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Electron-Phonon Coupling in Luminescent Europium Doped Hydride Perovskites Studied by Luminescence Spectroscopy, Inelastic Neutron Scattering, and First-Principle Calculations

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Abstract

We present a case study on the vibrational coupling of lattice phonons to the electronic $4f^7$ ($^8S_{7/2}$) - $4f^65d^1$ (e_g) transition of divalent europium in the hydrides and deuterides LiMH_3 and LiMD_3 ($M = \text{Sr}, \text{Ba}$). For low doping concentrations, these compounds show extraordinarily well-resolved vibronic fine structures at low temperatures. Besides luminescence emission spectroscopy of the europium-doped compounds, we carried out inelastic neutron scattering experiments of the europium-free compounds. The phonons coupling to the electronic transition are identified and a good agreement between the vibronic and the inelastic neutron scattering data is found. The low energy acoustic modes do not significantly change upon replacing hydride by deuteride, whereas a decrease by a factor of approximately $\sqrt{2}$ can be observed for the higher energy optic modes. Furthermore, we compare these experimental results to density functional calculations performed with the Vienna Ab Initio Simulation Package. Knowledge of the phonons of a host material is of great importance, because phonons have a large influence on the optical properties, such as linewidths or luminescence quenching. Hydride-containing host lattices are an ideal model system, since ^1H can easily be replaced by ^2D , so that isotope effects can be investigated.

Introduction

Rare earth-doped compounds play an important role in many optical applications, such as phosphors and plasma screens,¹⁻⁴ optical fibers and waveguides,⁵ bio-imaging and sensors,⁶⁻⁹ or quantum information technology and spectral filtering.¹⁰⁻¹² While the $4f$ electrons are so well-shielded by the filled $5s^2$ and $5p^6$ shells that the crystalline environment can be considered as a weak perturbation of the free ion level,^{10,13} the $5d$ electrons of the excited state of $4f$ - $5d$ transitions of ions such as Eu^{2+} , Ce^{3+} , or Yb^{2+} remain unshielded and can therefore show a large energy dependence of the $5d$ barycenter and the crystal field splitting on the crystalline environment.¹⁴ As a result, very different excitation and emission energies

are found, for example for Eu^{2+} or Ce^{3+} , in different host lattices.^{15,16} Additionally, except for rare cases of very high lying 5d levels and emission from, for example, the $^6\text{P}_{7/2}$ ^{17,18} or the $^6\text{I}_{7/2}$ ¹⁷ state of Eu^{2+} , the absorption and emission bands in these compounds are rather broad. For such broad 5d-4f emission spectra, it is, even at low temperature, rare that a well-resolved vibrational fine structure showing coupling to several modes can be observed, because the coupling strength is usually intermediate, i.e. the Huang-Rhys parameter S has the value $1 < S < 5$.¹⁹ However, some rare examples exist for Ce^{3+} , Pr^{3+} , and Tb^{3+} with well-resolved spectra that show a coupling to two vibrations in double perovskites.^{20–22} Recently, new types of host materials, ionic hydrides,^{23–27} complex hydrides,^{28,29} and mixed anionic compounds containing hydride,^{30–33} have been studied as hosts for Eu^{2+} -luminescence. Here, it was found that replacing less polarizable anions by the polarizable hydride anion can lead to wide redshifts of barycenter of the 5d levels. However, beside the interesting properties with regard to the centroid shift and crystal field splitting, hydrides as host lattices can also serve as model systems to study the correlation of the optical properties with lattice phonons. Already in 1980, Judd predicted that vibronic intensities increase with the degree of covalency,^{19,34} which is directly connected to the anion polarizability. Thus, ideal candidates for such studies are compounds with polarizable anions, such as nitrides or hydrides. Moreover, the replacement of ^1H by ^2D is an elegant tool for such studies. Knowledge on the relation between lattice phonons and electronic transitions is of great importance, because phonons can have a large influence on the optical properties, such as linewidths, luminescence quenching or the efficiency of up-conversion processes.^{35–39}

Here, we present a case study of LiMH_3 and LiMD_3 ($M = \text{Sr}, \text{Ba}$), which crystallize in the inverse cubic perovskite structure type in space group $Pm\bar{3}m$ (see Fig. 1).^{24,40–42} LiEuH_3 shows the same structure⁴³ with similar lattice parameters as those observed in LiSrH_3 and we assume that Eu^{2+} will occupy the M^{2+} site in doped compounds.

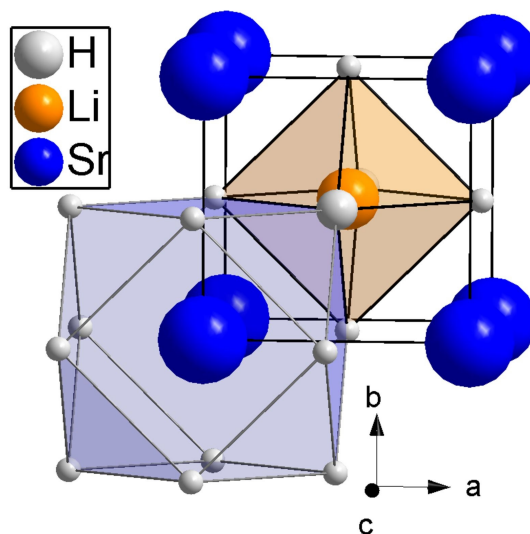


Figure 1: Crystal structure of LiSrH_3 showing the cuboctahedral surrounding of strontium and the octahedral coordination of lithium.

The Eu^{2+} -doped compounds represent an ideal model system for investigations of the vibrational coupling. In our previous study,²⁴ where we reported on the luminescence properties of samples with rather high europium concentrations, we were already able to detect some vibrational fine structure at low temperature. Here, we use highly diluted samples (europium concentrations of approx. 0.005 %) in order to obtain a better resolution of the fine structure. We carried out photoluminescence measurements on the samples with low europium concentrations as well as inelastic neutron scattering of the europium-free compounds LiBaH_3 , LiBaD_3 , and LiSrH_3 at 20 K. In an earlier work, we had confirmed the absence of phase transitions down to 10 K.²⁴ The obtained results are compared with the phonon dispersion curves and densities of states obtained by first-principle calculations.

Methods

Synthesis

Eu²⁺-doped samples LiMH₃ and LiMD₃ (M = Sr, Ba) for photoluminescence measurements were prepared by hydrogenation of the alloys. Alloys were synthesized by melting reactions of the elements (strontium, Alfa Aesar, distilled dendritic pieces 99.8%; europium rod, Alfa Aesar 99.8%, ; barium rod, Chempur, 99.3% (rest strontium); lithium, Alfa Aesar 99.8%; all mechanically surface cleaned) in niobium ampoules (at approx. 1100 K, depending on the metals used). In order to obtain low europium concentrations, samples of Sr:Eu with 0.5 mol% were prepared first and then diluted in order to reach the desired concentrations. Then, the alloys were hydrogenated in a hydrogen resistant autoclave (Inconel 718) at approx. 750 K and 100 bar H₂-pressure for 2-3 days (H₂ Praxair 99.8%, D₂ Praxair 99.8%). These samples contained no or only small amounts of the binary hydrides. Large amounts of europium-free LiMH₃ and LiMD₃ (M = Sr, Ba) samples for inelastic neutron scattering experiments were prepared from the binary hydrides (MH₂ had been obtained by hydrogenation of the metal before; starting materials LiH abcr 99.0 %, LiD abcr, 99.4%, barium Chempur, 99.3 %, strontium, Chempur, 99.0 %). The binary hydrides were mixed and heated in a hydrogen resistant autoclave (Inconel 718) at approx. 850 K and 20-40 bar for 2-5 days. Some of these samples contained more than 25% of the binary hydrides (compositions of the samples are given in the Supporting Information, Table 2). As large amounts of samples are needed for inelastic neutron scattering, the first synthesis approach was not suitable here and the presence of unreacted binary hydrides could not be avoided completely. Due to moisture sensitivity, all starting materials and products were handled in an argon-filled glovebox.

X-ray diffraction

Laboratory XRPD data of the samples used for inelastic neutron scattering (INS) were collected on a Huber G670 diffractometer with Guinier geometry at T = 297(2) K with CuK_{α1}

radiation. Laboratory XRPD data of the samples used for photoluminescence measurements were collected on a STOE Stadi P powder diffractometer (Stoe & Cie GmbH) with $\text{CuK}_{\alpha 1}$ radiation (Ge(111) monochromator) and Dectris Mythen DCS 1K Detector. For both cases, flat transmission samples with Apiezon grease between two Kapton foils were used. Crystal structures were refined using the program TOPAS [TOPAS version 5; Bruker AXS, www.bruker-axs.com]. Refined lattice parameters and the amount of residual binary hydride (deuteride) are given in the Supporting Information, Table 2.

Photoluminescence measurements

Photoluminescence spectra were recorded on a set-up constructed in-house with a tunable optical parametric oscillator pumped by a Nd:YAG laser (Ekspla NT342B-SH with 6 ns pulse length) as excitation source, a Jobin-Yvon HR250 monochromator, and a PI-Max ICCD camera for detection. Samples (enclosed in silica tubes of about 5 mm diameter) were placed in a closed-cycle helium cryostat (Janis CTI-Cryogenics Model CCS-150, together with a LakeShore temperature controller) and attached to the sample holder using high purity silver paint and copper sheets. Decay curves were recorded using a Jobin-Yvon HR250 monochromator and a photomultiplier tube.

Inelastic neutron scattering (INS)

Inelastic neutron scattering experiments were carried out at the high resolution broad band spectrometer TOSCA (indirect geometry time-of-flight spectrometer)^{44,45} at the ISIS Facility at 20 K. The signal was collected by detector banks in forward and backward scattering and for data analysis, the arithmetical average of those was used. To minimize the background, short-time high energy neutrons and the Γ -flash were removed by a Nimonic chopper. Additionally, the tail-cutter of the chopper also removes low energy neutrons in order to prevent frame overlap. The beam size of the instrument is about 4 x 4 cm. Samples were loaded into indium-wire sealed, flat-plate aluminum cans in an argon-filled glove box. Since MH_2

and MD₂ (M = Sr, Ba) were present in the corresponding samples, spectra of the binary hydrides were taken from the literature^{46,47} for subtraction or comparison.

Density functional calculations

Density functional calculations were carried out using the Vienna Ab initio Simulation Package,^{48–50} Version 5.4.1 together with PAWs.^{51,52} The Perdew-Burke-Ernzerhof (PBE) generalized gradient approximation (GGA)⁵³ was used to evaluate the exchange and correlation energies. PAW potentials with the valence electrons 5s²5p⁶6s² for Ba and 4s²4p⁶5s² for Sr atoms were used. In order to guarantee the reliability of the results, an optimization of the cutoff energies as well as the k-point meshes were carried out. For lithium strontium and lithium barium hydride, convergence for the total energy was obtained with a 9x9x9 Monkhorst-Pack grid centered at the Γ -point, resulting in 35k-points in the first irreducible Brillouin zone, and a plane-wave expansion cut-off of 1000.0 eV. Lattice parameters and atomic positions were optimized by the minimization of the Hellman-Feynman forces using a quasi-Newton algorithm. Criteria of 10⁻⁶ eV per Ångström and 10⁻⁹ eV were adopted for the convergence of the total energy and Hellman-Feynman forces, respectively. All performed calculations were non spin-polarized, since the compounds are known to be non-magnetic. Phonon dispersion curves were calculated with the PHONOPY code⁵⁴ using the density functional perturbation theory (DFPT).

Results

Vibronic transitions

Theoretical considerations

Electron-phonon interactions can give rise to vibronic transitions, which are optical transitions that simultaneously show a change in the electronic state and the vibrational state of

the system.¹⁹ For weak electron-phonon interactions, rather sharp lines can be observed,¹⁰ whereas a strong electron-phonon interaction yields broad bands and the influence of phonons also shows a temperature-dependence.³⁵

For a transition between two electronic states $\Psi_{(\text{electronic ground state, } n)}$ and $\Psi_{(\text{electronic excited state, } n')}$ with different initial and final vibrational states, the matrix element is given as follows:

$$\langle \Psi_{(\text{electronic ground state, } n)} | \vec{\mu} | \Psi_{(\text{electronic excited state, } n')} \rangle \quad (1)$$

with the transition electric dipole operator $\vec{\mu}$.¹⁹ The optical transition can consist of a number of vibronic transitions for the vibrational states $n'-n$. However, in the present study we consider the case of low temperature, where we assume that only the lowest vibrational level of the initial electronic state will be occupied. As a consequence, we only need to consider transitions between the vibrational states 0- n . The 0-0 transition is the zero phonon line, ZPL.

A symmetry analysis for the inverse cubic perovskite structure of LiMH_3 ($M = \text{Sr, Ba}$) in space group type $Pm\bar{3}m$ (corresponding to point group O_h in Schönflies notation) yields one acoustic mode of the irreducible representation T_{1u} and four optical modes,^{55–58} which are all triply degenerate. Three of the optical modes belong to the IR active representation T_{1u} , while one mode is hyper-Raman-active and belongs to the representation T_{2u} . The representations Γ of the acoustic and optical modes are given by:

$$\Gamma_{\text{acoustic}} = T_{1u} \quad (2)$$

$$\Gamma_{\text{optic}} = T_{2u} + 3T_{1u} \quad (3)$$

Since we can expect Eu^{2+} to occupy the strontium or barium site $1b$, its surrounding consists of a cuboctahedron and consequently, the lowest lying $5d^1$ level will be the two-fold-

degenerate 2E_g state. To determine if an optical transition is electric-dipole-allowed, the matrix element from Eq.1 can be evaluated as a product of irreducible representations with $\vec{\mu}$ transforming as T_{1u} .^{19,59} Since in our case the $\text{Eu}^{2+} 4f^7 - 5d^1$ transition is parity allowed, we can assume that this product contains A_{1g} :

$$[\Gamma_{\text{electronic ground state}} \times T_{1u} \times \Gamma_{\text{electronic excited state}}] \supset A_{1g} \quad (4)$$

The matrix element of a vibronic transition that couples with phonons belonging to the representation T_{1u} corresponds to the product:

$$[\Gamma_{\text{electronic ground state}} \times T_{1u} \times T_{1u} \times \Gamma_{\text{electronic excited state}} \times T_{1u}] \quad (5)$$

and since

$$[T_{1u} \times T_{1u}] \supset A_{1g} \quad (6)$$

we conclude that the product in Eq. 5 contains A_{1g} and therefore, the vibronic transition is allowed.

If we repeat this procedure for the hyper-Raman mode that belongs to the representation T_{2u} , we find that a transition involving a coupling of the electronic transition and the hyper-Raman mode belonging to T_{2u} is forbidden. In the photoluminescence spectra of the Eu^{2+} -doped perovskites showing the electric dipole $4f^7 ({}^8S_{7/2}) - 4f^6 5d^1 (e_g)$ transition, we therefore expect to only observe a coupling of the electronic transition with the IR-active modes of the representation T_{1u} .

Photoluminescence emission spectra

As already reported in previous results,²⁴ for $M = \text{Sr}$, the inverse perovskites $\text{LiMH}_3\text{:Eu}^{2+}$ and $\text{LiMD}_3\text{:Eu}^{2+}$ show an intense broad band emission in the yellow, whereas the barium

compounds emit in the green. Below approximately 80 K, a vibrational fine structure starts to appear, which becomes very well resolved at low temperature. Decay times were in the range of those reported in literature²⁴ or slightly longer, which is probably due to the low europium concentrations. In Fig.2, the emission spectra at 20 K under laser excitation at 350 nm are shown.

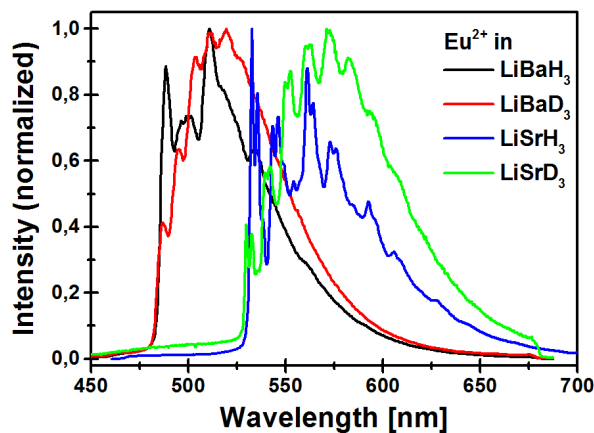


Figure 2: Photoluminescence emission spectra of Eu^{2+} (0.005%) in LiMH_3 and LiMD_3 ($M = \text{Sr, Ba}$) at 20 K. $\lambda_{ex} = 350$ nm.

Doped with only 0.005 % Eu^{2+} , all compounds show more resolved vibrational fine structures than observed in previous results with higher doping levels,²⁴ and the resolution is especially good for $M = \text{Sr}$. As observed before, the vibronic fine structures are more pronounced for the hydrides than the deuterides. To determine the phonon energies, the energy of the zero phonon line was subtracted from the spectra. The energies of the vibronic lines are listed in Table 1 and a pictorial assignment is shown for $\text{LiSrH}_3:\text{Eu}^{2+}$ in Fig.3.

The well-resolved vibronic pattern shows a coupling with at least three vibrations. For $\text{LiSrH}_3:\text{Eu}^{2+}$, three lines separated by about 100 cm^{-1} are repeated at about 370 cm^{-1} . The 100 cm^{-1} line is not affected by replacing hydride by deuteride. Upon substitution of hydrogen by deuterium, we expect vibration frequencies to decrease by the square root of the reduced mass ratio, e.g. approximately square-root of two, if hydrogen or deuterium atoms are strongly involved in these modes. We therefore conclude that mainly the metal atoms contribute to the mode. At approximately 960 cm^{-1} , another strong line is observed,

which shows replicas separated by approximately 100 cm^{-1} . Additionally, overtones involving further combinations of the modes are found. We have labeled the lines using the notation ν_{x-y-z} , where x, y and z stand for the number of vibrations involved in the 100 cm^{-1} (x), 370 cm^{-1} (y), and 960 cm^{-1} mode. Based on our considerations regarding the allowed coupling of modes to the electronic transition in Section Vibronic Transitions - Theoretical Considerations, we assume that the 370 and 960 cm^{-1} mode belong to the optical T_{1u} representations. A comparison with vibrational spectra in the literature⁵⁸ suggests that one additional mode belonging to one of the optical T_{1u} representation should be found around 1100 cm^{-1} . However, in this spectral region several lines from different overtones coincide, so that it is difficult to correctly assign the fourth allowed vibration. Thus, we limited the assignment to the vibrational couplings we believe that can be safely identified. We repeated the assignment for $\text{LiSrD}_3\text{:Eu}^{2+}$ as well as the barium compounds. In the later, we refrained from assigning the higher overtones, since the spectra are less well resolved.

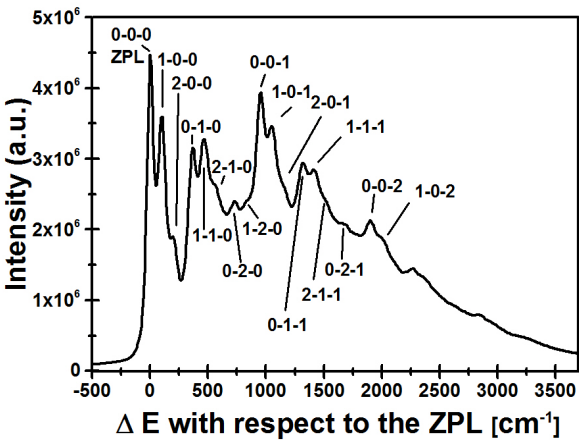


Figure 3: Photoluminescence emission spectrum of Eu^{2+} (0.005%) in LiSrH_3 at 20 K. $\lambda_{ex} = 350\text{ nm}$. Energies ΔE are given with respect to the zero phonon line.

Table 1: Vibronic transitions in the photoluminescence emission spectra at 20 K for the Eu^{2+} transition $4f^65d^1-4f^7(^8S_{7/2})$ in LiMH_3 and LiMD_3 ($\text{M} = \text{Sr}, \text{Ba}$) (Eu^{2+} 0.005 %). Energies are given with respect to their zero phonon lines. Assignments in LiSrH_3 were made based on very well-resolved spectra, whereas the spectra for $\text{M} = \text{Ba}$ are less well resolved, so that larger errors in the correct assignment may be expected.

Line	LiSrH_3 [cm^{-1}]	LiSrD_3 [cm^{-1}]	LiBaH_3 [cm^{-1}]	LiBaD_3 [cm^{-1}]
ν_{1-0-0}	105	105	92	92
ν_{2-0-0}	200	200	183	183
ν_{0-1-0}	367	258	340	270
ν_{1-1-0}	468	351	430	366
ν_{2-1-0}	568	440	530	450
ν_{0-2-0}	735	518	715	560
ν_{1-2-0}	835	618	-	-
ν_{0-0-1}	956	690	890	630
ν_{1-0-1}	1060	788	990	719
ν_{2-0-1}	1170	894	1080	815
ν_{0-1-1}	1320	950	1226	-
ν_{1-1-1}	1418	1040	-	-
ν_{2-1-1}	1515	1127	-	-
ν_{0-2-1}	1683	1200	-	-
ν_{0-0-2}	1902	1380	-	-
ν_{1-0-2}	2000	-	-	-

Inelastic Neutron Scattering

In contrast to the vibrational coupling in the electronic spectra, no selection rules hold for inelastic neutron scattering and INS is also not restricted to the center of the Brillouin zone. Because of the large incoherent neutron scattering cross section of hydrogen, it is an especially powerful tool for studying hydrides. According to Ref. 60, the incoherent neutron scattering cross sections for ^1H , ^2D , $^{\text{nat}}\text{Li}$, $^{\text{nat}}\text{Sr}$, and $^{\text{nat}}\text{Ba}$ are 80.27(6), 2.05(3), 0.92(3), 0.06(11), and 0.15(11) barn. However, since the cross section for ^1H is so much larger than the other elements involved, vibrations involving a high degree of hydrogen motion in the spectra will lead to enhanced signals, which can act as an effective selection rule.⁶¹ Small amounts of ^1H impurities in $^2\text{D}_2$ might also have a large influence on the spectral properties.

Furthermore, in INS multiple scattering processes (overtones and combinations) involving two or more 1-0 transitions are allowed and may also be observed.

In Fig. 4, the INS spectrum of LiSrH_3 is displayed. Since the samples contained SrH_2 (as detected by PXRD, see Supporting Information), the spectrum of SrH_2 was subtracted from the measured data.

The frequency distribution measured for LiSrH_3 shows vibrational modes at approximately 100 cm^{-1} , 360 cm^{-1} , 560 cm^{-1} , 650 cm^{-1} , 970 cm^{-1} , and 1160 cm^{-1} . The modes at 100 cm^{-1} , 360 cm^{-1} , and 970 cm^{-1} are in very good agreement with the data obtained from the analysis of the vibrational coupling to the Eu^{2+} 4f-5d transition and can be assigned to the modes ν_{1-0-0} , ν_{0-1-0} , and ν_{0-0-1} , where ν_{0-1-0} and ν_{0-0-1} are optical modes belonging to the representation T_{1u} . The mode at 560 cm^{-1} can be assigned to the replica ν_{2-1-0} . Furthermore, the mode at 1160 cm^{-1} is assigned to a further optical mode of the representation T_{1u} , which cannot be clearly identified in the photoluminescence spectrum due to overlaying replicas. Consequently, we assign the mode at 650 cm^{-1} to the optical mode of the representation T_{2u} with Hyper-Raman-activity, which cannot couple to the electric dipole transition of Eu^{2+} .

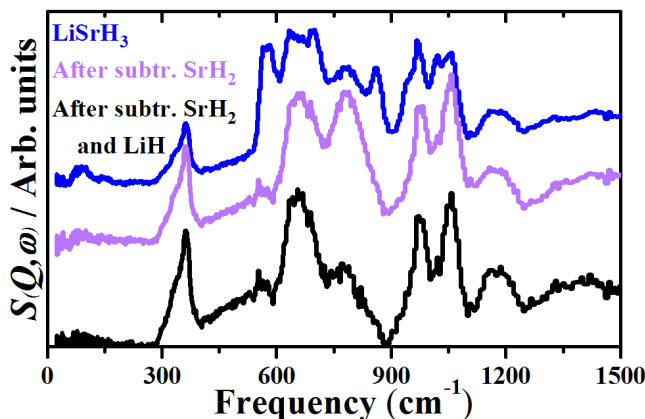


Figure 4: Frequency distributions for LiSrH_3 as measured and LiSrH_3 after subtracting the spectra of SrH_2 and LiH . Recorded at TOSCA at 20 K.

We also studied LiBaH_3 and LiBaD_3 by means of inelastic neutron scattering (data are given in the Supporting Information Figure 1). For both compounds, a mode at 90 cm^{-1} is

found, which is in good agreement with the optical spectra. Furthermore, the ν_{0-1-0} mode can be assigned at 340 cm^{-1} for the hydride and 300 cm^{-1} for the deuteride. For LiBaH_3 , also the ν_{0-0-1} is observed at $\sim 890\text{ cm}^{-1}$. However, the quality of the data is not as good as in case of the strontium compound.

Density Functional Calculations

For further clarifications, we also carried out first-principle calculations using a super cell approach as described in Section Experimental Details - density functional calculations. The equilibrium lattice constants determined from Birch-Murnaghan fourth-order equation of states (EOS) and the corresponding mechanical properties (bulk, shear and Young's moduli) are summarized in Table 2. Results are in good agreement with experiment and other theoretical work.^{62,63}

Table 2: Calculated structural and elastic parameters of BaLiH_3 , BaLiD_3 , SrLiH_3 , SrLiD_3 at zero pressure.

	a_0 (\AA)	B (GPa) ¹	G (GPa) ²	E (GPa) ³	ν ⁴
BaLiH₃					
Present paper	4.020	37.59	39.30	87.43	0.112
mBJ & GGA ⁶³	4.005	37.536	38.924		0.115
GGA ⁶²	3.989	37.83	39.99	87.48	0.093
Experimental	4.026				
BaLiD₃					
Present paper	4.020	37.25	39.33	87.27	0.109
Experimental	4.008				
SrLiH₃					
Present paper	3.808	42.68	46.11	101.70	0.103
mBJ & GGA ⁶³	3.81	42.970	46.019		0.100
GGA ⁶²	3.788	34.15	47.14	103.44	0.096
Experimental	3.835				
SrLiD₃					
Present paper	3.808	42.47	46.12	101.59	0.101
Experimental	3.818				

¹ Bulk, ² Shear and ³ Young's modulus; and ⁴Poisson's ratio

Figs. 5 and 6 show the dispersion and the phonon density of states (phonon DOS) of LiSrH₃ and LiSrD₃. For both the hydride and the deuteride, a low energy acoustic branch is observed, which appears at $\sim 110\text{ cm}^{-1}$ (0 cm^{-1} at the Γ -point). This is in good agreement with the mode found in the inelastic neutron scattering data (100 cm^{-1}). At 360 cm^{-1} , another intense mode can be found in the calculated LiSrH₃ spectra and assigned to the T_{1u} optical representation. This is also in good agreement with the mode found for the vibrational coupling to the Eu²⁺ electronic transition (367 cm^{-1}) and the mode found in the INS spectra (360 cm^{-1}). In the calculated phonon DOS for the deuteride LiSrD₃, the agreement with the vibronic coupling of this mode is not as good as in case of the hydride; the calculated mode around 310 cm^{-1} seems to correspond to the ν_{0-1-0} mode found at 260 cm^{-1} in the experimental data. In the calculated spectra, this corresponds to a decrease in energy of about 13 % from hydride to deuteride. Further modes can be found at $\sim 620\text{ cm}^{-1}$, $\sim 730\text{ cm}^{-1}$, $\sim 920\text{ cm}^{-1}$ and $\sim 1160\text{ cm}^{-1}$ in calculated spectra of LiSrH₃. The mode at 620 cm^{-1} likely corresponds to the mode observed at 650 cm^{-1} by INS. Modes at 920 and 1160 cm^{-1} are likely to correspond to the 970 and 1160 cm^{-1} modes found by photoluminescence and INS measurements.

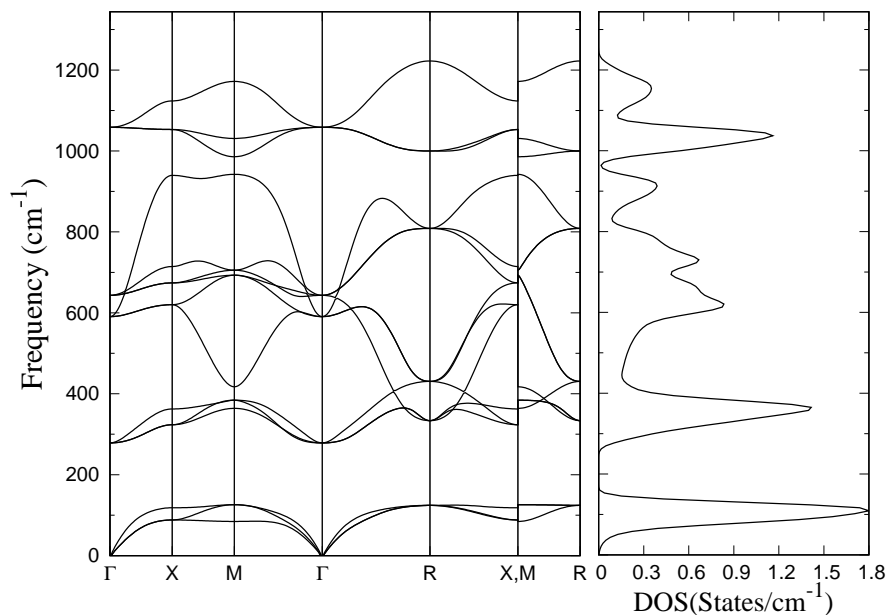


Figure 5: Phonon dispersion and density of states of LiSrH_3 as calculated by using the VASP package.

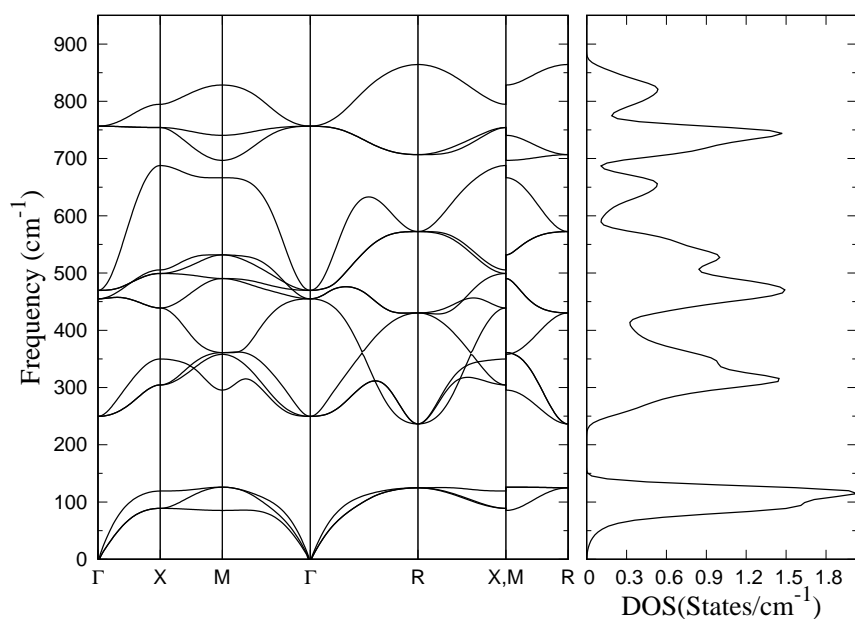


Figure 6: Phonon dispersion and density of states of LiSrD_3 as calculated by using the VASP package.

Comparison of the calculated spectra of the hydride LiSrH_3 and LiSrD_3 (Fig. 7) shows an approximate ratio of $\sqrt{2}$ for the higher energy modes, which is expected for optical

phonons.⁶⁴

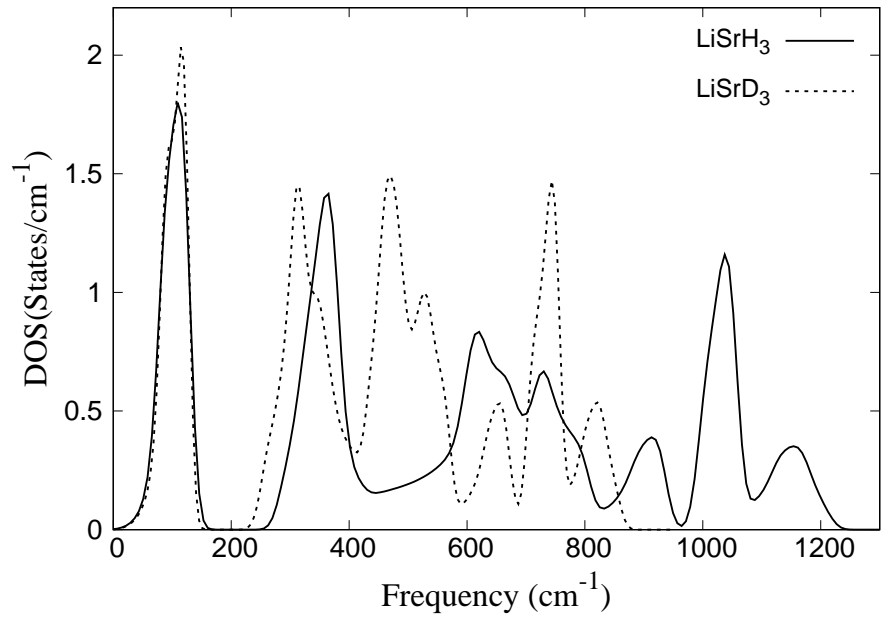


Figure 7: Comparison of the phonon densities of states of the hydride LiSrH₃ and the deuteride LiSrD₃ as calculated by using the VASP package.

The corresponding phonon distributions and phonon densities of states for the barium compounds are shown in the Supporting Information (Supporting Information Figures 2-4). Furthermore, the partial phonons DOS for LiBaH₃ (a), LiBaD₃ (b), LiSrH₃ (c), and LiSrD₃ (d) are shown in Fig. 8. As expected, the heavy atoms barium and strontium have their quasi-global contribution to the acoustical region, whereas the contributions of lithium and hydride are found in the optical region. For deuterides, the lithium vibrations seem to be more spread to higher optical phonon frequencies.

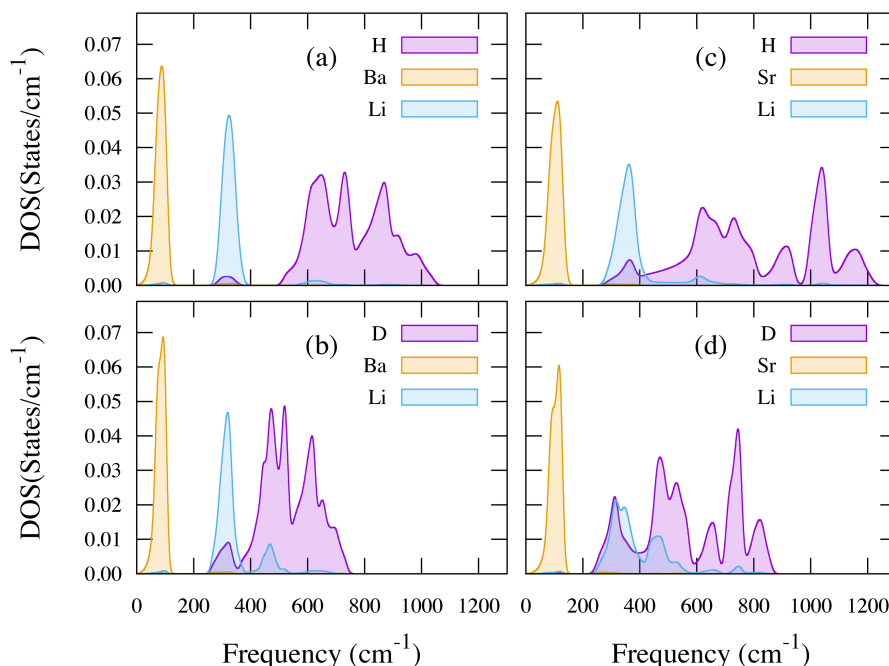


Figure 8: Comparison of the partial phonon densities of states for LiBaH₃ (a), LiBaD₃ (b), LiSrH₃ (c), and LiSrD₃ (d) as calculated by using the VASP package.

Discussion

In order to summarize the results obtained by different methods, a comparison of the values and the assignment of the fundamental modes found in the photoluminescence spectra, the inelastic neutron spectra and the density functional calculations is given in Table 3. While the vibrational modes are clearly resolved in the photoluminescence spectra for the strontium compounds, the resolutions is less good for the barium compounds. Thus, the assignment of the modes for the barium compounds can be expected to be less exact.

Table 3: Comparison of the fundamental modes found in the photoluminescence (PL) emission spectra at 20 K for the Eu^{2+} transition $4f^65d^1-4f^7(^8S_{7/2})$, the inelastic neutron spectra and the density functional calculations given in $[\text{cm}^{-1}]$.

	PL	INS	DFT
LiSrH ₃			
T _{1u} acoustic	105	100	110
T _{1u} optic	367	360	360
T _{2u} optic	-	650	620
T _{1u} optic	956	970	920
T _{1u} optic	-	1160	1160
LiSrD ₃			
T _{1u} acoustic	105	(-)	110
T _{1u} optic	258	(-)	310
T _{2u} optic	-	(-)	460
T _{1u} optic	690	(-)	660
T _{1u} optic	-	(-)	820
LiBaH ₃			
T _{1u} acoustic	92	90	90
T _{1u} optic	340	340	330
T _{2u} optic	-	650	650
T _{1u} optic	890	890	890
T _{1u} optic	-	950	930
LiBaD ₃			
T _{1u} acoustic	92	90	90
T _{1u} optic	270	300	330
T _{2u} optic	-	450	440
T _{1u} optic	630	620	625
T _{1u} optic	-	670	650

As already mentioned in Section Results - Photoluminescence spectra, comparison of the vibrations in the different compounds show that the mode ν_{1-0-0} and its replicas do not show any decrease by a factor of $\sqrt{2}$ upon replacing hydride by deuteride. We explain this observation by the fact that acoustic modes will be less affected by isotopic effects than optical modes,⁶⁴ because the isotopic shift will be proportional to the total mass of all the ions in the unit cell since they all participate in the mode, thus deuteration will have a negligible effect on the transition energy. However, there is a significant change in the intensity of the

zero phonon line when hydrides and deuterides are compared. Moreover, the 100 cm^{-1} mode in the photoluminescence spectra seems to correspond to an acoustic mode, even though it is unexpected to observe it in the optical spectra. The nature of these transitions remains unclear. For the higher energy modes ν_{0-1-0} , ν_{0-0-1} and their replicas, a decrease by a factor of $\sqrt{2}$ upon replacing hydride by deuteride can be approximately observed. For example, in the optical spectra we find about 370 cm^{-1} for ν_{0-1-0} in $\text{LiSrH}_3\text{:Eu}^{2+}$ and only $\sim 260\text{ cm}^{-1}$ in the corresponding deuterides. In the optical spectra of the barium compound, the ν_{0-1-0} mode can be found at 340 cm^{-1} for the hydride and $\sim 270\text{ cm}^{-1}$ for the deuteride. For the third mode ν_{0-0-1} , $\sim 960\text{ cm}^{-1}$ are found in $\text{LiSrH}_3\text{:Eu}^{2+}$ and 690 cm^{-1} for $\text{LiSrD}_3\text{:Eu}^{2+}$. In the barium compound, ν_{0-0-1} appears at 890 cm^{-1} in the hydride and 630 cm^{-1} in the deuteride. In conclusion, we assign the optical mode in LiSrH_3 at 370 cm^{-1} , which can also be observed both in INS and photoluminescence data, to a Sr-(LiH₃)-stretch mode (T_{1u}) and the mode at 1160 cm^{-1} , which could only clearly be identified in the INS data, to a Li-H-stretch mode (T_{1u}). Both assignments are in good agreement with literature data for cubic perovskites.^{57,58} For the two remaining modes at 650 (only observed in the INS, not the optical spectra) and 970 cm^{-1} , which both correspond to Li-H-bending modes, we propose a different assignment. In contrast to earlier suggestions⁵⁸ and based on the selection rules for the coupling of vibrational modes to the electric dipole 4f-5d transition of Eu^{2+} , we suggest to assign the mode at 650 cm^{-1} to the T_{2u} representation and the mode at 970 cm^{-1} to T_{1u} . The assignment in the deuteride and the barium compounds can be carried out accordingly, however, the resolution of the optical spectra is not as good as for the strontium compounds and also the quality of our inelastic neutron scattering data is lower. The low energy modes assigned to acoustic modes do not show any decrease by a factor of $\sqrt{2}$ upon replacing hydride by deuteride, while the modes assigned to optical modes do.

Information on the vibrational modes can also be interesting for the understanding of the optical thermal stabilities of the compounds. A comparison of the luminescence quenching temperatures for the different perovskites²⁷ obtained from luminescence emission lifetime

measurements yielded luminescence quenching temperatures ($T_{50\%}$) of 165 K for LiSrH₃, 175 K for LiSrD₃, 210 K for LiBaH₃ and 230 K for LiBaD₃. From the luminescence quenching temperatures ($T_{50\%}$), it is possible to estimate the energy barrier ΔE for thermal quenching (difference between the relaxed lowest d-states and the conduction band) using the simple equation:³⁷

$$\Delta E = \frac{T_{50\%}}{680} [eV] \quad (7)$$

The phonon energies are directly related to the photoionization behavior, since thermal quenching in such systems follows an Arrhenius relation, where the temperature dependence of the luminescence emission intensity is given as follows:^{27,37}

$$I(T) = \frac{I(0)}{1 + \frac{\Gamma_0}{\Gamma_\nu} \exp\left(\frac{-\Delta E}{k_B T}\right)} \quad (8)$$

(Boltzmann's constant k_B , energy barrier for thermal quenching ΔE , radiative decay rate of the Eu²⁺ 5d states Γ_ν , attempt rate for thermal quenching Γ_0). It is well-known that the attempt rate has a similar magnitude as the maximum phonon frequency.³⁸ Consequently, the observed higher phonon energies in hydrides are responsible for the observed slightly faster thermal quenching in hydrides. The situation is less clear for the substitution of strontium with barium. It may be suggested that the slightly higher phonon frequencies in the strontium compounds also play a role, however, the different sizes of the band gaps or the different position of the conduction band, respectively, may have a larger impact.

Conclusions

In summary, the combination of highly well-resolved vibrational fine structures in the Eu²⁺ 5d-4f luminescence emission, especially for LiSrH₃:Eu²⁺, inelastic neutron scattering experiments and density functional calculations allowed the assignment of several vibrational

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2
3 modes and we observed a decrease in the energies of the optic modes upon replacing hydride
4 by deuteride. Since substitution of hydride by deuteride leads to a doubling of the anion
5 mass, hydrides are ideal candidates for such studies and the results show that the combi-
6 nation of resolved fine structures in rare earth doped samples showing photoluminescence
7 and INS, as well as calculations, represent a powerful tool for the investigation of lattice
8 phonons in ionic hydrides. It is also possible to directly relate the photoionization behavior
9 to the changes in the vibrational frequencies which result from the substitution of hydride
10 by deuteride. To gain a better overview, it may be worthwhile to extend the combination of
11 studies using inelastic neutron scattering as well as the determination of optical properties,
12 such as the luminescence quenching temperature, to other hydride and deuteride containing
13 host materials as well as other materials with polarizable anions, such as nitrides, in the
14 future.
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51 Supporting Information Available

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54 The following files are available free of charge.

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56 Structural data for all samples, inelastic neutron data, and calculated phonon dispersions
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and densities of states for LiBaH_3 and LiBaD_3 :

References

- (1) Schnick, W. Shine a Light with Nitrides. *Phys. Stat. Sol. RRL* **2009**, *3*, A113–A114.
- (2) Jüstel, T.; Nikol, H.; Ronda, C. New Developments in the Field of Luminescent Materials for Lighting and Displays. *Angew. Chem. Int. Ed.* **1998**, *37*, 3084–3103.
- (3) Feldmann, C.; Jüstel, T.; Ronda, C. R.; Schmidt, P. J. Inorganic Luminescent Materials: 100 Years of Research and Application. *Adv. Funct. Mater.* **2003**, *13*, 511–516.
- (4) Höppe, H. A. Recent Developments in the Field of Inorganic Phosphors. *Angew. Chem. Int. Ed.* **2009**, *48*, 3572–3582.
- (5) Eliseeva, S. V.; Bünzli, J.-C. G. Rare Earths: Jewels for Functional Materials of the Future. *New. J. Chem.* **2011**, *35*, 1165–1176.
- (6) Bünzli, J.-C.; Piguet, C. Taking Advantage of Luminescent Lanthanide Ions. *Chem. Soc. Rev.* **2005**, *34*, 1048–1077.
- (7) Binnemans, K. Lanthanide-Based Luminescent Hybrid Materials. *Chem. Rev.* **2009**, *109*, 4283–4374.
- (8) Bünzli, J.-C. G.; Eliseeva, S. V. Intriguing Aspects of Lanthanide Luminescence. *Chem. Sci.* **2013**, *4*, 1939–1949.
- (9) Müller-Buschbaum, K.; Beuerle, F.; Feldmann, C. MOF Based Luminescence Tuning and Chemical/Physical Sensing. *Microporous Mesoporous Mater.* **2015**, *216*, 171–199.
- (10) Thiel, C. W.; Böttger, T.; Cone, R. L. Rare-Earth-Doped Materials for Applications in Quantum Information Storage and Signal Processing. *J. Lumin.* **2011**, *131*, 353–361.

- (11) Goldner, P.; Ferrier, A.; Guillot-Noël, O. In *Handbook on the Physics and Chemistry of Rare Earths*; Bünzli, J.-C. G., Pecharsky, V. K., Eds.; North Holland, Amsterdam, 2015; Chapter Rare Earth-Doped Crystals for Quantum Information Processing, pp 1–78.
- (12) Kunkel, N.; Goldner, P. Recent Advances in Rare Earth Doped Inorganic Crystalline Materials for Quantum Information Processing. *Z. Anorg. Allg. Chem.* **2018**, *644*, 66–76.
- (13) Macfarlane, R. M. High-Resolution Laser Spectroscopy of Rare-Earth Doped Insulators: a Personal Perspective. *J. Lumin.* **2002**, *100*, 1–20.
- (14) Blasse, G.; Grabmeier, B. C. *Luminescent Materials*; Springer, Berlin, 1994.
- (15) Dorenbos, P. Energy of the First $4f^7$ - $4f^65d$ Transition of Eu^{2+} in Inorganic Compounds. *J. Lumin.* **2003**, *104*, 239–260.
- (16) Dorenbos, P. Ce^{3+} 5d-Centroid Shift and Vacuum Referred 4f-Electron Binding Energies of all Lanthanide Impurities in 150 Different Compounds. *J. Lumin.* **2013**, *135*, 93–104.
- (17) Ellens, A.; Meijerink, A.; Blasse, G. ^6I Emission and Vibronic Transitions of Eu^{2+} in KMgF_3 . *J. Lumin.* **1994**, *59*, 293–301.
- (18) Meijerink, A. Spectroscopy and Vibronic Transitions of Divalent Europium in LiBaF_3 . *J. Lumin.* **1993**, *55*, 125–138.
- (19) Blasse, G. Vibronic Transitions in Rare Earth Spectroscopy. *Int Rev. Phys. Chem.* **1992**, *11*, 71–100.
- (20) Tanner, P. A.; Mak, C. S. K.; Edelstein, N. M.; Murdoch, K. M.; Liu, G.; Huang, J.; Seijo, L.; Barandiarán, Z. Absorption and Emission Spectra of Ce^{3+} in Elpasolite Lattices. *J. Am. Chem. Soc.* **2003**, *125*, 13225–13233.

- (21) Duan, C. K.; Tanner, P. A.; Meijerink, A.; Makhov, V. 4f5d Transitions of Tb³⁺ in Cs₂NaYF₆: The Effect of Distortion of the Excited-State Configuration. *J. Phys. Chem. A* **2011**, *115*, 9188–9191.
- (22) Ning, L.; Mak, C. S. K.; Tanner, P. A. High-Spin and Low-Spin f-d Transitions of Tb³⁺ in Elpasolite Hosts. *Phys. Rev. B: Condens. Matter Mater. Phys.* **2005**, *72*, 085127.
- (23) Kunkel, N.; Kohlmann, H.; Sayede, A.; Springborg, M. Alkaline-Earth Metal Hydrides as Novel Host Lattices for Eu^{II} Luminescence. *Inorg. Chem.* **2011**, *50*, 5873–5875.
- (24) Kunkel, N.; Meijerink, A.; Kohlmann, H. Bright Yellow and Green Eu(II) Luminescence and Vibronic Fine Structures in LiSrH₃, LiBaH₃ and Their Corresponding Deuterides. *Phys. Chem. Chem. Phys.* **2014**, *16*, 4807–4813.
- (25) Kunkel, N.; Meijerink, A.; Springborg, M.; Kohlmann, H. Eu(II) Luminescence in the Perovskite Host Lattices KMgH₃, NaMgH₃ and Mixed Crystals LiBa_xSr_{1-x}H₃. *J. Mat. Chem. C* **2014**, *2*, 4799–4804.
- (26) Kunkel, N.; Böttcher, R.; Pilling, T.; Kohlmann, H.; Pöpl, A. Eu²⁺-Containing Luminescent Perovskite-Type Hydrides Studied by Electron Paramagnetic Resonance. *Z. Phys. Chem.* **2016**, *230*, 931–942.
- (27) Kunkel, N.; Sontakke, A. D.; Kohaut, S.; Viana, B.; Dorenbos, P. Thermally Stimulated Luminescence and First-Principle Study of Defect Configurations in the Perovskite-Type Hydrides LiMH₃:Eu²⁺ (M = Sr, Ba) and the Corresponding Deuterides. *J. Phys. Chem. C* **2016**, *120*, 29414–29422.
- (28) Marks, S.; Heck, J. G.; Habicht, M. H.; Oña-Burgos, P.; Feldmann, C.; Roesky, P. W. [Ln(BH₄)₂(THF)₂] (Ln = Eu, Yb) - A Highly Luminescent Material. Synthesis, Properties, reactivity, and NMR Studies. *J. Am. Chem. Soc.* **2012**, *134*, 16983–16986.

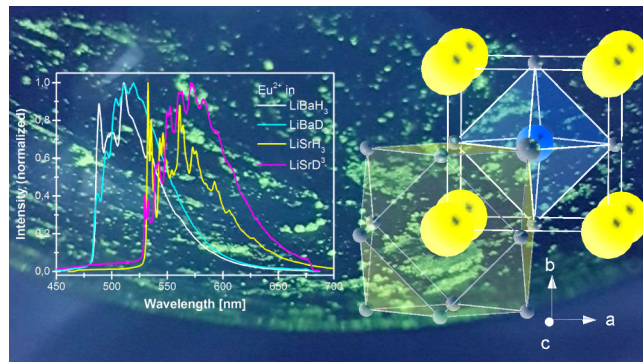
- (29) Schouwink, P.; Ley, M. B.; Tissot, A.; Hagemann, H.; Jensen, T. R.; Smrčok, L.; Černý, R. Structure and Properties of Complex Hydride Perovskite Materials. *Nat. Commun.* **2014**, *5*:5706, 1–10.
- (30) Kunkel, N.; Meijerink, A.; Kohlmann, H. Variation of the Eu^{II} Emission Wavelength by Substitution of Fluoride by Hydride in Fluorite-Type Compounds $\text{EuH}_x\text{F}_{2-x}$ ($0.20 \leq x \leq 0.67$). *Inorg. Chem.* **2014**, *53*, 4800–4802.
- (31) Kunkel, N.; Rudolph, D.; Meijerink, A.; Rommel, S.; Weihrich, R.; Kohlmann, H.; Schleid, T. Green Luminescence of Divalent Europium in the Hydride Chloride EuHCl . *Z. Anorg. Allg. Chem.* **2015**, *641*, 1220–1224.
- (32) Kunkel, N.; Kohlmann, H. Ionic Mixed Hydride Fluoride Compounds: Stabilities Predicted by DFT, Synthesis, and Luminescence of Divalent Europium. *J. Phys. Chem. C* **2016**, *120*, 10506–10511.
- (33) Rudolph, D.; Enseling, D.; Jüstel, T.; Schleid, T. Crystal Structure and Luminescence Properties of the First Hydride Oxide Chloride with Divalent Europium: $\text{LiEu}_2\text{HOCl}_2$. *Z. Anorg. Allg. Chem.* **2017**, *643*, 1525–1530.
- (34) Judd, B. R. Vibronic Contributions to Ligand-Induced Pseudo-Quadrupole Absorption of Rare Earth Ions. *Phys. Scr.* **1980**, *21*, 543–548.
- (35) McCumber, D. E.; Sturge, M. D. Linewidth and Temperature Shift of the R Lines in Ruby. *J. Appl. Phys.* **1963**, *34*, 1682–1684.
- (36) Equall, R. W.; Cone, R. L.; Macfarlane, R. M. Homogeneous Broadening and Hyperfine Structure of Optical Transitions in $\text{Pr}^{3+}:\text{Y}_2\text{SiO}_5$. *Phys. Rev. B: Condens. Matter Mater. Phys.* **1995**, *52*, 3963–3969.
- (37) Dorenbos, P. Thermal Quenching of Eu^{2+} 5d-4f Luminescence in Inorganic Compounds. *J. Phys.: Condens. Matter* **2005**, *17*, 8103–8111.

- (38) Ueda, J.; Dorenbos, P.; Bos, A. J. J.; Meijerink, A.; Tanabe, S. Insight Into the Thermal Quenching Mechanism for $\text{Y}_3\text{Al}_5\text{O}_{12}:\text{Ce}^{3+}$ Through Thermoluminescence Excitation Spectroscopy. *J. Phys. Chem. C* **2015**, *119*, 25003–25008.
- (39) Ronda, C., Ed. *Luminescence. From Theory to Applications*; Wiley-CVH, 2008.
- (40) Greedan, J. E. Synthesis and Crystal Growth of SrLiH_3 and EuLiH_3 Ternary Hydrides with the Perovskite Structure. *J. Cryst. Growth* **1970**, *6*, 119–124.
- (41) Messer, C. E.; Eastman, J. C.; Mers, R. G.; Maeland, A. J. Ternary Perovskite Phases in Systems of Lithium Hydride with Barium, Strontium and Calcium Hydrides. *Inorg. Chem.* **1964**, *3*, 776–778.
- (42) Messer, C. E.; Levy, I. S. Systems of Lithium Hydride with Alkaline Earth and Rare Earth Hydrides. *Inorg. Chem.* **1965**, *4*, 543–548.
- (43) Kohlmann, H.; Yvon, K. The Crystal Structure of EuH_2 and EuLiH_3 by Neutron Powder Diffraction. *J. Alloys Compd.* **2000**, *299*, L16–L20.
- (44) Colognesi, D.; Celli, M.; Cilloco, F.; Newport, R. J.; Parker, S. F.; Rossi-Albertini, V.; Sacchetti, F.; Tomkinson, J.; Zoppi, M. TOSCA Neutron Spectrometer: The Final Configuration. *Appl. Phys. A: Mater. Sci. Process.* **2002**, *74*, S64–S66.
- (45) Parker, S. F.; Fernandez-Alonso, F.; Ramirez-Cuesta, A. J.; Tomkinson, J.; Rudic, S.; Pinna, R. S.; Gorini, G.; Castañon, J. F. Recent and Future Developments on TOSCA at ISIS. *J. Phys. Conf. Series* **2014**, *554*, 012003.
- (46) Barrera, G. D.; Colognesi, D.; Mitchell, P. C. H.; Ramirez-Cuesta, A. J. LDA or GGA? A Combined Experimental Inelastic Neutron Scattering and Ab Initio Lattice Dynamics Study of Alkali Metal Hydrides. *Chem. Phys.* **2005**, *317*, 119–129.

- (47) Colognesi, D.; Barrera, G. D.; Ramirez-Cuesta, A. J.; Zoppi, M. Hydrogen Self-Dynamics in Orthorhombic Alkaline Earth Hydrides Through Incoherent Inelastic Neutron Scattering. *J. Alloys Compd.* **2007**, *427*, 18–24.
- (48) Kresse, G.; Furthmüller, J. Efficient Iterative Schemes for *Ab Initio* Total-Energy Calculations Using a Plane-Wave Basis Set. *Phys. Rev. B: Condens. Matter Mater. Phys.* **1996**, *54*, 11169–11186.
- (49) Kresse, G.; Furthmüller, J. Efficiency of Ab-Initio Total Energy Calculations For Metals and Semiconductors Using Plane-Wave Basis Sets. *Comp. Mater. Sc.* **1996**, *6*, 15–50.
- (50) Hafner, J. *Ab initio* Simulations of Materials Using VASP: Density-Functional Theory and Beyond. *J. Comput. Chem.* **2008**, *29*, 2044–2078.
- (51) Blöchl, P. E. Projector-Augmented Wave Method. *Phys. Rev. B: Condens. Matter Mater. Phys.* **1994**, *50*, 17953–17979.
- (52) Kresse, G.; Joubert, D. From Ultrasoft Pseudopotentials to the Projector Augmented-Wave Method. *Phys. Rev. B: Condens. Matter Mater. Phys.* **1999**, *59*, 1758–1775.
- (53) Perdew, J. P.; Burke, K.; Ernzerhof, M. Generalized Gradient Approximation Made Simple. *Phys. Rev. Lett.* **1996**, *77*, 3865–3868.
- (54) Togo, A.; Tanaka, I. First Principles Phonon Calculations in Materials Science. *Scr. Mater.* **2015**, *108*, 1–5.
- (55) Aroyo, M. I.; Perez-Mato, J. M.; Capillas, C.; Kroumova, E.; Ivantchev, S.; Madariaga, G.; Kirvo, A.; Wondratschek, H. Bilbao Crystallographic Server I: Databases and Crystallographic Computing Programs. *Z. Krist.* **2006**, *221*, 15–27.
- (56) Aroyo, M. I.; Kirov, A.; Capillas, C.; Perez-Mato, J. M.; Wondratschek, H. Bilbao Crystallographic Server II: Representations of Crystallographic Point Groups and Space Groups. *Acta Crystallogr., Sect. A: Found. Crystallogr.* **2006**, *62*, 115–128.

- (57) Hunt, G. R.; Perry, C. H.; Fergunson, J. Far-Infrared and Transmittance of Potassium Magnesium Fluoride and Magnesium Fluoride. *Phys. Rev.* **1964**, *134*, A688–A691.
- (58) Maeland, A. J. Ternary Hydrides Possessing the Cubic Perovskite Structure. II. Vibration Spectra by the Inelastic Scattering of Cold Neutrons. *J. Chem. Phys.* **1969**, *51*, 2915–2919.
- (59) Henderson, B.; Imbusch, G. *Optical Spectroscopy of Inorganic Solids*; Oxford Science Publications, 2010.
- (60) Sears, V. F. Neutron Scattering Lengths and Cross Sections. *Neutron News* **1992**, *3*, 26–38.
- (61) Maeland, A. J. Vibration Spectra of the Orthorhombic Alkaline-Earth Hydrides by the Inelastic Neutron Scattering of Cold Neutrons and Infrared Transmission Measurements. *J. Chem. Phys.* **1970**, *52*, 3952–3956.
- (62) Ghebouli, B.; Ghebouli, M. A.; Fatmi, M. First-Principles Study of Structural, Elastic, Electronic and Optical Properties of Perovskite Hydrides XLiH_3 ($\text{X} = \text{Ba}$ and Sr) Under Pressure. *Eur. Phys. J. Appl. Phys.* **2010**, *51*, 20302.
- (63) Yalcin, B. G.; Salmankurt, B.; Duman, S. Investigation of Structural, Mechanical, Electronic, Optical, and Dynamical Properties of Cubic BaLiF_3 , BaLiH_3 , and SrLiH_3 . *Mater. Res. Expr.* **2016**, *3*, 036301.
- (64) Kittel, C. *Einführung in die Festkörperphysik*; Oldenburg Wissenschaftsverlag, 2013.

TOC graphic



Electron-Phonon Coupling in Luminescent Europium Doped Hydride Perovskites Studied by Luminescence Spectroscopy, Inelastic Neutron Scattering, and First-Principle Calculations

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Table S1: Wyckoff sites for the inverse cubic perovskite structure type of LiMH_3 or LiMD_3 (M = Sr, Ba), space group $Pm\bar{3}m$.

site	site sym.	atom	x	y	z	occ.
$1a$	$m\bar{3}m$	Li	0	0	0	1
$1b$	$m\bar{3}m$	M	1/2	1/2	1/2	1
$3d$	$m\bar{3}m$	H/D	1/2	0	0	1

Table S2: Refined lattice parameters and interatomic distances (in pm) in $\text{LiMH}_3\text{:Eu}^{2+}$ and $\text{LiMD}_3\text{:Eu}^{2+}$, Eu^{2+} 0.005% or 0%, respectively (M = Sr, Ba), space group $Pm\bar{3}m$, X-ray data were collected on a STOE Stadi P powder diffractometer (Stoe & Cie GmbH) with $\text{CuK}_{\alpha 1}$ radiation or a Huber G670 diffractometer with Guinier geometry at T = 297(2) K with $\text{CuK}_{\alpha 1}$ radiation, as well as amount of side product MH_2 or MD_2 in w%. Estimated standard deviations provided by the program TOPAS are given in parenthesis. Calculated values for comparison (Vienna Ab Initio Program Package).

$\text{LiSrH}_3\text{:Eu}^{2+}$ (0.005 mol%)	a = 383.461(2)	
Eu/Sr – H	271.148(2)	
Li – H	191.731(2)	SrH_2 0 w%
$\text{LiSrD}_3\text{:Eu}^{2+}$ (0.005 mol%)	a = 381.765(2)	
Eu/Sr – D	269.948(2)	
Li – D	190.882(2)	SrD_2 0 w%
$\text{LiBaH}_3\text{:Eu}^{2+}$ (0.005 mol%)	a = 402.526(7)	
Eu/Ba – H	284.629(4)	
Li – H	201.263(4)	BaH_2 8.8(9) w%
$\text{LiBaD}_3\text{:Eu}^{2+}$ (0.005 mol%)	a = 400.870(2)	
Eu/Ba – D	283.458(2)	
Li – D	200.435(2)	BaD_2 0 w%
LiSrH_3	a = 382.475(17)	
Sr – H	270.451(12)	
Li – H	191.238(9)	SrH_2 25.2 w%
LiBaH_3	a = 401.370(3)	
Ba – H	283.811(2)	
Li – H	200.069(2)	BaH_2 27.6 w%
LiBaD_3	a = 400.840(4)	
Ba – D	283.437(3)	
Li – D	200.420(2)	BaD_2 48.2 w%
Calculated values (VASP)		
LiSrH_3	a = 380.8	
Sr – H	269.3	
Li – H	190.4	
LiSrD_3	a = 380.8	
Sr – D	269.3	
Li – D	190.4	
LiBaH_3	a = 402.0	
Ba – H	284.3	
Li – H	201.0	
LiBaD_3	a = 402.0	
Ba – D	284.3	
Li – D	201.0	

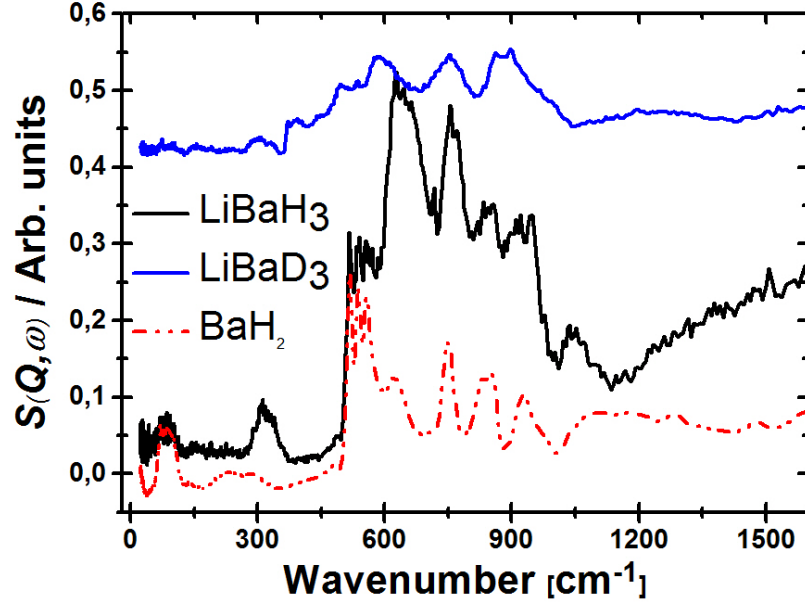


Figure S1: Frequency distributions for LiBaH_3 and LiBaD_3 as measured and BaH_2 for comparison. Recorded at TOSCA at 20 K.

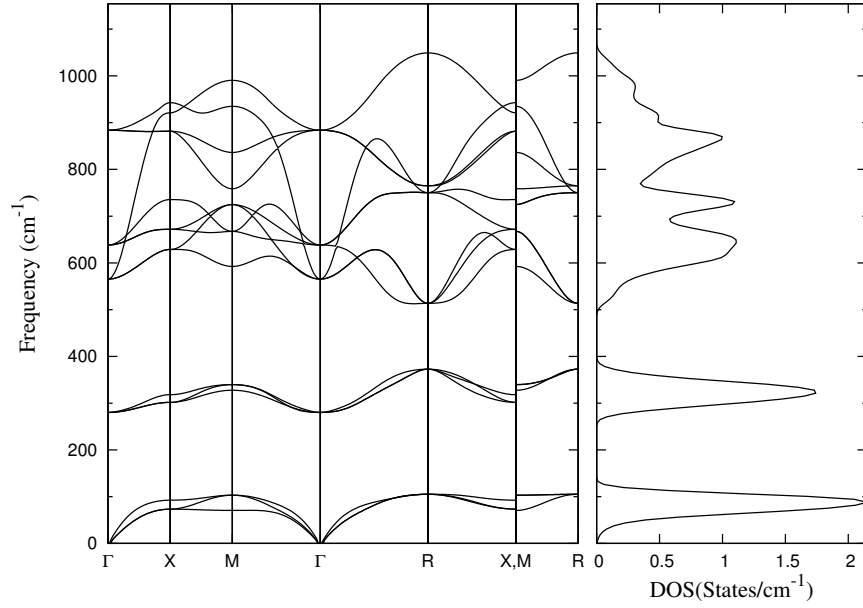


Figure S2: Phonon dispersion and density of states of LiBaH_3 as calculated by using the VASP program package.

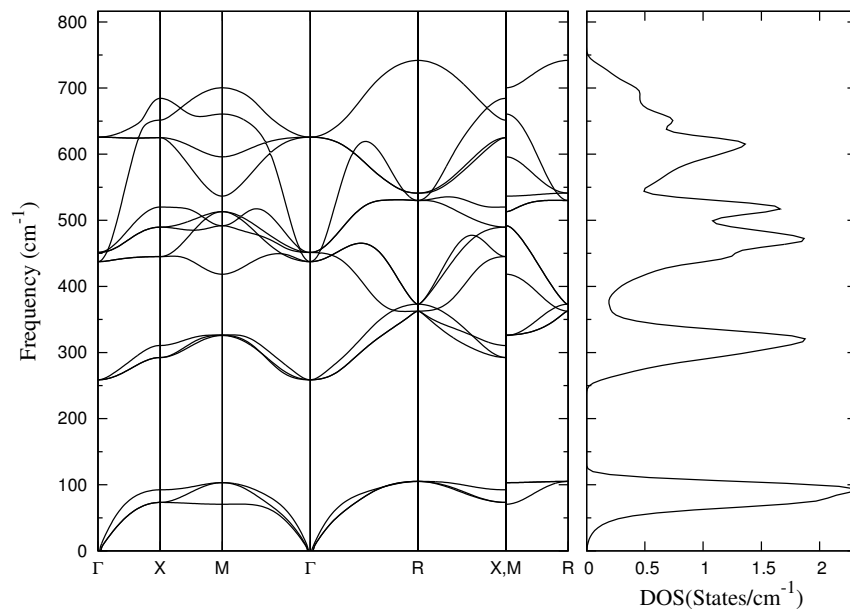


Figure S3: Phonon dispersion and density of states of LiBaD_3 as calculated by using the VASP program package.

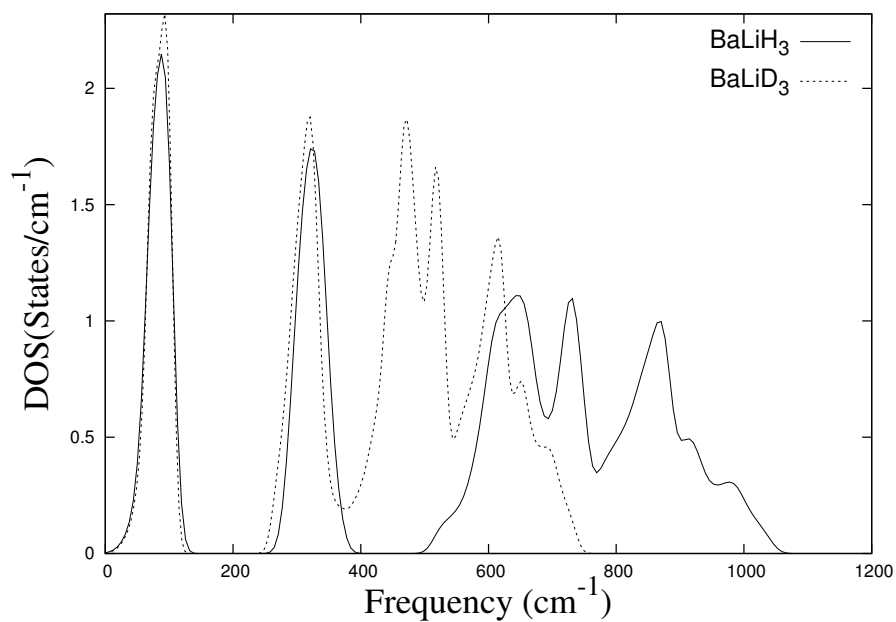


Figure S4: Comparison of the phonon densities of states of the hydride LiBaH_3 and the deuteride LiBaD_3 as calculated by using the VASP program package.