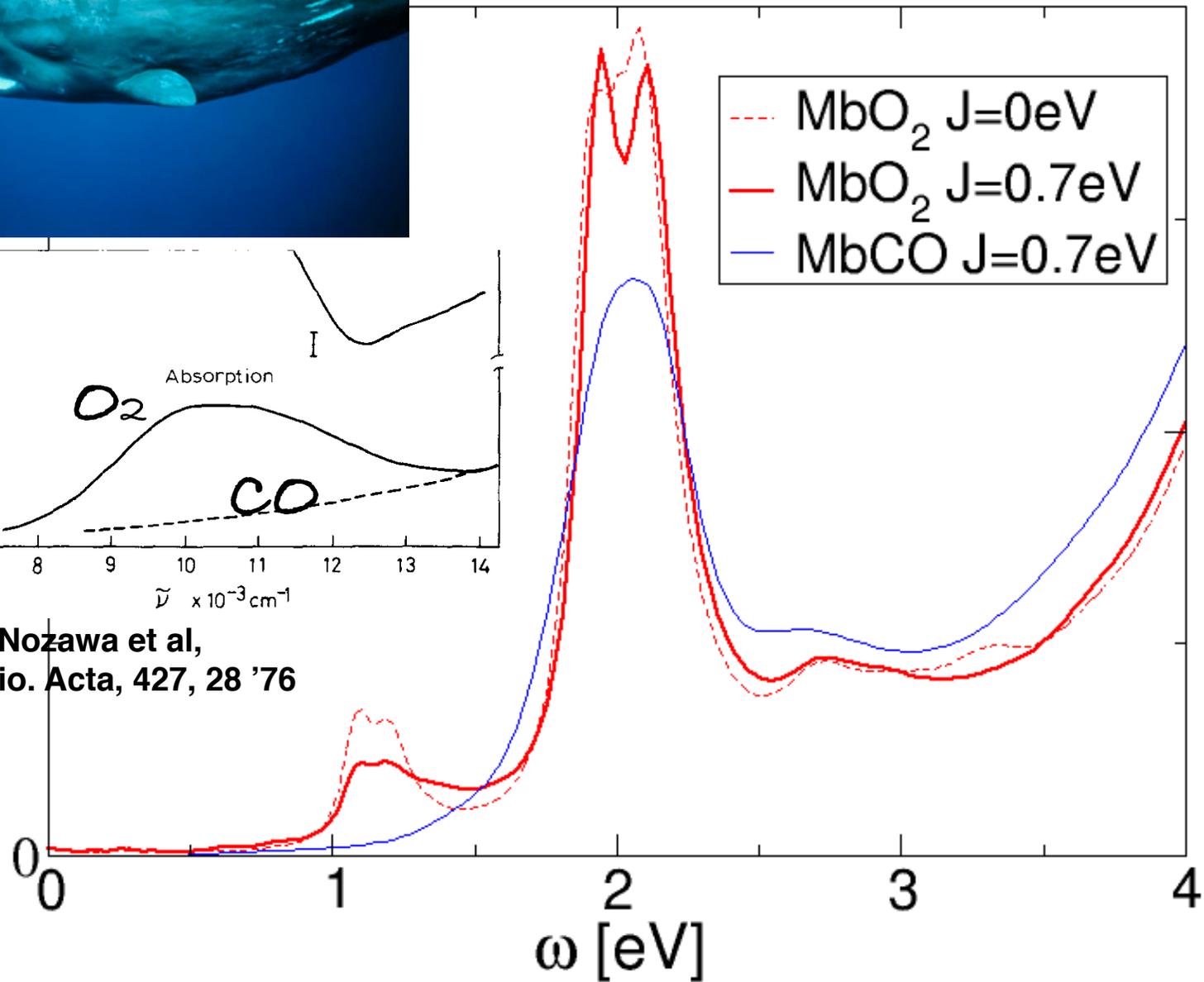
Optical absorption



Optical absorption



Nozawa et al,
Bio. Bio. Acta, 427, 28 '76

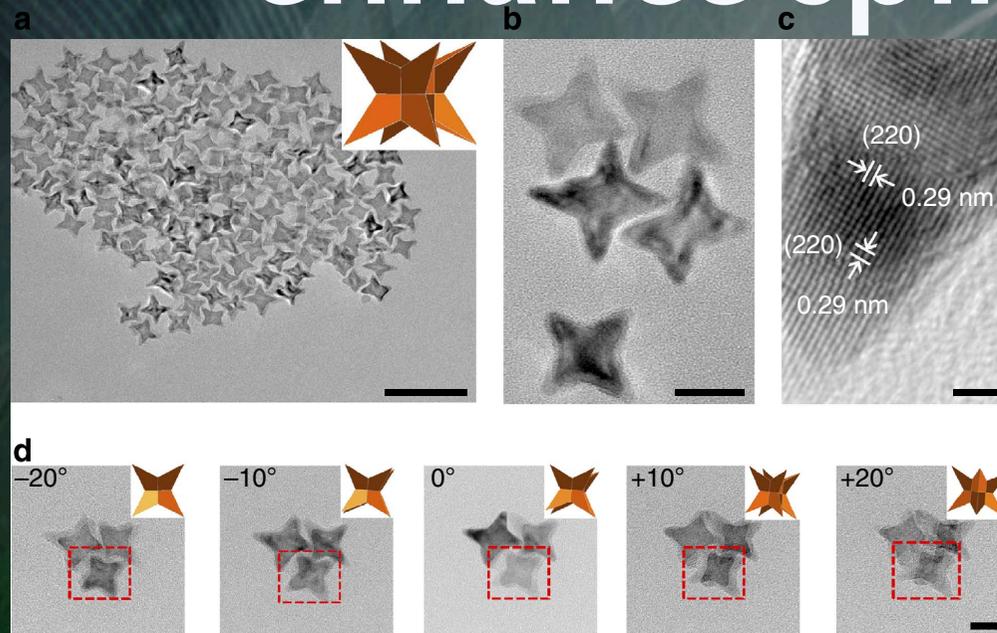


MANO:

Magnetism of Agregates of Nano-Particle Oxides

- ❑ **KCL - St Thomas-Hospital project**
- ❑ **Strongly correlated Nano-particles (Fe_2O_3)**
- ❑ **Size 5nm-50nm, supra-paramagnetic (fluctuating moment)**
- ❑ **Challenges: quantum confinement, spin canting, strong correlations, surface disorder, coating effect**
- ❑ **Applications:**
 - ❑ **MRI Contrast agent (by magnetic relaxation of the moment)**
 - ❑ **Field driven vectors (DC applied field) for targeted drug delivery**
 - ❑ **Local heat dissipation (AC applied field) for tumor treatment**

Nano-particle: doping to enhance spin relaxation



Zhao et al,
Ncomms 4:2266 '13



Doping NP Fe_3O_4 with NaCl

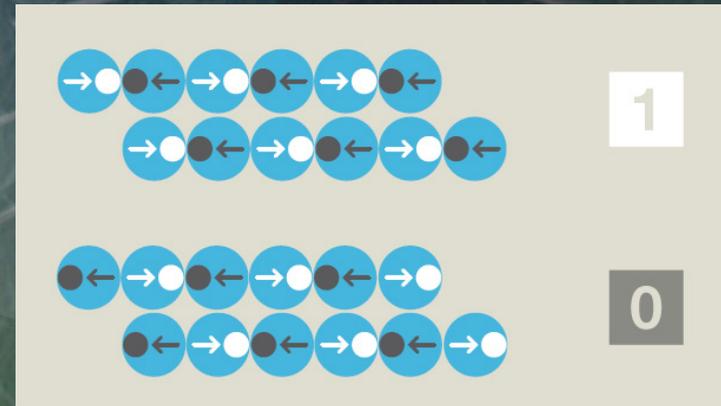
Unexpected effect: change of symmetry of the NP, it forms an octapod

Enhances the effective radius, which gives better MRI contrast

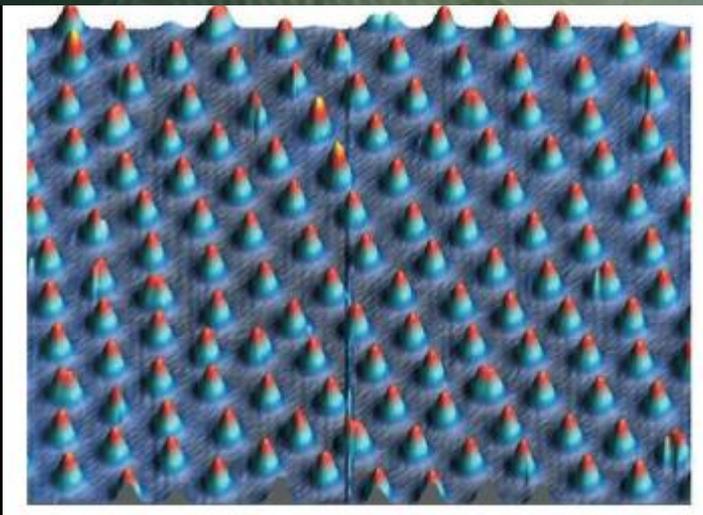
- Unexpected: doping affects both magnetism AND structural properties
- Need for predictions and microscopic understanding of the chemical substitution

Self-assembly

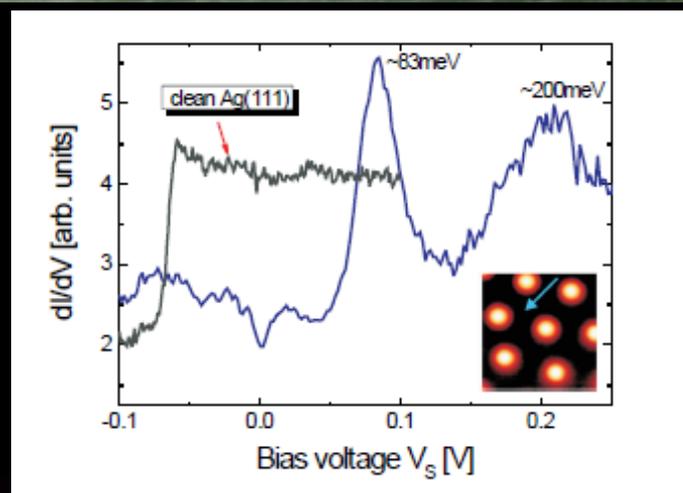
- Realization of d- and f- diluted superlattices with different inter-atom distance (Ag/Ce(111) and Fe/Cu(111)).
- Atomic scale memory device
- Not science fiction : IBM 12-Fe bit



Andreas J Heinrich's group,
Science, 335, 196 '12



a) STM image of Ce
(red) on Ag(111)

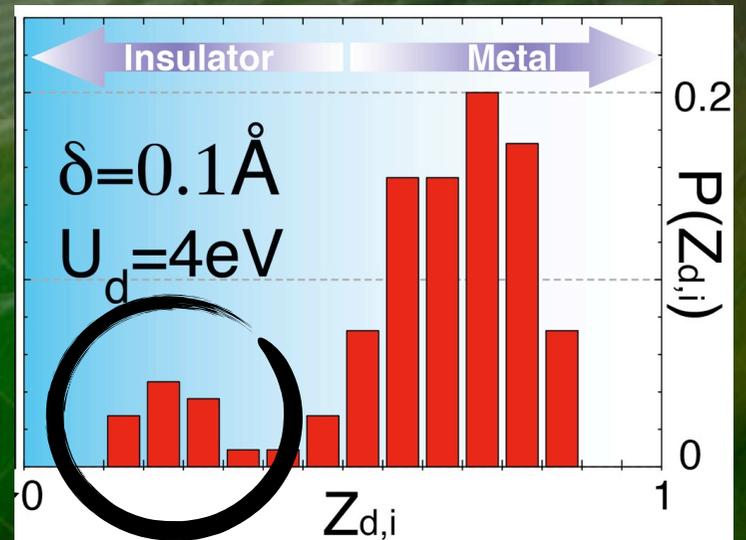
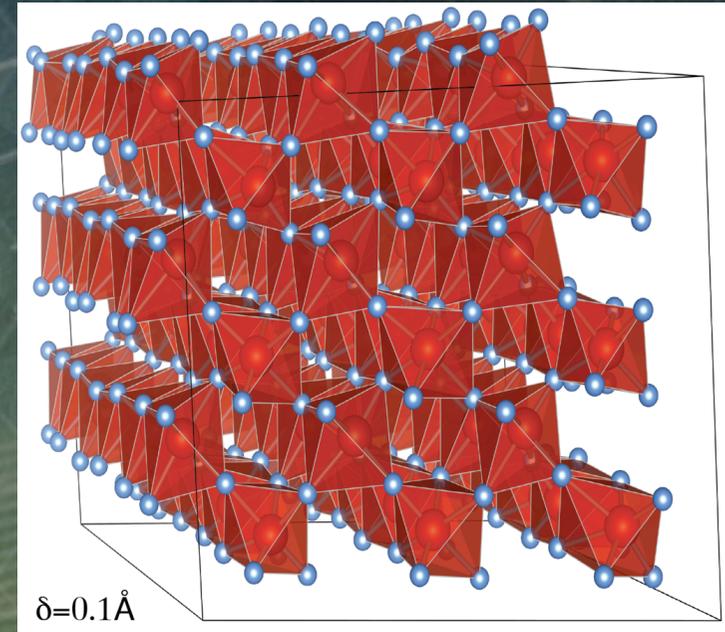
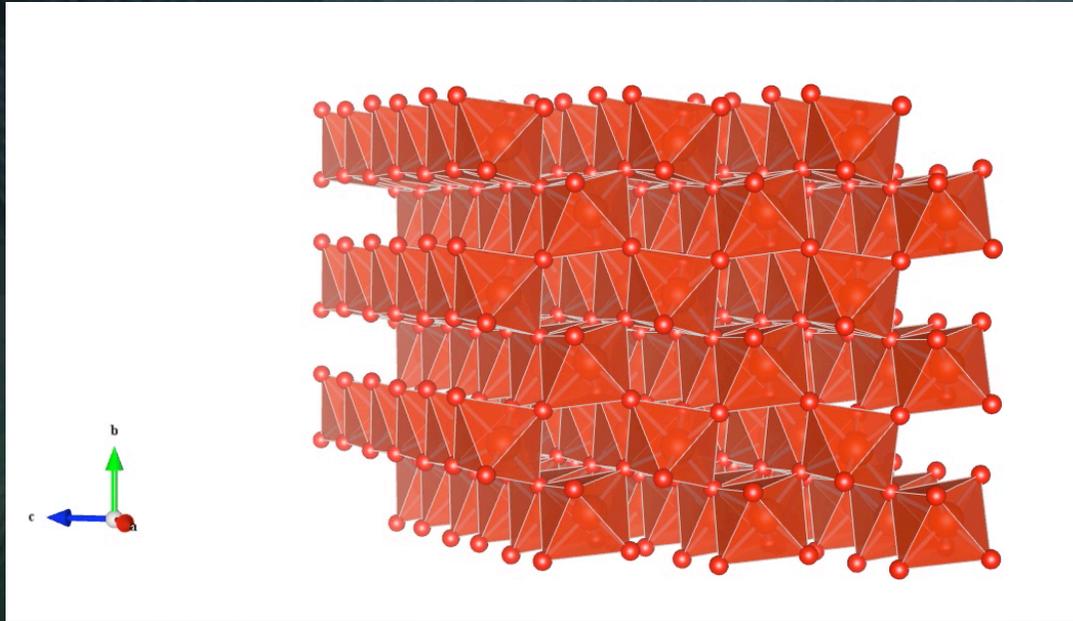


b) Spectra of clean
Ag(111) and with Ce
superlattice

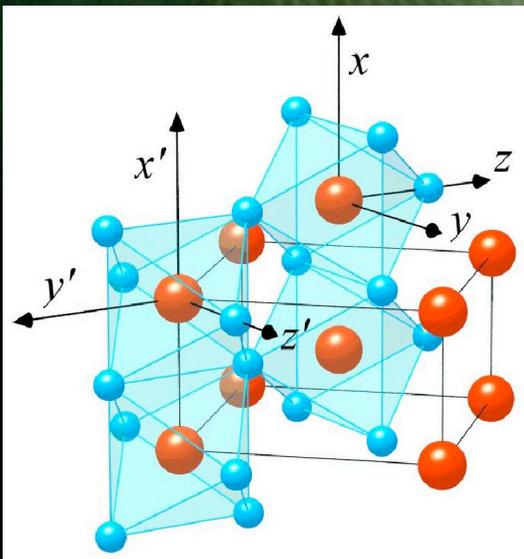
W-D Scheider's group, PRL 92, 16101 '04

Disorder in Vanadium dioxide

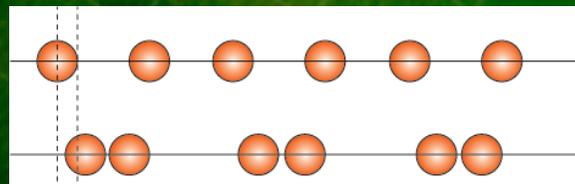
Static disorder



Monoclinic M_1 phase, rutile axis along "a" axis



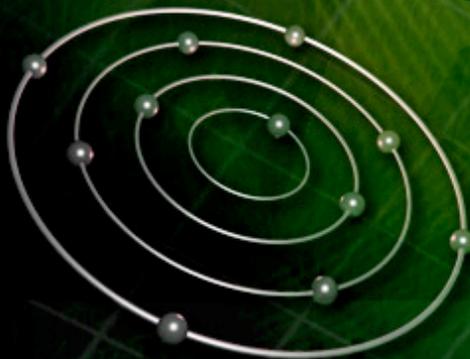
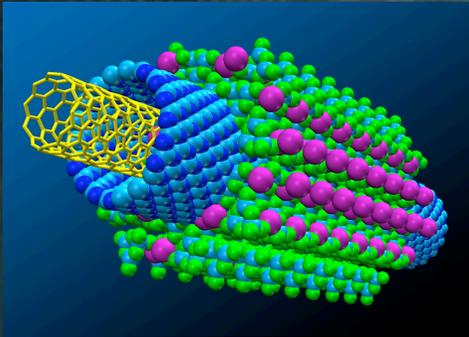
$T=290\text{K}$: structural phase transition



Insulator-like pockets



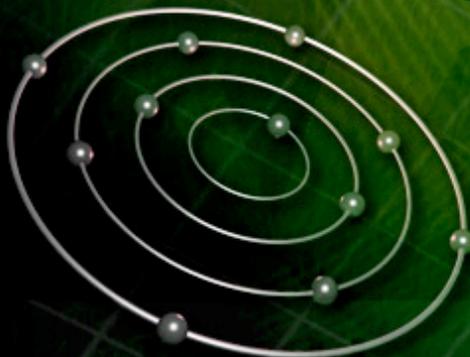
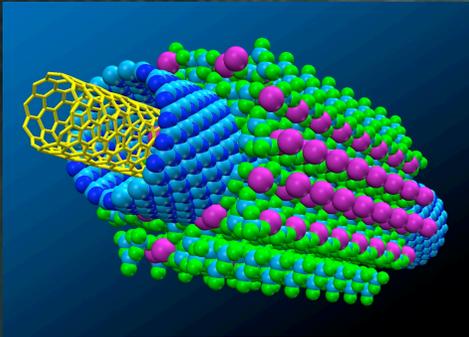
Obstacles



- professional integration : bringing the new tools to the community. Coding is professional, Delivery is very poor (tutorials, graphical interfaces, ...)
- Onetep/Castep : Part of Material Studio (Accelrys). Range of users significant, professional data pipeline.
- Range of users is small, but potential for significant advances is large, we need to make a case !
- engage better with industries, pharmaceuticals, material driven rather than method-driven
- Hub with upfront range of expertise (TYC example in London)
- Interdisciplinary science : hard to deliver, hard to review (funding agencies). On paper attractive.



Obstacles

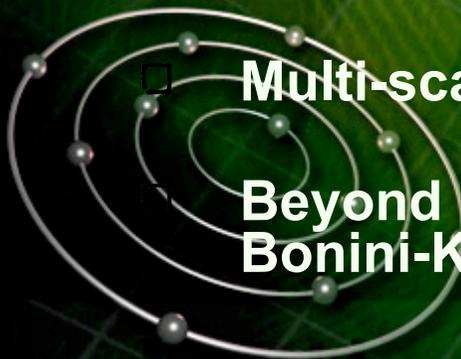


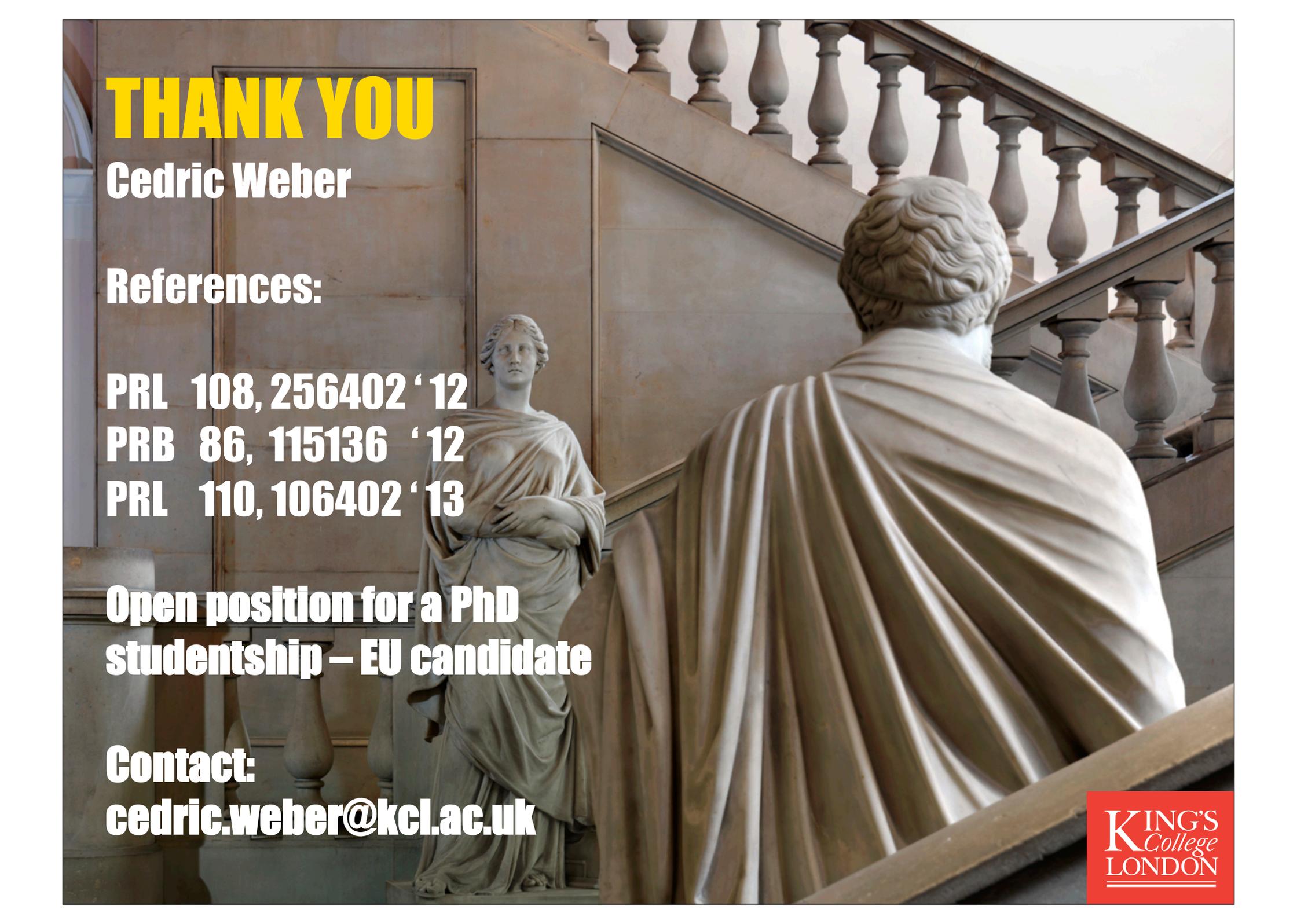
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Chelsea vs. Bayern Munich

Conclusion / Roadmap to drug design

- Promising direction : extensions to non-periodic systems (nano-crystals, molecules, self-assembly ...)
- nano-structure tailoring / drug design - Roadmap
 - Geometry optimization (now only ONETEP/DFT+U)
 - DMFT treatment restricted in energy or spatial range, need more
 - DMFT + Molecular dynamic (Forces , ...), we have energies, but we need better “value/money” algorithms
 - GW and DMFT+GW in real space for molecules
 - Data mining
 - Multi-scale approaches (effective Kohn-Sham potential ...)
 - Beyond linear response : DMFT+Boltzman equation (Nicola Bonini-KCL, transport)



The background of the entire page is a photograph of a grand, classical staircase. The staircase is made of light-colored stone and features a balustrade with decorative balusters. In the foreground, a large, white marble statue of a man, seen from the back, is draped in a heavy, flowing robe. In the middle ground, another white marble statue of a woman, also draped in a robe, stands on a lower level of the staircase. The lighting is soft, highlighting the textures of the stone and the folds of the robes.

THANK YOU

Cedric Weber

References:

PRL 108, 256402 '12

PRB 86, 115136 '12

PRL 110, 106402 '13

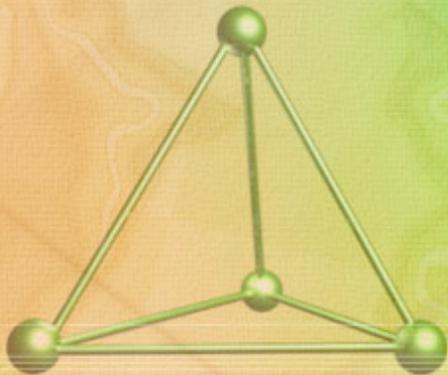
**Open position for a PhD
studentship – EU candidate**

Contact:

cedric.weber@kcl.ac.uk

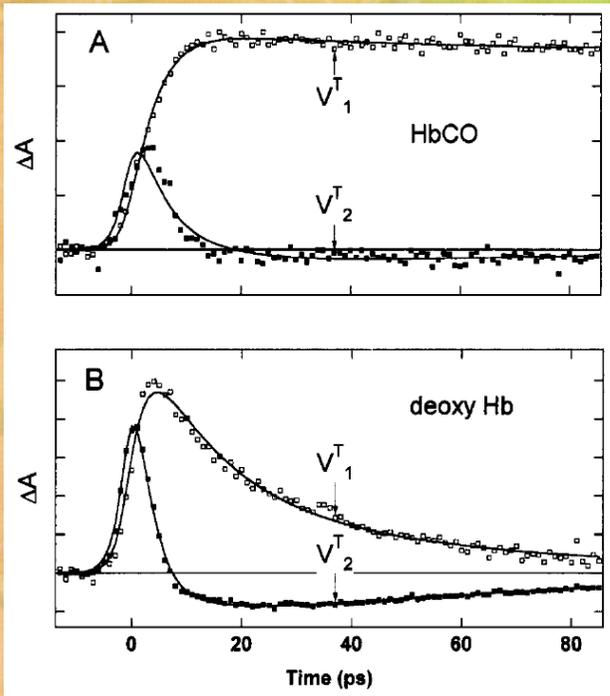
KING'S
College
LONDON

THE END

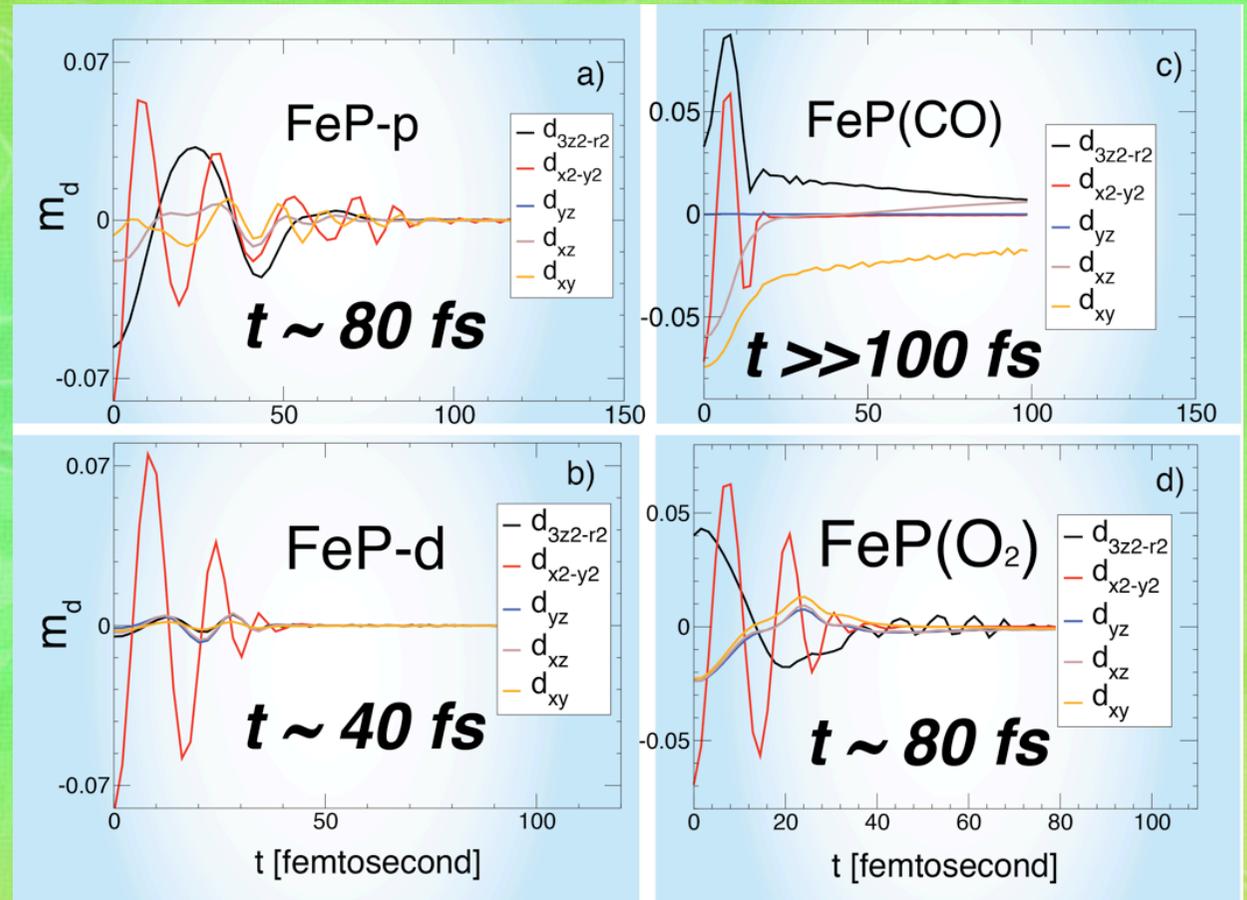
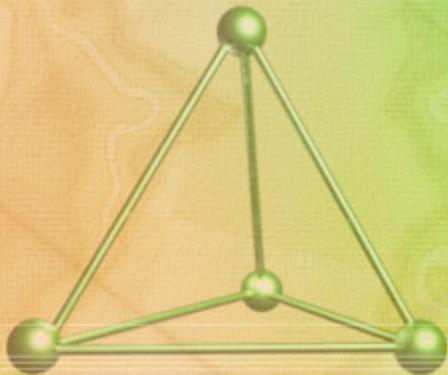


Relaxation, femto dynamics

T=0 polarization, relaxation



Photolysis excitation, shift in Raman spectra versus time, (Franzen, Biophys. Journal 80'01)



Topology / response dependence
Protein characterization by time dependence

AIM - Entanglement - bath/impurity

- Decomposition of the ground state (and excited states) in impurity and bath parts
- Reduced density matrix of the impurity ρ

$$\hat{\rho} = \sum_i e^{-\beta E_i} \text{Tr}_B |i\rangle \langle i|$$

- Diagonalization of ρ yields the von Neuman entropy:

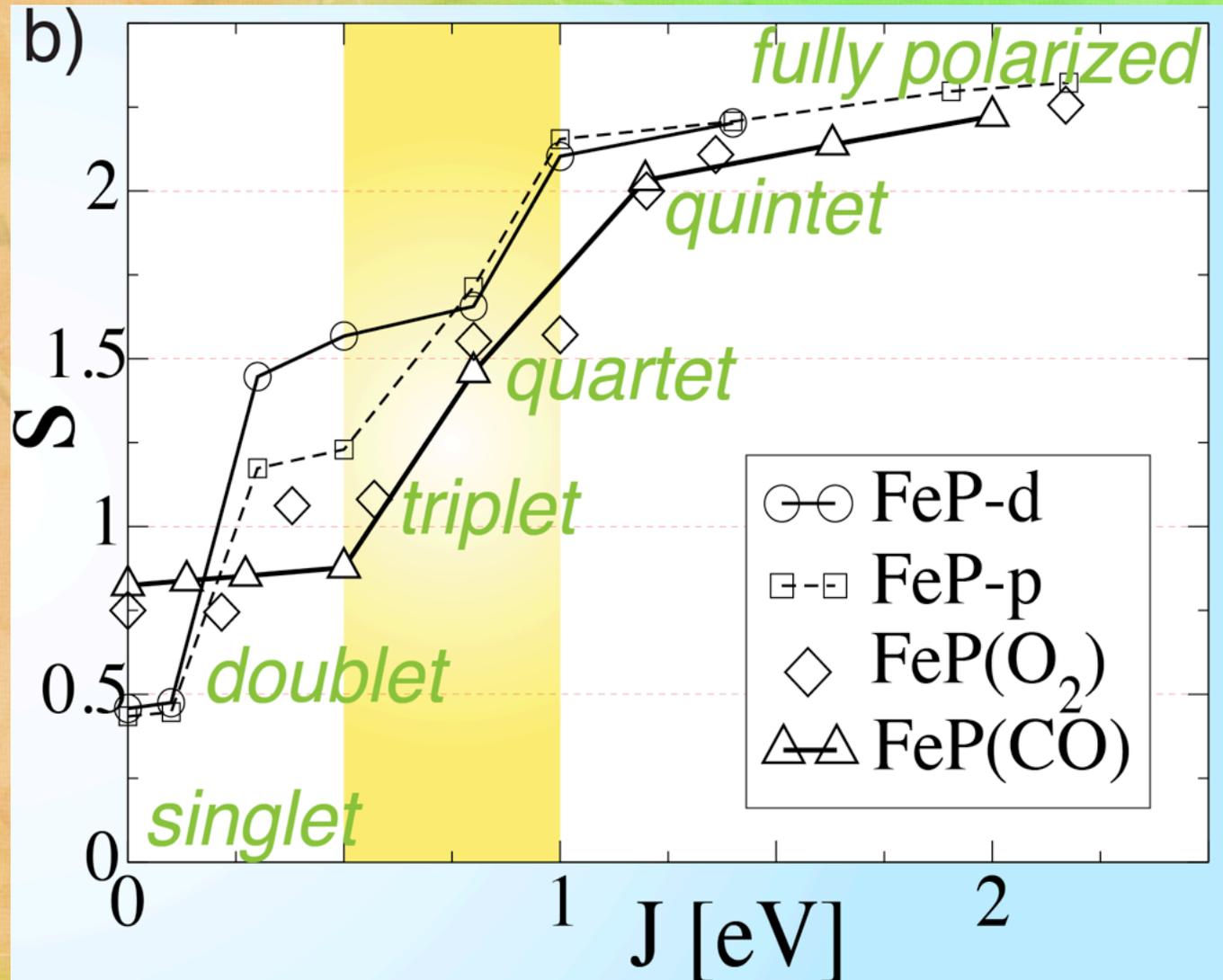
$$\Lambda = -k_B \sum_k \lambda_k \ln(\lambda_k)$$

- Eigenvectors are “cartoon” representation of the dominant states



Fluctuating magnetic moment

$$S = \sqrt{\langle \hat{S}\hat{S} \rangle - (\langle \hat{S} \rangle)^2} = \sqrt{\langle \hat{S}\hat{S} \rangle} \quad S = s(s+1)$$

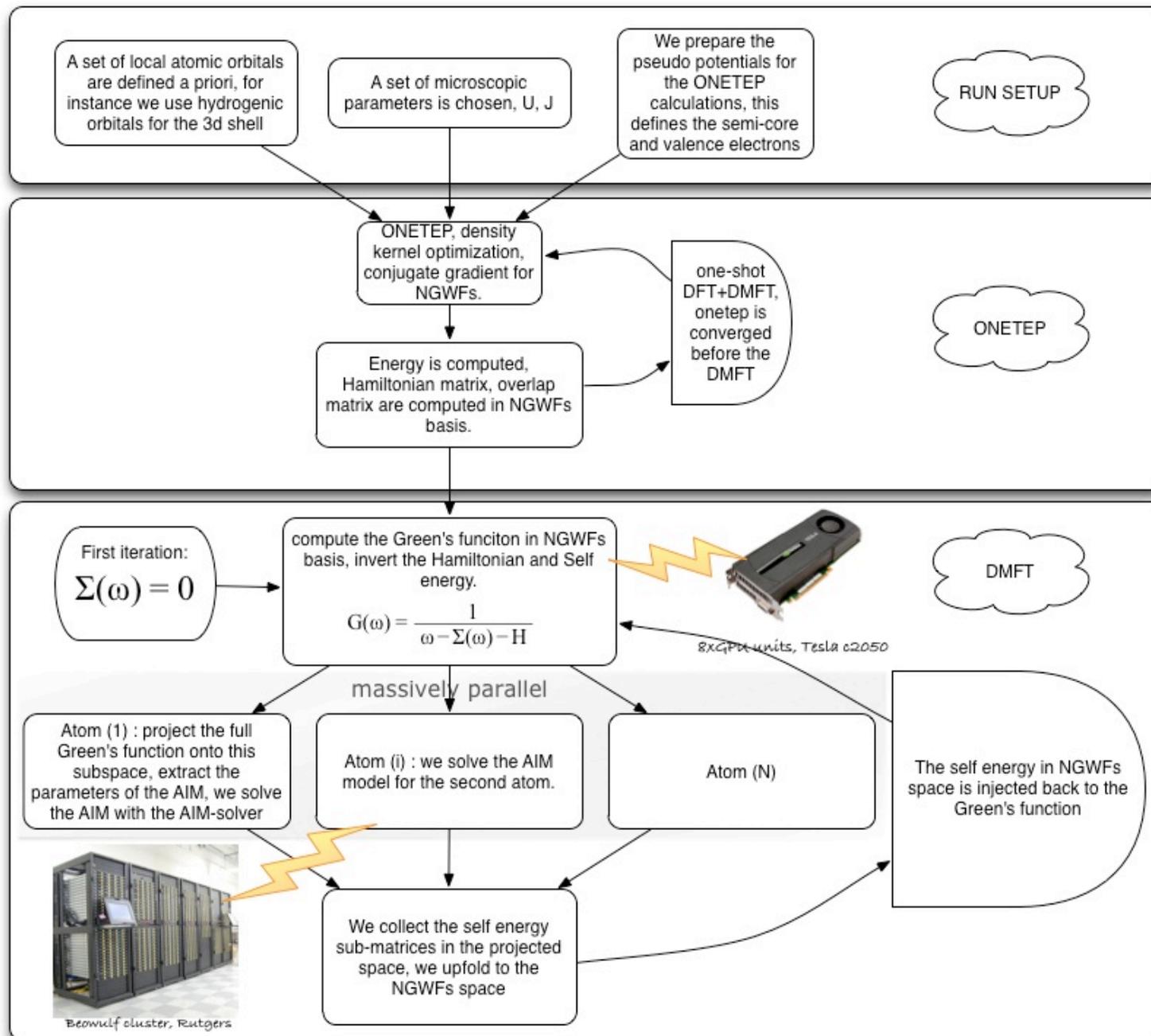


no-symmetry breaking
(paramagnetic
solution)

not in a classical
representation of a
triplet state for
 $J \sim 0.8 \text{ eV}$

Experimentally :
strong dependence of
the spin state with
respect to small
modifications in the
structure

Workflow



1) Setup the problem (pseudo-potentials, crystallographic structure, screened interactions)

2) Converge the DFT calculations

3) Invert the Self-energy and hamiltonian (GPU)

4) Project the Green's function on many atomic local problems.

5) Solve the AIM local problems in parallel (MPI+OPENMP).

6) Upfold back the projected Self-energy to the large Kohn-Sham Hilbert space

Where do the electron go?

J transfers charge to hydroxyl groups

atom	$\Delta n(r)$
Iron d orbitals	-0.52
Nitrogen ring	-0.25
hydroxyl groups	+0.77

TABLE III: Variation of the charge $\Delta n(r) = n(r, J = 0.8) - n(r, J = 0)$ in FeP induced by the Hund's coupling.

	J	$d_{x^2-y^2}$	$d_{3z^2-r^2}$	d_{xz}	d_{xy}	d_{yz}
FeP	0	0.85	1.86	1.24	1.98	0.82
FeP	0.8	1.10	1.75	1.08	1.14	1.08
FeP(CO)	0	1.06	0.86	1.99	1.06	1.99
FeP(CO)	0.8	1.14	1.33	1.16	1.05	1.85
FeP(O ₂)	0	0.72	1.82	1.25	1.87	1.28
FeP(O ₂)	0.8	1.03	1.07	1.18	1.97	1.09

**increase J :
empties
doublets**

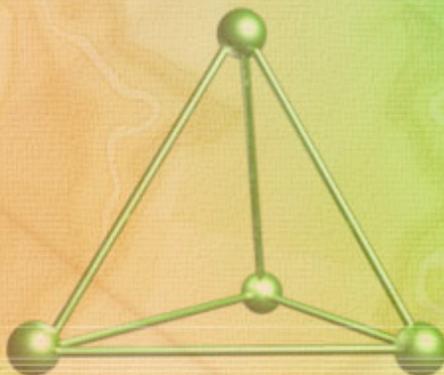
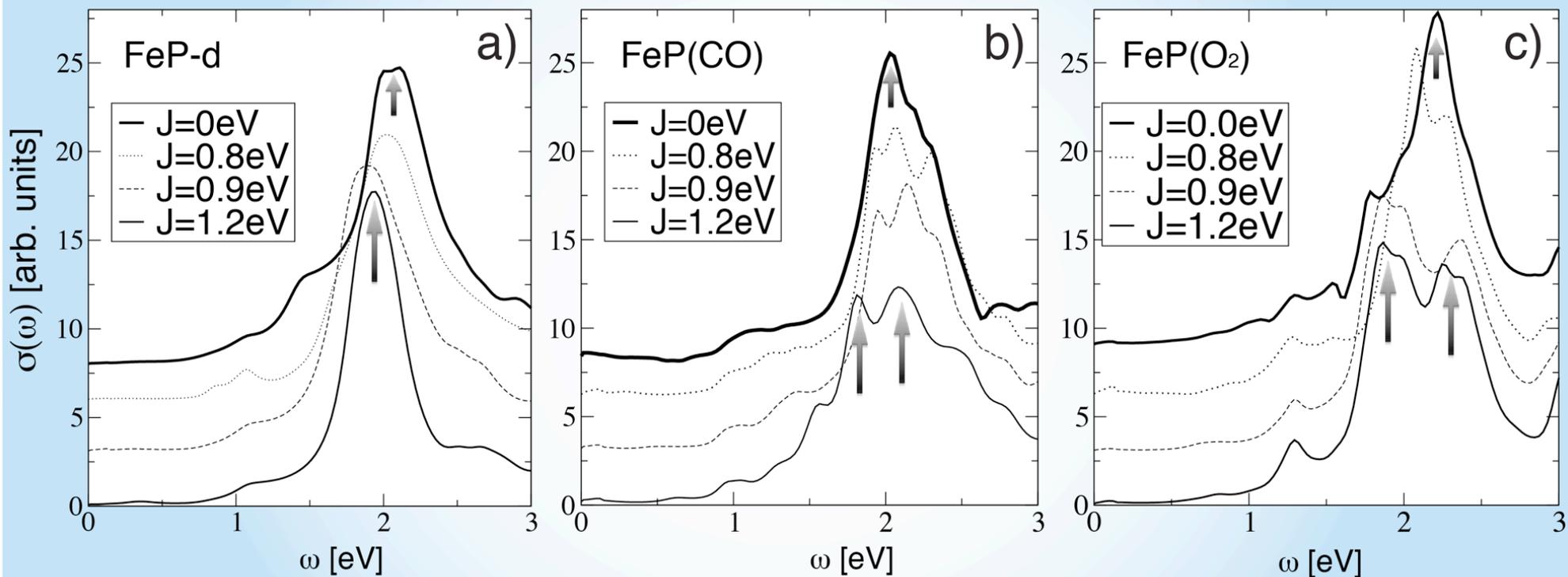
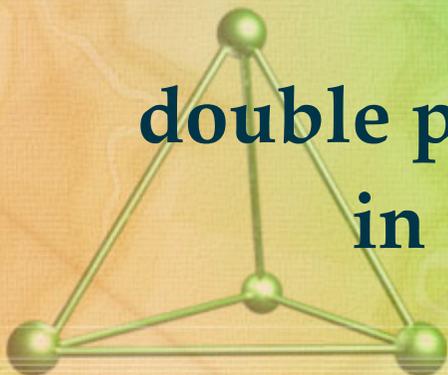


TABLE I: Average occupations n_d^α of the iron d orbitals for FeP, FeP(CO) and FeP(O₂), for $J=0$ and $J=0.8$.

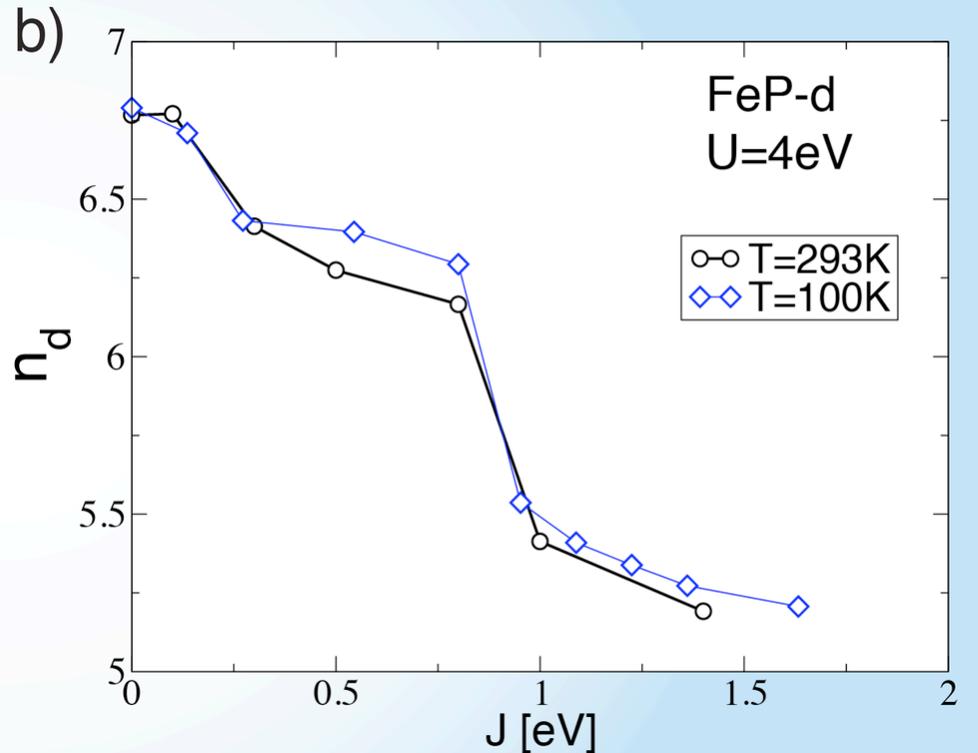
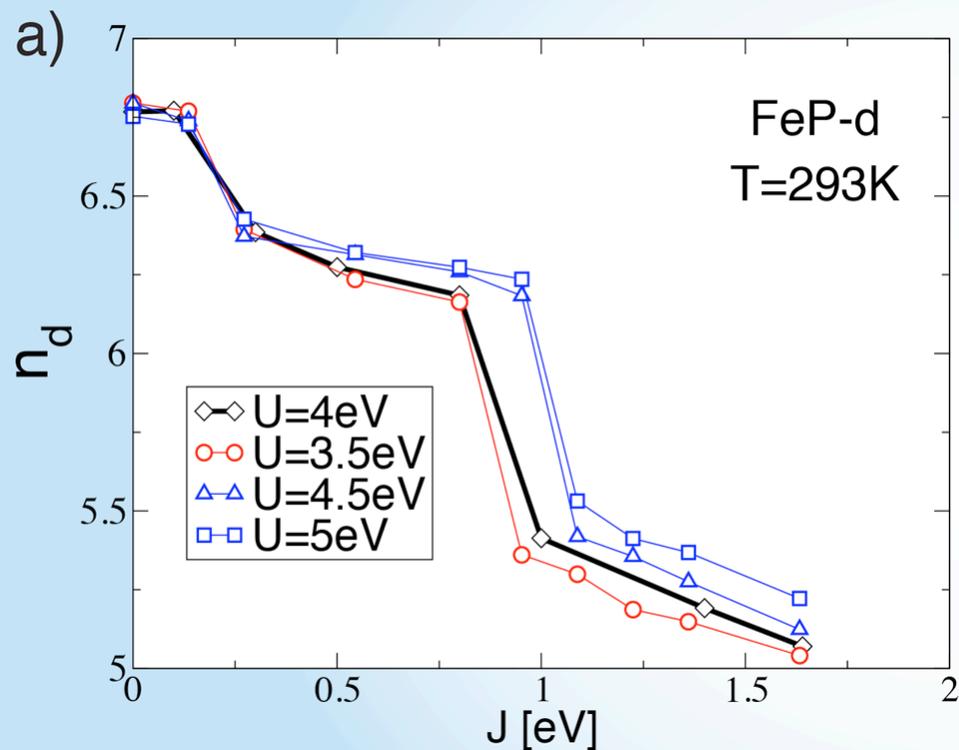
Optics : J dependence



**double peak structure (present in experiments)
in oxy-heme emerges as J increases**



U and temperature variations

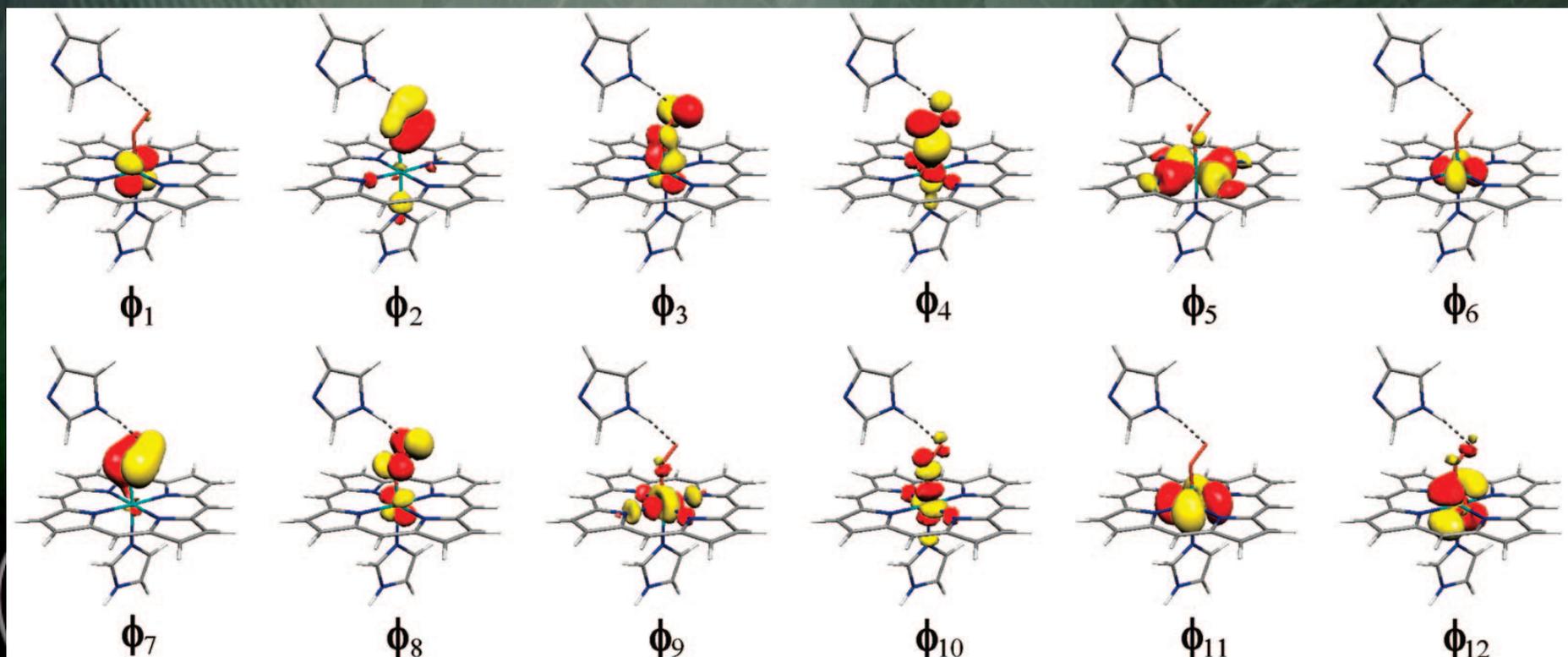


Weak U and temperature dependence

Weak variation of the charge with U for J=0

Quantum Chemistry Approaches

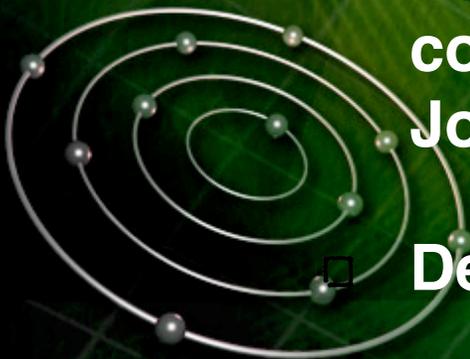
- Important to choose carefully orbitals
- Example : set of active orbitals for heme :



□ JACS, 130, 14778 ' 08

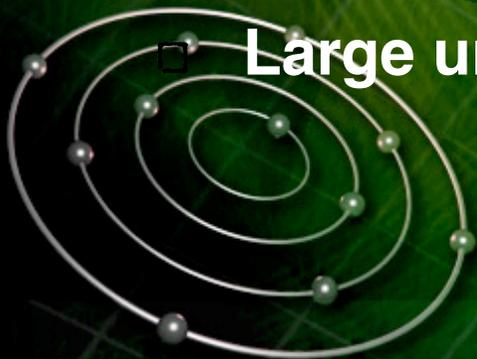
Convergence of DMFT and CI

- Hartree Fock type approaches used as solver for DMFT
- "Dynamical mean-field theory from a quantum chemical perspective"
- D. Zgid and G. Chan J. Chem. Phys., 134, 094115 (2011)
- Quantum Monte Carlo to sample CI configurations, see e.g. Booth GH, Chan GKL, Journal of Chemical Physics, 138, 029901 (2013)
- Decoupling of correlated atom from system



DFT+DMFT

- Extensive set DFT+DMFT packages in the plane wave basis
 - Wien2K+DMFT (K.Haule)
 - Wien2K+TRIQS (M. Aichhorn, M.Ferrero, O. Parcollet)
 - DFT+DMFT in LMTO basis (A.I. Lichtenstein)
 - DFT+DMFT in Abinit (B. Amadon) ... and others ...
- $\Gamma=0$ approach, requirements :
 - DMFT in localized basis set
 - Real space approach
 - Large unit-cells
 - The catch (problematic to some extent ...) :
 - non-orthogonal basis set



DMFT solver

Finite temperature Lanczos solver

AIM defined by a set of local basis of atomic orbitals (c operator) connected to a bath (a operator) :

$$H_{imp} = \sum_{\sigma=\uparrow\downarrow, ij=1}^m \epsilon_{ij\sigma} (a_{i\sigma}^\dagger a_{j\sigma} + h.c.) + \sum_{\sigma=\uparrow\downarrow, i=1}^m V_{i\sigma} (a_{i\sigma}^\dagger c_\sigma + h.c.) + U \hat{n}_\uparrow \hat{n}_\downarrow - \mu \hat{n}$$

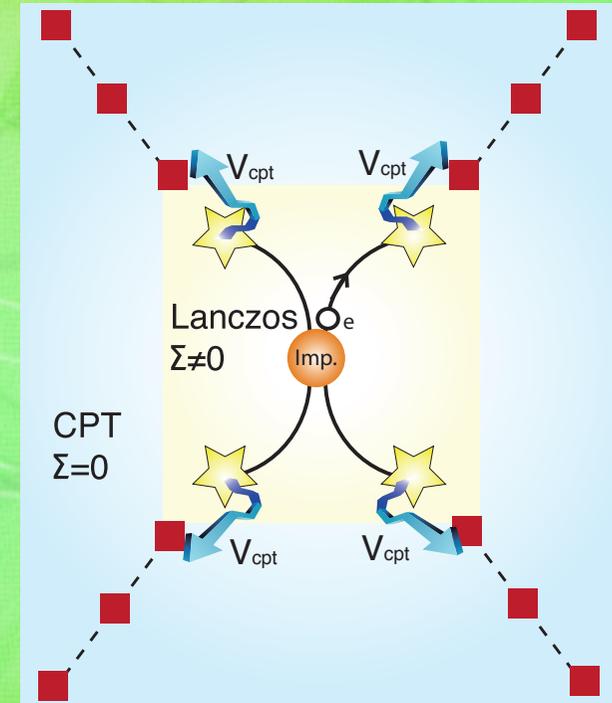
This hamiltonian yield the hybridization function:

$$\Delta^{\text{ED}}(i\omega_n) = \mathbf{V}^\dagger (i\omega_n - \epsilon)^{-1} \mathbf{V}$$

We enforce that Himp reproduces the DMFT hybridization:

$$d = \sum_{\omega < \omega_0} \frac{|\Delta^{\text{ED}}(i\omega_n) - \Delta(i\omega_n)|^2}{\omega_n}$$

Hybrid Lanczos solver



cluster perturbation theory (CPT) :

$$\lambda \left(\sum_i |V_{CPT}^i|^2 \right)$$

$$\mathbf{G}^{CPT}(i\omega_n) = \frac{1}{\mathbf{G}^{V_0}(i\omega_n)^{-1} - \mathbf{V}_{CPT}}$$

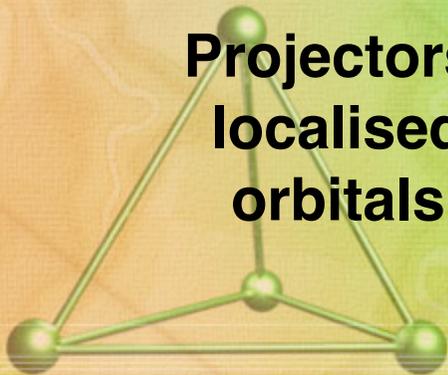
Molecular dynamical mean-field theory

Lattice Dyson equation: $G_{i\sigma l;j\sigma'm}(\omega) = G_{i\sigma l;j\sigma'm}^0(\omega) + [G^{(0)}(\omega)\Sigma(\omega)G(\omega)]_{i\sigma l;j\sigma'm}$

GF Matrix representation:

$$\hat{G}^{-1}(\omega) = \begin{pmatrix} \omega + \mu - \Sigma_1(\omega) & t_{12} & t_{13} & \dots & t_{1N} \\ t_{21} & \omega + \mu - \Sigma_2(\omega) & t_{23} & \dots & t_{2N} \\ t_{31} & t_{32} & \omega + \mu - \Sigma_3(\omega) & \dots & t_{3N} \\ \dots & \dots & \dots & \dots & \dots \\ t_{N1} & t_{N2} & t_{N3} & \dots & \omega + \mu - \Sigma_N(\omega) \end{pmatrix}$$

Projectors,
localised
orbitals



Local projected Green's function:

$$G_{ii}^{-1}(\omega) = g_{ii}^{-1}(\omega) - \Sigma_i(\omega)$$

DMFT
solver

DMFT density kernel

Green's function written in the basis of a set of NGWFs :

$$G^{\alpha\beta}(i\omega_n) = ((i\omega_n + \mu)S_{\alpha\beta} - H_{\alpha\beta} - \Sigma_{\alpha\beta})^{-1}$$

DMFT - projection on a set of atomic wave-function $\{\phi\}$:

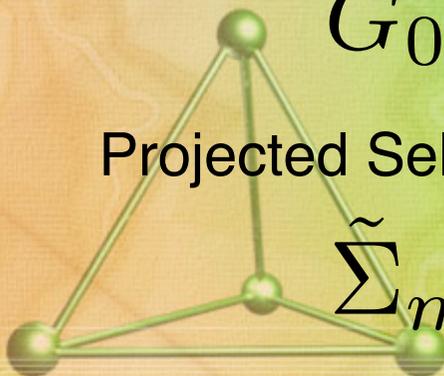
$$W_{m\alpha}^{(I)} = \langle \varphi_m^{(I)} | \phi_\alpha \rangle \quad V_{\alpha m}^{(I)} = \langle \phi_\alpha | \varphi_m^{(I)} \rangle$$

Projected Green's function:

$$\tilde{G}_{0mm'}(i\omega_n) = W_{m\alpha} G^{\alpha\beta}(i\omega_n) V_{\beta m'}$$

Projected Self energy:

$$\tilde{\Sigma}_{mm'}(i\omega_n) = W_{m\alpha} \Sigma^{\alpha\beta}(i\omega_n) V_{\beta m'}$$



Anderson Impurity Model

DMFT AIM local problem Hybridization of the AIM is given by:

$$\Delta(i\omega_n) = (i\omega_n + \mu) \tilde{\mathbf{O}} - \tilde{\Sigma} - \mathbf{E}^{imp} - \tilde{\mathbf{G}}^{-1}$$

with :

$$\tilde{\mathbf{O}} = (\mathbf{W}\mathbf{S}^{-1}\mathbf{V})^{-1} \quad E^{imp} = \tilde{\mathbf{O}}\mathbf{W} (\mathbf{S}^{-1}\mathbf{H}\mathbf{S}^{-1}) \mathbf{V}\tilde{\mathbf{O}}$$

Obtain the self-energy from the local problem, and upfold back to NGWF space. How can we upfold ? It should be the inverse operation :

$$\tilde{\Sigma}(\omega = \infty) = \tilde{\mathbf{O}}\mathbf{W} (\mathbf{S}^{-1}\Sigma_{upfolded}(\omega = \infty)\mathbf{S}^{-1}) \mathbf{V}\tilde{\mathbf{O}}$$

$$\Sigma_{upfolded} = \mathbf{V}\tilde{\Sigma}\mathbf{W} \quad (\tilde{\mathbf{O}}\mathbf{W}\mathbf{S}^{-1})\mathbf{V} = 1$$

$$\mathbf{W}(\mathbf{S}^{-1}\mathbf{V}\tilde{\mathbf{O}}) = 1$$

Causal ! But this simplification is only for $\Gamma=0$! The k dependence of the overlap matrix complicates everything.