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# Halogenated Metal-Organic Framework Glasses and Liquids

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**ABSTRACT:** The synthesis of four novel crystalline zeolitic imidazolate framework (ZIF) structures using a mixed-ligand approach is reported. The inclusion of both imidazolate and halogenated benzimidazolate-derived linkers leads to glass-forming behavior by all four structures. Melting temperatures are observed to depend on both electronic and steric effects. Solid-state NMR and terahertz (THz)/Far-IR demonstrate the presence of a Zn-F bond for fluorinated ZIF glasses. *In situ* THz/Far-IR spectroscopic techniques reveal the dynamic structural properties of crystal, glass and liquid phases of the halogenated ZIFs, linking the melting behavior of ZIFs to the propensity of the ZnN<sub>4</sub> tetrahedra to undergo thermally-induced deformation. The inclusion of halogenated ligands within MOF-glasses improves their gas uptake properties.

## Introduction

Current research on metal-organic frameworks (MOFs) has concentrated predominantly on the properties of ordered crystalline phases.<sup>1,2</sup> However, there is growing recognition of the importance of the physical properties of MOFs. In particular, the role of disorder, defects and structural flexibility in installing beneficial physical behavior is now widely studied.<sup>3–8</sup> It has in fact recently been demonstrated that crystallinity itself is not a prerequisite for many of the properties for which MOFs are well known, such as porosity, catalytic activity, and chemical and structural tunability.<sup>6,9–12</sup> Consequently, non-crystalline or amorphous MOFs are attracting attention for applications in *e.g.* ion conductivity and molecular encapsulation.<sup>13</sup>

Conventional amorphization of MOFs is achieved through the application of high pressures, thermal treatment or ball milling of crystalline frameworks.<sup>12,13</sup> Recently, the macroscale melting of several zeolitic imidazolate frameworks (ZIFs, including ZIF-4 (**cag**), ZIF-62 (**cag**) and ZIF-76 (**Ita**)) has been reported at *ca.* 650–900 K.<sup>9,14–17</sup> The melting proceeds on a subnanosecond timeframe *via* dissociation of the Zn-Im (Im – imidazolate, C<sub>3</sub>H<sub>3</sub>N<sub>2</sub>) bond, before reassociation of the Im with a different

Zn center. The energy required to break the Zn-Im bond has been previously calculated as *ca.* 95 kJ/mol at 840 K for ZIF-4 [Zn(Im)<sub>2</sub>].<sup>9</sup> The high-temperature liquid MOF exhibits a very high viscosity, which suppresses any re-crystallization of the glasses formed by melt-quenching.<sup>18</sup> The glasses formed upon cooling possess a continuous random network structure, like that of amorphous SiO<sub>2</sub>.<sup>9,13</sup>

Among the ZIF glasses known to date, ZIF-62 [Zn(Im)<sub>x</sub>(bIm)<sub>2-x</sub>] (bIm – benzimidazolate, C<sub>7</sub>H<sub>5</sub>N<sub>2</sub>) is of particular interest because it has the lowest melting temperature.<sup>16,18</sup> The glass, *a<sub>g</sub>*-ZIF-62, exhibits reversible host-guest adsorption behavior, with gas uptakes of 20.1 (CO<sub>2</sub>, 273 K) and 9.3 (H<sub>2</sub>, 77 K) cm<sup>3</sup> STP g<sup>-1</sup> at 1 bar.<sup>19</sup> The liquid MOF phase is however also of interest as it provides a platform from which materials with specific physical and mechanical properties can be designed and shaped.<sup>20</sup> One example of such a material is the MOF crystal-glass composite formed by embedding the as-synthesized phase of MIL-53(Al) (containing unreacted ligand within the pores) within a ZIF-62 glass, which results in stabilization of the high-temperature, open-pore phase of MIL-53(Al) at room temperature.<sup>21,22</sup>

Currently, there are over 80,000 known MOF structures, though melting behavior has mainly been confined to the ZIF family.<sup>13,23</sup> Various strategies have been proposed to construct new systems displaying accessible liquid states<sup>24</sup> or, alternatively, to lower the melting point ( $T_m$ ) of existing systems. Such efforts are highly valuable due to (i) the provision of a wider operational temperature window between melting and decomposition, which allows regulation of liquid viscosity, and (ii) the ability to combine a wider range of crystalline structures within the liquid state due to the lower temperatures needed for crystal-glass composite processing. Our understanding of the various factors which affect melting remains however relatively poor, as does our knowledge of the structure, dynamics and chemical reactivity of the liquid MOF state.

In one of the few studies performed in the area, high-pressure high-temperature diffraction demonstrated the negative melting-curve behavior of ZIF-62.<sup>19</sup> The structure shows a reduction in melting temperature at high pressures (up to 2 GPa), though related materials with accessible dense phases do not show this behavior.<sup>19,25</sup> The melting temperature of ZIF-62 can also be regulated *via* alternating the Im:bIm ligand ratio in the structure.<sup>16,18</sup>

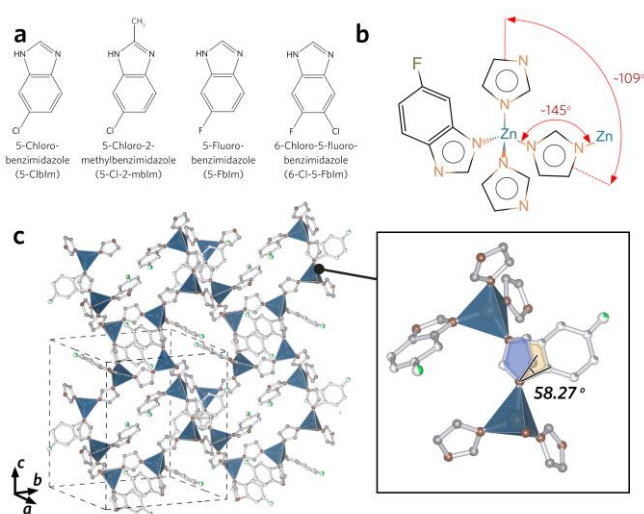
Coudert and co-workers<sup>26</sup> have also investigated the differences in thermal response between ZIF-4 and ZIF-8 [ $\text{Zn}(\text{mIm})_2$ ] (mIm – 2-methylimidazole,  $\text{C}_4\text{H}_5\text{N}_2$ ). The experimental  $T_m$  of ZIF-4 is 863 K, whilst ZIF-8 decomposes before any possible melting event.<sup>14,26</sup> Computationally, the  $T_m$  of ZIF-4 and ZIF-8 were found to be between 1200 – 1500 K and above 1750 K, respectively. This difference was ascribed to a greater energy barrier for diffusion in ZIF-8, which itself was proposed to be a result of the lack of stabilizing dispersive interactions for mIm linkers in the liquid state (due to the more porous structure of ZIF-8 compared to ZIF-4).

Here, motivated by the prospect of expanding the family of ZIF glasses, realizing new methods to regulate  $T_m$  and  $T_g$  (glass transition temperature) and developing a greater understanding of melting behavior and dynamics of the liquid ZIF state, we explore the synthesis of a series of novel glass-forming halogenated ZIF structures. The incorporation of halogenated ligands within MOFs has recently been shown to improve the stiffness of frameworks,<sup>27</sup> whilst they are also capable of introducing permanent dipole moments and facilitating responses to electric fields.<sup>28</sup> The electron withdrawing nature of -F and -Cl functionalized ligands also provides an avenue to regulate the melting behavior of the ZIFs, yet this has not been probed. In this work, four novel mixed ligand ZIFs are constructed using halogenated bIm-based ligands. A combined array of techniques including <sup>19</sup>F solid-state nuclear magnetic resonance (NMR) and *in situ* TeraHertz/Far-Infrared (THz/Far-IR) spectroscopy are used to study the effect of halogenation upon melting behavior, and the dynamic structure of the resultant liquids. The porosity and mechanical behavior of the novel ZIF glasses formed upon melt quenching are characterized, finding improvement over their non-halogenated counterparts.

## Results and Discussion

### Crystalline halogenated ZIF structures

In order to produce ZIF glasses with accessible permanent porosity, we selected 5-chlorobenzimidazole (5-ClbIm –  $\text{C}_7\text{H}_4\text{N}_2\text{Cl}$ ), 5-chloro-2-methylbenzimidazole (5-Cl-2-mbIm –  $\text{C}_8\text{H}_6\text{N}_2\text{Cl}$ ), 5-fluorobenzimidazole (5-FbIm –  $\text{C}_7\text{H}_4\text{N}_2\text{F}$ ),



**Figure 1.** (a) Representations of the halogenated benzimidazolate ligands used in this study. (b) Building unit of ZIF-UC-4 (5-FbIm) with four independent linkers. (c) Representative slice of ZIF-UC-4 (5-FbIm) structure (N, orange; C, grey;  $\text{ZnN}_4$  tetrahedra, blue; F, green), showing the six- and four-membered rings. For clarity, only the range of fractional coordinates from 0.3 to 0.8 along *a* axis is shown, hydrogen and solvent atoms are omitted. Insert highlights a 58.27° orientation deviation of the 5-FbIm linker from the corresponding Im linker position.

and 6-chloro-5-fluorobenzimidazole (6-Cl-5-FbIm –  $\text{C}_7\text{H}_3\text{N}_2\text{ClF}$ ) linkers, alongside Im (Figure 1a). This followed from a previous report of the use of Im to promote melting, and 5-ClbIm as a bulky linker to prevent the collapse of pore channels upon glass formation, allowing the diffusion of guest molecules in and out of the formed quenched glasses.<sup>17</sup> These linkers also provided an array of halogenated functionalities, in order to study the effect on melting behavior.

Four isostructural frameworks based on the glass-forming ZIF-62 structure were synthesized *via* a solvothermal process, adapted from that of ZIF-62 (see **Supporting Information**).<sup>17,21</sup> These materials are referred to as ZIF-UC-2 (6-Cl-5-FbIm), ZIF-UC-3 (5-Cl-2-mbIm), ZIF-UC-4 (5-FbIm) and ZIF-UC-5 (5-ClbIm) respectively. UC indicates the University at which these structures were synthesized. The addition of aqueous sodium hydroxide solution, as reported by others,<sup>29</sup> was found to be beneficial for the inclusion of bIm-based linkers in ZIF-UC-3-5, due to facile deprotonation in solution.<sup>29</sup> After the removal of occluded ligands from the internal pores, single-crystal X-ray diffraction was carried out on suitable crystals. All four frameworks are found to crystallize in the orthorhombic *Pbca* space group with the same 4- and 6-membered ring containing **cag** network topology as ZIF-62 (Figure S1-S13 and Table S1-S8).<sup>30</sup>

Like the reported ZIF-62 structure, the asymmetric units of ZIF-UC-2 to ZIF-UC-5 include two independent Zn atoms, and four organic linkers (Figure 1b). The preference for structural inclusion of the halogenated ligands, described by their total occupancy factors, follows the order 5-FbIm (ZIF-UC-4) > 5-ClbIm (ZIF-UC-5) > 5-Cl-2-mbIm (ZIF-UC-3) > 6-Cl-5-FbIm

**Table 1.** Structural information from single-crystal X-ray diffraction. For solvent occupancy, for ZIF-UC-2 (6-Cl-5-FbIm) and ZIF-UC-4 (5-FbIm), the residual electronic density was too low to assign solvent molecules in the asymmetric unit. ZIF-UC-3 and ZIF-UC-4 were refined considering just DMF solvent molecules because this solvent was used in a higher amount in comparison with DEF (see refinement details in the SI).

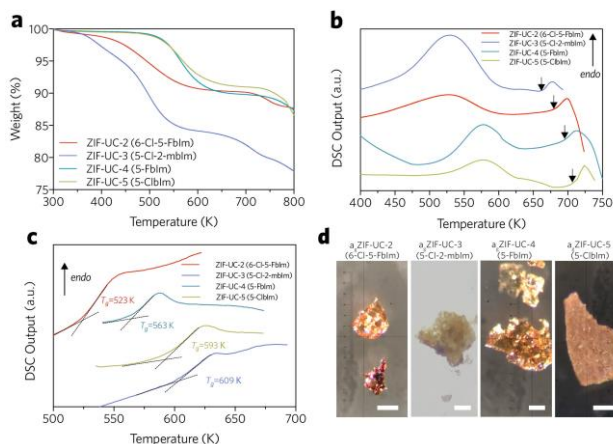
Materials	Cell parameters (Å)			Unit cell volume (Å <sup>3</sup> )	Halogenated linker total occupancy (%)	Solvent occupancy (%)
	<i>a</i>	<i>b</i>	<i>c</i>			
ZIF-UC-2 (6-Cl-5-FbIm)	15.5560(5)	15.6538(5)	18.0875(6)	4404.5(2)	26	-
ZIF-UC-3 (5-Cl-2-mbIm)	15.8749(6)	16.2805(6)	18.0681(6)	4669.7(4)	38	78(3)
ZIF-UC-4 (5-FbIm)	15.6970(8)	15.9694(1)	17.9282(2)	4494.1(5)	68	-
ZIF-UC-5 (5-ClbIm)	15.7260(8)	16.0184(1)	18.1617(1)	4575.0(5)	57	39(9)

(ZIF-UC-2) (**Table 1**). This tendency is consistent with the trend expected for their steric indexes,  $\delta$  (the product of the van der Waals volume of the linker and its size),<sup>31</sup> where bulkier halogenated ligands have lower probabilities of substituting the Im linker in the framework structure. This is consistent with our previous work.<sup>32</sup>

Restrictions on ligand occupancy originate from either the symmetry operations of the specific crystallographic site, or from the steric indexes of the halogenated linkers (see **Supporting Information**). This leads to a distribution of the halogenated ligands in the four structures, which results in different degrees of disorder. In the first case, the corresponding sites have a maximum occupancy of 50%, which prevents symmetry-related bIm-derived linkers coming into contact with each other. In the second case, the increased steric index of bIm-derived linkers relative to Im, results in a shift of the halogenated benzimidazole linker away from the original position to avoid clashing with a neighbor Im linker in a 100% occupied position. This shift results in partial occupancy of the crystallographic sites normally occupied by solvent molecules,<sup>32</sup> reducing the occupancy of the halogenated linker due to the competition with solvent molecules.

Therefore, systems in which the total occupancy factor of the halogenated linker is less than 50% contain just one disordered imidazole position, and only two different configurations for the asymmetric unit (**Figure 1c** and **Figures S1-S13**), *i.e.* ZIF-UC-2 (6-Cl-5-FbIm) and ZIF-UC-3 (5-Cl-2-mbIm) (**Table 1**). Structures with higher occupancies, on the other hand, possess two disordered linker positions. This is the case for ZIF-UC-4 (5-FbIm) and ZIF-UC-5 (5-ClbIm). This results in four possible configurations for the asymmetric unit, creating a higher degree of structural disorder in the system.

The bulk phase purity of the samples was further confirmed by Pawley refinement of powder X-ray diffraction data (PXRD), using lattice parameters from the determined single-crystal structures. Good agreement between the two confirms the formation of phase-pure samples of halogenated ZIFs (**Figure S14**). <sup>1</sup>H solution NMR spectroscopy (**Figure S15-18**) was carried out, by digesting samples of crystalline ZIF-UC-2 to ZIF-UC-5 in a mixture of DCI/D<sub>2</sub>O (35%; 0.1 mL) and DMSO-*d*<sub>6</sub> (0.5 mL). The NCHN chemical shifts for Im (~9 ppm region, NCHN<sub>Im</sub>), the substituted bIm linkers (~9.5 ppm for NCHN<sub>5-ClbIm</sub>, NCHN<sub>5-FbIm</sub>, NCHN<sub>6-Cl-5-FbIm</sub> and ~2.75 ppm for NCHN<sub>3N5-Cl-2-mbIm</sub>) are well resolved. Their relative integrations are used to determine the relative ratio of the two linkers in each mixed-linker MOF, and subsequently the chemical formula for each ZIF structure: ZIF-UC-2 [Zn(Im)<sub>1.84</sub>(6-Cl-5-



**Figure 2.** (a) TGA and (b) DSC profiles of the crystalline powder heating at a rate of 20 K min<sup>-1</sup> under argon. (c) Second upscan DSC profiles of the melt-quenched glass sample at a rate of 10 K min<sup>-1</sup>. (d) Microscopic image of the melt-quenched glass samples. The scale bars are 25 μm.

FbIm)<sub>0.16</sub>·(DMF)<sub>0.10</sub>], ZIF-UC-3 [Zn(Im)<sub>1.74</sub>(5-Cl-2-mbIm)<sub>0.26</sub>·(DEF)<sub>0.08</sub>], ZIF-UC-4 [Zn(Im)<sub>1.63</sub>(5-FbIm)<sub>0.37</sub>·(DEF)<sub>0.07</sub>] and ZIF-UC-5 [Zn(Im)<sub>1.63</sub>(5-ClbIm)<sub>0.37</sub>·(DMF)<sub>0.07</sub>(DEF)<sub>0.04</sub>] (**Table 2**).

### Thermal behavior and glass formation

Thermogravimetric analysis (TGA) carried out on the crystalline samples shows the removal of the remaining solvents (*N,N*-dimethylformamide and *N,N*-diethylformamide) at *ca.* 500 K, followed by thermal decomposition at *ca.* 730 K for ZIF-UC-2 (6-Cl-5-FbIm), 720 K for ZIF-UC-3 (5-Cl-2-mbIm), 750 K for ZIF-UC-4 (5-FbIm) and ZIF-UC-5 (5-ClbIm) (**Figure 2a**). Initial differential scanning calorimetry (DSC) scans on the crystalline samples contains endotherms at *ca.* 530 K corresponding to the loss of solvent from the frameworks, followed by endotherms ascribed to melting starting at *ca.* 679 K for ZIF-UC-2 (6-Cl-5-FbIm), 663 K for ZIF-UC-3 (5-Cl-2-mbIm), 694 K for ZIF-UC-4 (5-FbIm) and 705 K for ZIF-UC-5 (5-ClbIm) (**Figure 2b** and **Table 2**).

In a separate experiment, to avoid excessive thermal decomposition, crystalline samples were heated above their corresponding melting onset temperatures (693 K for ZIF-UC-2 (6-Cl-5-FbIm), 683 K for ZIF-UC-3 (5-Cl-2-mbIm), 708 K for ZIF-UC-4 (5-FbIm) and 723 K for ZIF-UC-5 (5-ClbIm)), and then cooled back to room temperature. The glasses formed are termed a<sub>g</sub>ZIF-UC-2 (6-Cl-5-FbIm), a<sub>g</sub>ZIF-UC-3 (5-Cl-2-mbIm), a<sub>g</sub>ZIF-UC-4 (5-FbIm) and a<sub>g</sub>ZIF-UC-5 (5-ClbIm), with terminology in accordance with previous publications,<sup>32,33</sup>

**Table 2.** Composition and thermal behavior of halogenated ZIF crystals and their melt-quenched glasses.

	ZIF-UC-2 (6-Cl-5-FbIm)	ZIF-UC-3 (5-Cl-2-mbIm)	ZIF-UC-4 (5-FbIm)	ZIF-UC-5 (5-ClbIm)	ZIF-62 <sup>14</sup>
Crystal precursor (Single crystal)	Zn(Im) <sub>1.87</sub> (6-Cl-5-FbIm) <sub>0.13</sub>	Zn(Im) <sub>1.81</sub> (5-Cl-2-mbIm) <sub>0.19</sub>	Zn(Im) <sub>1.66</sub> (5-FbIm) <sub>0.34</sub>	Zn(Im) <sub>1.72</sub> (5-ClbIm) <sub>0.28</sub>	-
Crystal precursor ( <sup>1</sup> H Liquid NMR)	Zn(Im) <sub>1.84</sub> (6-Cl-5-FbIm) <sub>0.16</sub> ·(DMF) <sub>0.10</sub>	Zn(Im) <sub>1.74</sub> (5-Cl-2-mbIm) <sub>0.26</sub> ·(DEF) <sub>0.08</sub>	Zn(Im) <sub>1.63</sub> (5-FbIm) <sub>0.37</sub> ·(DEF) <sub>0.07</sub>	Zn(Im) <sub>1.63</sub> (5-ClbIm) <sub>0.37</sub> ·(DMF) <sub>0.07</sub> (DEF) <sub>0.04</sub>	Zn(Im) <sub>1.75</sub> (bIm) <sub>0.25</sub>
Melt-quenched glass ( <sup>1</sup> H Liquid NMR)	Zn(Im) <sub>1.90</sub> (6-Cl-5-FbIm) <sub>0.10</sub>	Zn(Im) <sub>1.75</sub> (5-Cl-2-mbIm) <sub>0.25</sub>	Zn(Im) <sub>1.63</sub> (5-FbIm) <sub>0.37</sub>	Zn(Im) <sub>1.69</sub> (5-ClbIm) <sub>0.31</sub>	-
Melting onset temperature ( <i>T</i> <sub>m</sub> for crystal)	679 K	663 K	694 K	705 K	673 K
Glass transition temperature ( <i>T</i> <sub>g</sub> for glass)	523 K	609 K	563 K	593 K	591 K
<i>T</i> <sub>g</sub> / <i>T</i> <sub>m</sub>	0.77	0.92	0.81	0.84	0.88

and were used for subsequent *ex situ* studies. This was followed by a second DSC heating scan, at 10 K min<sup>-1</sup>, which identifies *T*<sub>g</sub>s in each case (**Figure 2c** and **Table 2**). All melt-quenched glass samples display only diffuse scattering in their PXRD patterns, whilst optical- and scanning electron microscopy confirm the formation of coherent, continuous and glassy morphologies upon melt quenching (**Figure 2d**, **S19** and **S20**). The Fourier-transform infrared spectroscopy (FTIR) spectra for the crystals and corresponding melt-quenched glasses show similar features (**Figure S21**). Due to the relatively close nature of the melting and thermal decomposition temperatures, the melting process was still however accompanied by a small quantity of thermal decomposition for two of the crystalline samples. Specifically, <sup>1</sup>H NMR spectroscopy carried out on the glasses (**Table 2**) confirms a partial loss of the bIm-derived ligand component in a<sub>g</sub>ZIF-UC-2 [Zn(Im)<sub>1.90</sub>(6-Cl-5-FbIm)<sub>0.10</sub>] and a<sub>g</sub>ZIF-UC-5 [Zn(Im)<sub>1.69</sub>(5-ClbIm)<sub>0.31</sub>]. However, a<sub>g</sub>ZIF-UC-3 [Zn(Im)<sub>1.74</sub>(5-Cl-2-mbIm)<sub>0.26</sub>] and a<sub>g</sub>ZIF-UC-4 [Zn(Im)<sub>1.63</sub>(5-FbIm)<sub>0.37</sub>] are relatively unchanged compared with their crystalline precursors (**Figure S22-25**).

The glass transition indicates the transition from solid to liquid-like behavior upon heating. Glasses with similar organic ligand content possess transition temperatures correlated with the van der Waals radius of the added side-functional group(s); *i.e.* a<sub>g</sub>ZIF-UC-4 (5-FbIm) (563 K and 1.47 Å) and a<sub>g</sub>ZIF-UC-5 (5-ClbIm) (593 K and 1.75 Å). This has also been found previously and is related to larger functional groups increasing the extent of non-covalent interactions within glass, or liquid structures.<sup>14,35</sup> A lower than expected *T*<sub>g</sub> for a<sub>g</sub>ZIF-UC-2 (6-Cl-5-FbIm) was observed at 523 K, which was ascribed to the low bIm-derived linker content in the glass phase.

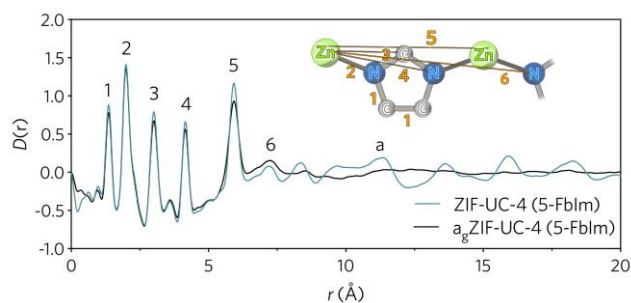
Trends in melting point are less clear. The onset of melting of the structures in this study, including ZIF-62 itself, are separated by 50 K. The 19 K difference between ZIF-UC-4 (5-FbIm) (*T*<sub>m</sub> = 694 K) and ZIF-UC-5 (5-ClbIm) (*T*<sub>m</sub> = 705 K) is consistent with the greater inductive withdrawal effect of -F compared with -Cl. This experimental observation was further confirmed by density functional theory (DFT) calculations carried out (**Figure S26-27**), which demonstrates an average difference of 2.88 kJ/mol difference in Zn-N bond energy between Zn(Im)<sub>3</sub>(5-FbIm) and Zn(Im)<sub>3</sub>(5-ClbIm) tetrahedra.

ZIF-UC-3 (5-Cl-2-mbIm) possesses the lowest *T*<sub>m</sub> observed (663 K), *i.e.* lower than that of ZIF-62 (*T*<sub>m</sub> = 673 K).<sup>14,17</sup> The activation free energy,  $\Delta F^\ddagger$ , for melting has previously been demonstrated to follow the van't Hoff law,  $\Delta F^\ddagger(T) = \Delta U^\ddagger - T \Delta S^\ddagger$ , where  $\Delta U^\ddagger$  and  $\Delta S^\ddagger$  are activation energy and entropy, respectively.<sup>9</sup> From a thermodynamic point of view, the melting occurs through an activation process leading to bond breakage at the point of metastable phase of the ZIF crystal precursor, rather than at the limit of stability of the liquid phase.<sup>36</sup> The ZIF-UC-3 (5-Cl-2-mbIm) structure contains the most sterically large ligand, and thus the crystal structure formed is the most dense out of the four materials studied here. For dense ZIFs such as this one, it follows that greater dispersive interactions between the framework and the partially dissociated ligand will reduce the energy barrier to melting. For highly porous frameworks on the other hand, low level of dispersive interactions between framework and detached ligands generate a high activation free energy barrier for melting.<sup>26</sup>

### Glass structure

The atomic structures of the crystalline ZIFs and their corresponding melt-quenched glasses were probed *via* synchrotron X-ray total scattering. The lack of crystallinity in the glasses is confirmed by the absence of sharp features in their structure factors (**Figure S28-31**). Pair distribution functions (PDFs) for ZIF-UC-4 (5-FbIm) and a<sub>g</sub>ZIF-UC-4 (5-FbIm) (**Figure 3**) were obtained from the total scattering data *via* appropriate corrections and subsequent Fourier transform (see **Supporting Information**). As expected, the crystalline samples contain long-range oscillations in the *D*(*r*) on length scales above 7 Å, while the glass samples are relatively featureless in this region. For example, the peak in the *D*(*r*) for crystalline ZIF-62-5-FbIm at 11.4 Å, marked with an 'a', is associated with the distance between the Zn metal center and the F atom on the next nearest bIm ligand. This feature is absent in the glass phase, consistent with Zn-N bond breakage and rearrangement upon melting. Generally, below 7 Å, the PDF traces are similar for crystal and glass.<sup>18</sup>

Peak assignment was carried out using the PDFGUI software,<sup>37</sup> which was used to calculate partial PDFs for the consistent atom



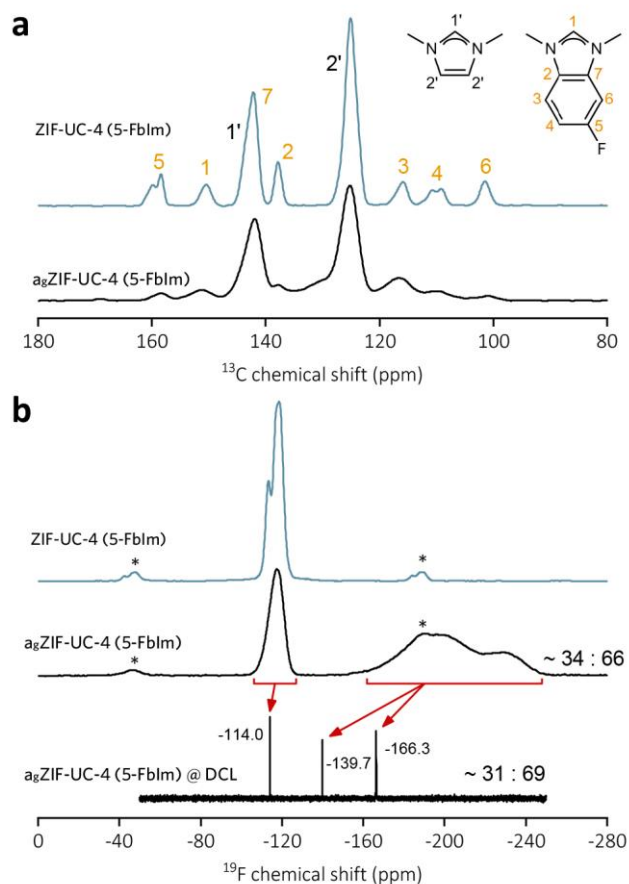
**Figure 3.**  $D(r)$  for ZIF-UC-4 (5-FbIm) and  $a_g$ ZIF-UC-4 (5-FbIm) samples. Insert highlights the peak assignment in the structures.

pairs in each system. For ZIF-UC-4 (5-FbIm), the first feature at 1.3 Å is assigned to C-C or C-N, and the intense peak at 2 Å to the Zn-N distance, which confirms Zn-N association in the glass. Correlations between Zn, and the second and third nearest N atoms of Im ligands, can be identified from the PDFs at 5.9 and 7.3 Å (labelled 4 and 6 in **Figure 3**). These features are relatively unchanged after vitrification suggesting the preservation of this short-range order in the glass (**Figure 3**). Similar PDF results are also obtained for ZIF-UC-2 (6-Cl-5-FbIm), ZIF-UC-3 (5-Cl-2-mbIm) and ZIF-UC-5 (5-ClbIm) (**Figure S32-S34**).

**Figure 4** shows the  $^{13}\text{C}$  and  $^{19}\text{F}$  magic angle spinning (MAS) NMR spectra of ZIF-UC-4 (5-FbIm) and  $a_g$ ZIF-UC-4 (5-FbIm). In the  $^{13}\text{C}$  MAS NMR spectrum of the crystalline material, the signals of all carbon site bIm-derived and Im ligands are clearly resolved. Assignments were performed using chemical shift prediction (ACD/I-lab NMR predictor),  $^1\text{H}$ - $^{13}\text{C}$  Lee-Goldburg cross polarization magic angle spinning (LG-CPMAS) and  $^{19}\text{F}$ - $^{13}\text{C}$  CPMAS measurements (**Figure S35**). The  $^{19}\text{F}$  MAS NMR spectrum of the crystalline sample contains one main peak at *ca.* -115 ppm, which is composed of at least two contributions (**Fig. 4b**). This is explained by the relatively low 5-FbIm/Im ratio present in the material, meaning that two types of Zn center exist: those which are coordinated to one 5-FbIm and three Im ligands, and those which are surrounded by four Im ligands. Hence, there are pores in the material into which two 5-FbIm ligands protrude (the positions of which were related by symmetry, **Figure 1c**), and pores into which only one 5-FbIm ligand protrudes. The resulting local environments of 5-FbIm appear sufficiently different to give rise to resolved  $^{19}\text{F}$  NMR signals.

The  $^{13}\text{C}$  MAS NMR spectrum of  $a_g$ ZIF-UC-4 (5-FbIm) exhibits broader signals at roughly the same isotropic chemical shifts as the spectrum of ZIF-UC-4 (5-FbIm). Broadening of the signals is an expected effect of increased structural disorder and the loss of crystallinity. The  $^{19}\text{F}$  MAS NMR spectrum of  $a_g$ ZIF-UC-4 (5-FbIm) is however more informative. In addition to the signals at about -115 ppm (which are again slightly broader than the corresponding signals in the crystalline sample), several new, broad signals appear between -170 ppm and -240 ppm. NMR measurements show that F nuclei contributing to the narrow signals and F nuclei contributing to the broad signals share the same particles (**Figure 4b**, **Figure S36**).

We therefore ascribe the remarkable change in the chemical shift to a considerable change in the nature of the immediate bonding environment of F upon glass formation.  $^{19}\text{F}$ - $^{13}\text{C}$  CPMAS NMR measurements confirm this (**Figure S37**). Selec-



**Figure 4.**  $^{13}\text{C}$  (a) and  $^{19}\text{F}$  (b) MAS NMR spectra of ZIF-UC-4 (5-FbIm) and  $a_g$ ZIF-UC-4 (5-FbIm). The spectrum at the bottom of (b) is the  $^{19}\text{F}$  solution NMR spectrum of  $a_g$ ZIF-UC-4 (5-FbIm), digested in DCl/DMSO- $d_6$ . Approximate intensity ratios for the signals at about -114.0 ppm and all the other signals in the  $^{19}\text{F}$  spectra of the glass and digested glass are presented. Arrows denote correlations between the  $^{19}\text{F}$  NMR signals of the solid glass and digested glass. Asterisks mark positions of the spinning sidebands.

tive cross-polarization from  $^{19}\text{F}$  nuclei, contributing to the narrow  $^{19}\text{F}$  signal of the glass, clearly enhances the signal of C5 C nuclei, *i.e.* of the nuclei belonging to C atoms within benzyl ring, which are directly bonded to F. On the contrary, selective cross-polarization from  $^{19}\text{F}$  nuclei, contributing to the broad  $^{19}\text{F}$  signal of the glass, gives rise to a C spectrum in which only the two signals of Im can be detected. Surprisingly, this observation suggests that the F that contributes to the broad, shifted signal, is in fact not directly attached to a bIm ligand.

To further verify this observation, the  $^{19}\text{F}$  NMR spectrum of a digested sample of  $a_g$ ZIF-UC-4 (5-FbIm) is measured (**Figure 4b**). The spectrum shows that only a fraction of F nuclei resonate at -114 ppm, *i.e.* at the chemical shift typical for F of the dissolved 5-FbIm. The spectrum contains also a weak signal at -139.7 ppm and a far stronger signal at -166.3 ppm, which can be consistent with F atoms attached to heavy atoms. Based on these observations, we suggest the formation of Zn-F bonds during melting and vitrification. This would explain the observed shift of about -200 ppm in the solid-state  $^{19}\text{F}$  MAS NMR spectrum of  $a_g$ ZIF-UC-4 (5-FbIm), which agrees well with the  $^{19}\text{F}$  shift in  $\text{ZnF}_2$ .<sup>38</sup> Other sources of the shifted signal are investigated, and ruled out, including the presence of paramagnetic

species (through  $^{19}\text{F}$  relaxation measurements), and the presence of halogen bonds (through DFT-based calculations of the expected  $^{19}\text{F}$  chemical shifts) (**Figures S38 and S39**). Very similar observations are found also for  $a_g\text{ZIF-UC-2}$  (6-Cl-5-FbIm) (**Figure S40**). We suspect that the mechanism of dissociation may be similar to thermally activated radical processes in *e.g.* halogenated benzene compounds (*e.g.* chlorobenzene).<sup>39</sup> We might also suggest that the fate of the molecule after defluorination, is to perhaps form a benzyne intermediate, allowing further reaction with imidazole to form heterocyclic structures or, alternatively, to form *N*-Alkyl-*N'*-arylimidazolium salts. This is however far beyond the scope of this work and investigations would be necessary.<sup>40–42</sup>

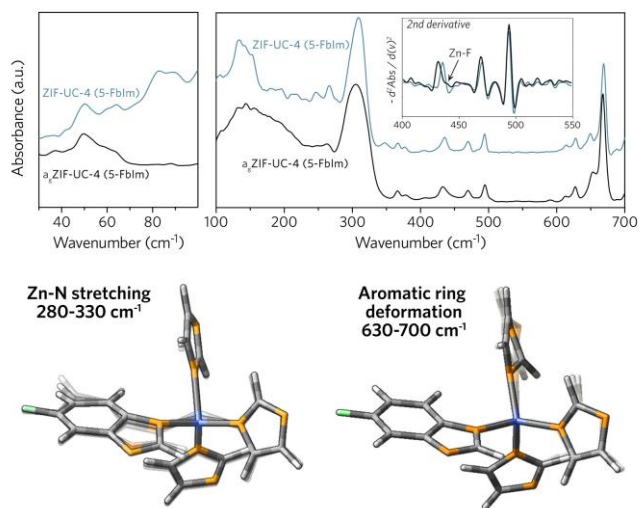
### THz/Far-IR study of bonding environments

While the mid-range IR region is usually associated with fundamental molecular vibrations, THz/Far-IR spectra can reveal spectral features associated with quasi-localized and collective framework dynamic properties, which provides an effective probe to study dynamic framework properties upon glass formation.<sup>43</sup> Synchrotron THz/Far-IR absorption spectroscopy has been shown to be useful in determining the spectra of dynamic vibrational motions up to  $\sim 700\text{ cm}^{-1}$ ,  $\sim 21\text{ THz}$ . The spectra above this region are not shown due to the cut-off by the beam splitter used for this beamline instrument.

Compared to the ZIF precursors ( $\text{Zn}(\text{NO}_3)_2$ , HIm and 5-F-1-HbIm, **Figure S41**), there are three main sources of low-energy vibrations of crystalline ZIF-UC-4 (5-FbIm) within the spectral region below  $700\text{ cm}^{-1}$  (**Figure 5**); in-plane and out-of-plane aromatic ring deformations (*ca.*  $630\text{--}700\text{ cm}^{-1}$ ), Zn-N stretching movements (*ca.*  $280\text{--}330\text{ cm}^{-1}$ ) and 4-membered and 6-membered ring collective motions (*ca.*  $30\text{--}200\text{ cm}^{-1}$ ).<sup>44</sup> The experimental spectrum shows similarities with the theoretical DFT calculations on the  $\text{Zn}(\text{Im})_3(5\text{-FbIm})$  tetrahedron ( $150\text{--}700\text{ cm}^{-1}$ , **Figure S42**). It should be emphasized that for the low-wavenumber region it is difficult to establish a good agreement between experimental and calculated data due to the complexity of the collective framework vibrations ( $23\text{--}100\text{ cm}^{-1}$ , **Figure S43**).<sup>45</sup>

For the melt-quenched glass phase, a clear peak broadening compared with the crystal precursor is observed, especially in the region of  $23\text{--}200\text{ cm}^{-1}$  (**Figure 5**), which has been linked to the collective movement of gate-opening and breathing effects involving 4- and 6-membered rings.<sup>44</sup> Similar behavior is also observed for crystalline and glass ZIF-UC-2 (6-Cl-5-FbIm) (**Figure S44–45**), ZIF-UC-3 (5-Cl-2-mbIm) (**Figure S46–47**), ZIF-UC-5 (5-ClbIm) (**Figure S48–49**) and ZIF-62 (**Figure S50–51**). In addition, the presence of a broad peak centered at *ca.*  $440\text{ cm}^{-1}$ , as suggested by the 2nd derivative spectra in the insert, is consistent with the emergence of a Zn-F bond after vitrification.<sup>46</sup> DFT calculations on a  $\text{Zn}(\text{Im})_3(\text{F})$  tetrahedron confirm this, with good agreement in peak position (*ca.*  $420\text{ cm}^{-1}$ , **Figure S52**). The emergence of a broad peak (*ca.*  $460\text{ cm}^{-1}$ ) assigned to Zn-F is also observed for ZIF-UC-2 (6-Cl-5-FbIm) after vitrification (**Figure S44**), whilst this broad peak is (evidently) absent for the ZIFs which do not contain the fluoro moiety - ZIF-UC-3 (5-Cl-2-mbIm), ZIF-UC-5 (5-ClbIm) and ZIF-62 (**Figure S46, 48 and 50**).

### Dynamic properties of crystal, glass and liquid



**Figure 5.** THz/Far-IR spectra of ZIF-UC-4 (5-FbIm) and  $a_g\text{ZIF-UC-4}$  (5-FbIm) with the insert being the 2nd derivative of the spectra highlighting the emergence of a shoulder peak at around  $440\text{ cm}^{-1}$  after melt quenching, and the schematic diagram of deformation modes of  $\text{ZnIm}_3(5\text{-FbIm})$  tetrahedra and aromatic rings used for DFT calculations. Zn: blue; N: orange; F: green; C: grey; H: white.

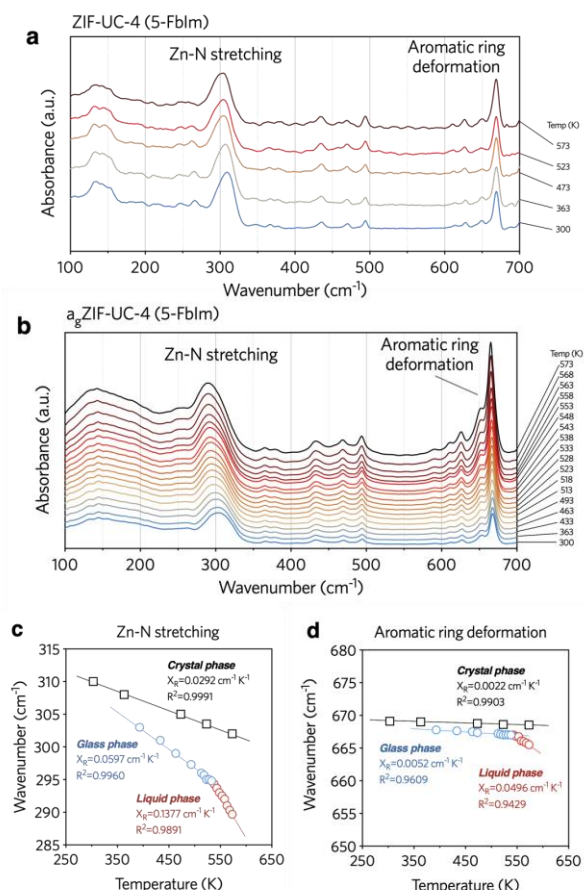
The relatively low glass transition temperature of the halogenated ZIF glasses opens opportunities to probe vibrational properties and low-frequency conformational dynamics in the liquid and glass phases, and compare these to those present in the crystalline solids. *In situ* heating experiments up to  $573\text{ K}$  were performed at the Australian Synchrotron THz/Far-IR beamline. An in-house built gas purge-box was used to maintain an argon environment during the heating process (**Figure S53**). Due to the absence of thermal hysteresis for the THz/Far-IR spectra during heating and subsequent cooling steps (**Figure S54**), only the heating cycle was investigated.

Heating of ZIF-UC-4 (5-FbIm) and  $a_g\text{ZIF-UC-4}$  (5-FbIm) to  $573\text{ K}$  do not lead to a significant change in peak shape, with only a redshift in the peak positions for Zn-N tetrahedra observed (*ca.*  $310\text{--}300\text{ cm}^{-1}$  and  $305\text{--}290\text{ cm}^{-1}$ , **Figure 6a–b** and **Figure S55**). Ryder *et al.* previously demonstrated that the quasi-delocalized  $\text{ZnN}_4$  vibrational modes obey the Bose-Einstein distribution:<sup>47,48</sup>

$$v(T) = v_0 - \frac{X_R T_c}{\exp(\frac{T_c}{T}) - 1} \quad \text{where} \quad T_c = \frac{h\nu_0}{k_B}$$

where  $\nu_0$  is the frequency of the vibration at  $0\text{ K}$ ,  $T_c$  is the vibrational temperature (K), which is obtained from statistical thermodynamics *via* the Planck constant,  $h$ , and the Boltzmann constant,  $k_B$ . In particular, they defined the linear curve fitting slope,  $X_R$  as the spectral position shift coefficient.  $X_R$  was demonstrated to be correlated to the thermally-induced deformation of the  $\text{ZnN}_4$  environment, and thus affected by the exact nature of the coordinative environment. Whilst a detailed quantum mechanical analysis of this Bosonic behaviour lies beyond the scope of this paper, we applied the same logic here to deduce values of  $X_R$  for all crystalline and glass phases.

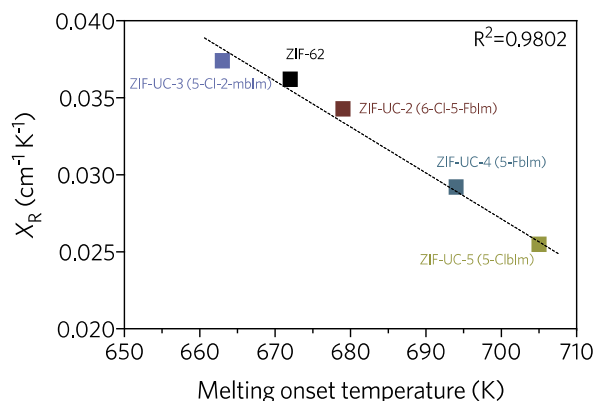
Compared with the crystal precursor, a clear increase of the  $X_R$  value of Zn-N stretching for the glass phase is observed (**Figure 6c**). This is followed by a more significant enhancement above the  $T_g$  ( $563\text{ K}$ ) for the liquid phase, which demonstrates the greater flexibility of the Zn environment in the liquid phase.



**Figure 6.** Synchrotron THz/Far-IR spectra during the heating of ZIF-UC-4 (5-FbIm) and a<sub>g</sub>ZIF-UC-4 (5-FbIm). Temperature-resolved *in situ* THz/Far-IR spectra for (a) ZIF-UC-4 (5-FbIm) and (b) a<sub>g</sub>ZIF-UC-4 (5-FbIm) up to 573 K. (c) and (d) are the peak position shifts for the Zn-N stretching and aromatic ring deformation modes respectively (see text for a detailed explanation of the graphs).

As expected, negligible changes are observed in the deformation-related behavior of the aromatic ring within crystalline ZIF-UC-4 (5-FbIm) upon heating (**Figure 6d**). The value of  $X_R$  is a magnitude smaller than that for the Zn-N bond stretching—attributed to the rigidity of the aromatic ring.<sup>49</sup> However, this same mode in a<sub>g</sub>ZIF-UC-4 (5-FbIm) displays a substantial redshift when the temperature approaches  $T_g$  (ca. 563 K, **Figure 6d**). For example, while the  $X_R$  for Zn-N of a<sub>g</sub>ZIF-UC-4 (5-FbIm) increases by 1.3 times above the  $T_g$ , the a<sub>g</sub>ZIF-UC-4 (5-FbIm)  $X_R$  for aromatic ring deformation increases to nearly 10 times of the original value (0.0052 to 0.0496 cm<sup>-1</sup> K<sup>-1</sup>, **Figure 6d**) upon heating above  $T_g$ .

To further confirm the phenomenon, we conducted *in situ* THz/Far-IR heating tests for all halogenated ZIFs (ZIF-UC-2 (6-Cl-5-FbIm): **Figure S56-58**; ZIF-UC-3 (5-Cl-2-mbIm): **Figure S59-61**; ZIF-UC-5 (5-ClbIm): **Figure S62-64**), together with ZIF-62 (**Figure S65-67**), and observed similar behavior for all crystals and glasses. The  $T_g$  of a<sub>g</sub>ZIF-UC-3 (5-Cl-2-mbIm), a<sub>g</sub>ZIF-UC-5 (5-ClbIm) and a<sub>g</sub>ZIF-62 are all above 573 K, *i.e.* the maximum temperature of this experiment, and thus little variance in  $X_R$  values is noted.<sup>21</sup> However, similar changes in  $X_R$  above the  $T_g$  of ca. 523 K, are observed for a<sub>g</sub>ZIF-UC-2 (6-Cl-5-FbIm). Interestingly, the Zn-N stretching  $X_R$  for different ZIF crystals follows the order of ZIF-UC-3 (5-Cl-2-mbIm)



**Figure 7.** Correlation between the melting onset temperature and  $X_R$ .

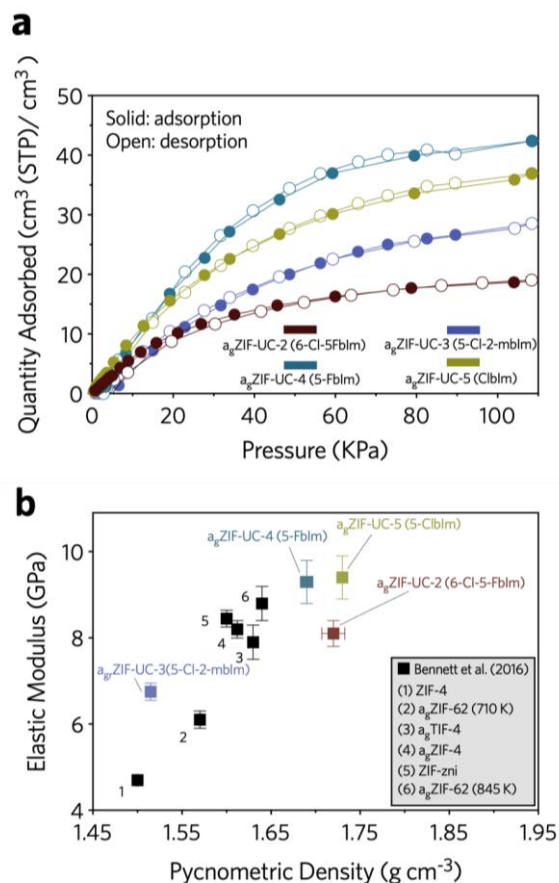
(0.0374 cm<sup>-1</sup> K<sup>-1</sup>) > ZIF-62 (0.0362 cm<sup>-1</sup> K<sup>-1</sup>) > ZIF-UC-2 (6-Cl-5-FbIm) (0.0343 cm<sup>-1</sup> K<sup>-1</sup>) > ZIF-UC-4 (5-FbIm) (0.0292 cm<sup>-1</sup> K<sup>-1</sup>) > ZIF-UC-5 (5-ClbIm) (0.0255 cm<sup>-1</sup> K<sup>-1</sup>), which shows excellent agreement with their melting onset temperatures (**Figure 7**, **Table S9**).

The decrease in melting temperature with greater values of  $X_R$  is consistent with the computationally calculated melting mechanism for ZIFs. For example, melting has been theorized to proceed *via* metal-linker bond breaking at a critical temperature, where the ratio of the thermally-induced vibrational amplitude of N atoms metal-linked atoms (*i.e.* N) and the M-N bond distance reach a critical point, *i.e.* Lindemann's law.<sup>9</sup> Since  $X_R$  describes the increasing deformation of the ZnN<sub>4</sub> tetrahedra, it must also be correlated to both bond distance and the thermal vibration of the ligand. We also observe the melting tendency has no clear correlation with the Zn-N bond strength at room temperature (**Figure S68**).

### Porosity

ZIF-UC-2 (6-Cl-5-FbIm), ZIF-UC-3 (5-Cl-2-mbIm), ZIF-UC-4 (5-FbIm) and ZIF-UC-5 (5-ClbIm) all display a low N<sub>2</sub> uptake at 77 K (**Figure S69**) with relatively low level of BET surface areas, in the range of ca. 7.10 - 86.73 m<sup>2</sup>/g.<sup>50</sup> These values are consistent with the data reported in the literature, where the relatively dense structure of ZIF-62 blocks the entry and exit of N<sub>2</sub> gas molecules into its inner framework structure.<sup>51</sup> On the other hand, smaller gas molecules like CO<sub>2</sub> can diffuse into the crystalline framework, and the absence of adsorption/desorption hysteresis indicated the rapid adsorptive/desorptive kinetics for these crystals (measured at 273 K, **Figure S70**). Our earlier work demonstrates that a reduction in porosity, and thus gas adsorption capacity, is to be expected after vitrification of a crystalline ZIF.<sup>19</sup> A substantial degree of microporosity can however, be preserved due to the presence of bulky bIm-based ligands, which prevents the total collapse of the porous interior upon liquid formation.<sup>11,17</sup> CO<sub>2</sub> adsorption isotherms at 273K for the melt-quenched glass samples confirm this is also the case here (**Figure 8a**): all melt-quenched ZIF glasses show reversible adsorption of CO<sub>2</sub>, suggesting the relatively rapid diffusion of guest molecules within these microporous structures.

The incorporation of fluorinated ligands can improve the CO<sub>2</sub> uptake for both crystalline and glass materials. For example, despite ZIF-UC-4 (5-FbIm) and ZIF-UC-5 (5-ClbIm) exhibiting comparable quantities of bulky bIm-based ligands (**Table 2**),



**Figure 8** (a) CO<sub>2</sub> gas adsorption/desorption isotherm for melt-quenched ZIF glasses. (b) Ashby-style plot of existing MOF glasses in this work, alongside the materials reported.<sup>14</sup>

ZIF-UC-4 (5-FbIm) exhibits a markedly higher CO<sub>2</sub> uptake which is rationalized in terms of the stronger interaction between CO<sub>2</sub> guests and framework F atoms in the narrow channels of the MOF.<sup>52</sup> Similar behavior of other F-functionalized MOFs has been reported by other researchers.<sup>53–55</sup> After melting, a<sub>g</sub>ZIF-UC-4 (5-FbIm) displays a CO<sub>2</sub> uptake of 42 cm<sup>3</sup> (STP) / cm<sup>3</sup> at 1 bar, which is *ca.* 15 % higher than a<sub>g</sub>ZIF-UC-5 (5-ClbIm) and *ca.* 30 % higher than a<sub>g</sub>ZIF-62.<sup>19</sup>

### Mechanical properties

Nanoindentation has been previously applied to investigate the mechanical properties including Young's moduli (*E*) and hardness of crystalline and amorphous MOFs.<sup>14,21</sup> The average Young's modulus was recorded from load-displacement data (200 - 1000 nm) on polished glass samples (**Figure S71-74**). Both a<sub>g</sub>ZIF-UC-4 (5-FbIm) and a<sub>g</sub>ZIF-UC-5 (5-ClbIm) show comparable values of *E*  $\approx$  9.4 and 9.3 GPa, respectively, placing them roughly at the upper bound of the existing ZIF glasses (**Figure 8b** and **Figure S75**). The mechanical properties of MOF materials do not always follow density-stiffness rules, but are determined more by the primary (metal-organic coordination structure) and the secondary network, the nonbonded interactions, which provide extra support for the structural weak points.<sup>56</sup> Consistent with these reports, we postulate that the presence of halogenated moieties on the bIm-derived ligands can introduce more effective secondary network structures *via* nonbonded interactions, or indeed by the Zn-F bonds, which

raise *E* in these cases above that found for a<sub>g</sub>ZIF-62 (*E*  $\approx$  6.1 - 8.8 GPa, **Figure 8b**). Attempts have also been made to prepare bulk halogenated glass samples, by first pelletizing samples of a<sub>g</sub>ZIF-UC-3 (5-Cl-2-mbIm), before annealing above the glass *T<sub>g</sub>*.<sup>57</sup> The resultant bulk glass (termed as a<sub>g</sub>ZIF-UC-3 (5-Cl-2-mbIm)) has a dark brown, opaque appearance, with a diameter of 13 mm (**Figure S76**), and Young's modulus of 6.66  $\pm$  0.14 GPa.

### Conclusion

Four new crystalline ZIFs, ZIF-UC-2 (6-Cl-5-FbIm), ZIF-UC-3 (5-Cl-2-mbIm), ZIF-UC-4 (5-FbIm) and ZIF-UC-5 (5-ClbIm) have been reported and have all been shown to undergo melting. The melting of ZIFs has previously been reported to be affected by the crystal structure of the precursor, the stoichiometric ratio of different ligands, and the type of metal node.<sup>16,18</sup> The array of experimental techniques provided here show that the presence of electron withdrawing moieties on the benzimidazole ligand may weaken the Zn-N bond, and also lead to a small reduction in the melting temperature of the ZIF crystal. The results also show that the presence of bulkier ligands results in a lower free activation energy through dispersive interaction-based stabilization of dissociated ligands in the liquid phase. These findings are encapsulated by *in situ* THz/Far-IR studies, which reveal the connection between Zn-N bond motion and melting behavior. Specifically, they provide evidence of a correlation between the ZIF melting temperature and the propensity of thermal-induced Zn-N polyhedral deformation, which provides experimental evidence of melting mechanism being underpinned by the thermally-induced vibrational amplitude of the N atoms reaching a critical point, which leads to the breaking of the M-N bond.

In addition, solid-state NMR results provide specific evidence for the formation of Zn-F bonding in the glass phase, implying a reduction in overall network connectivity. The extent of this rearrangement process is dependent upon treatment temperature, though the fate of the proportion of fluorobenzimidazole ligands from which F detaches is beyond the scope of this work. The presence of a fluorinated moiety within the glass of ZIF-UC-4 (5-FbIm) also notably improves gas uptake for CO<sub>2</sub>, compared with ZIF-62 and ZIF-UC-5 (5-ClbIm) containing chlorinated benzimidazole ligands.

The results help explain the glass forming ability of ZIF materials, which were analysed using their respective *T<sub>g</sub>* / *T<sub>m</sub>* ratios (**Table 2**). This ratio for conventional metallic, covalent and ionic melt quenched glasses follows the empirical Kauzmann "2/3 Law",<sup>34</sup> and has been used previously to suggest the resistance to recrystallization of the supercooled liquid state.<sup>18</sup> The highest GFA (0.92) for the samples studied here is observed for ZIF-UC-3 (5-Cl-2-mbIm). This is ascribed to the presence of the sterically bulky 5-Cl-2-mbIm ligand, which suppresses the extent of diffusion in the liquid phase. This is higher than the reported GFA for ZIF-62 (0.88). At the same time, slightly lower GFAs for the ZIF-UC-2 (0.77) and ZIF-UC-4 (0.81) are attributed to the formation of Zn-F bonds, which breaks the continuity of the random network and improves the diffusivity of the ligands in ZIF liquid, thus lowering *T<sub>g</sub>*. The GFAs for the halogenated ZIFs are still higher than the conventional ionic, metallic and covalent glasses.

Importantly, together with recent progress on lowering the melting temperature of ZIFs through (i) adjustment of organic linker ratio, (ii) inclusion of multiple organic linkers and (iii) cobalt, these results may be used to further lower the melting temperature of ZIFs. Such research is essential if industrially viable working temperatures are to be achieved and larger scale studies on the applications of such glasses are to be performed.

## ASSOCIATED CONTENT

### Supporting Information

The Supporting Information is available free of charge on the ACS Publications website.

Experimental and characterization procedures. Data on crystal structure, PXRD and Pawley refinement, liquid-state NMR, solid-state NMR, DFT calculation for charge density distribution, SEM image, FTIR, X-ray total scattering, pair distribution function, synchrotron THz/FarIR spectra, gas sorption, nanoindentation and density.

Crystallographic data for ZIF-UC-2 (ClFbIm), ZIF-UC-3 (ClmbIm), ZIF-UC-4 (FbIm) and ZIF-UC-5 (ClbIm).

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